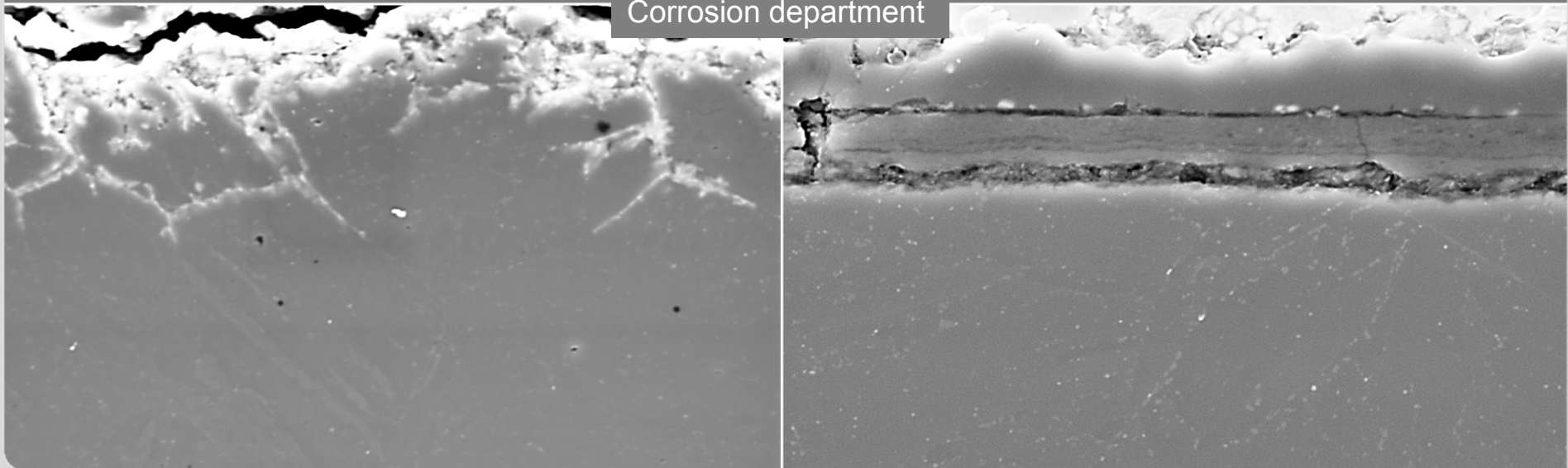


Corrosion of 9%Cr Ferritic / Martensitic Steels at 450 and 550°C in Flowing Pb-Bi Eutectic with 10^{-7} mass% Dissolved Oxygen

Valentyn Tsisar, Carsten Schroer, Olaf Wedemeyer, Aleksandr Skrypnik, Jürgen Konys

INSTITUTE FOR APPLIED MATERIALS – APPLIED MATERIALS PHYSICS (IAM-WPT)

Corrosion department



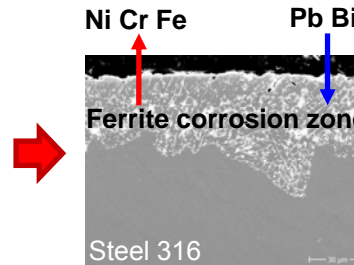
Liquid metal corrosion - background

Issue !

- ❑ Dissolution of Ni, Cr and Fe from the steel by liquid metal:
 - Formation of weak corrosion zone with ferrite structure on austenitic matrix
 - Liquid metal penetrates into the ferrite

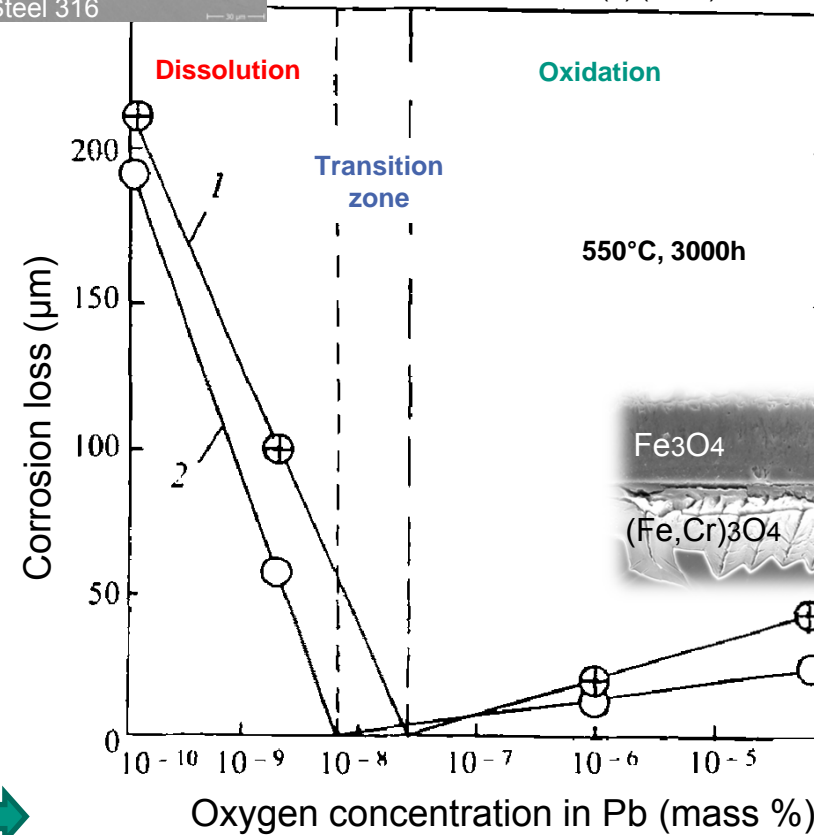
Solution !?

- ❑ Oxidation instead of dissolution:
 - Formation of continuous and protective oxide layer
 - Long-term operation of scale in protective mode



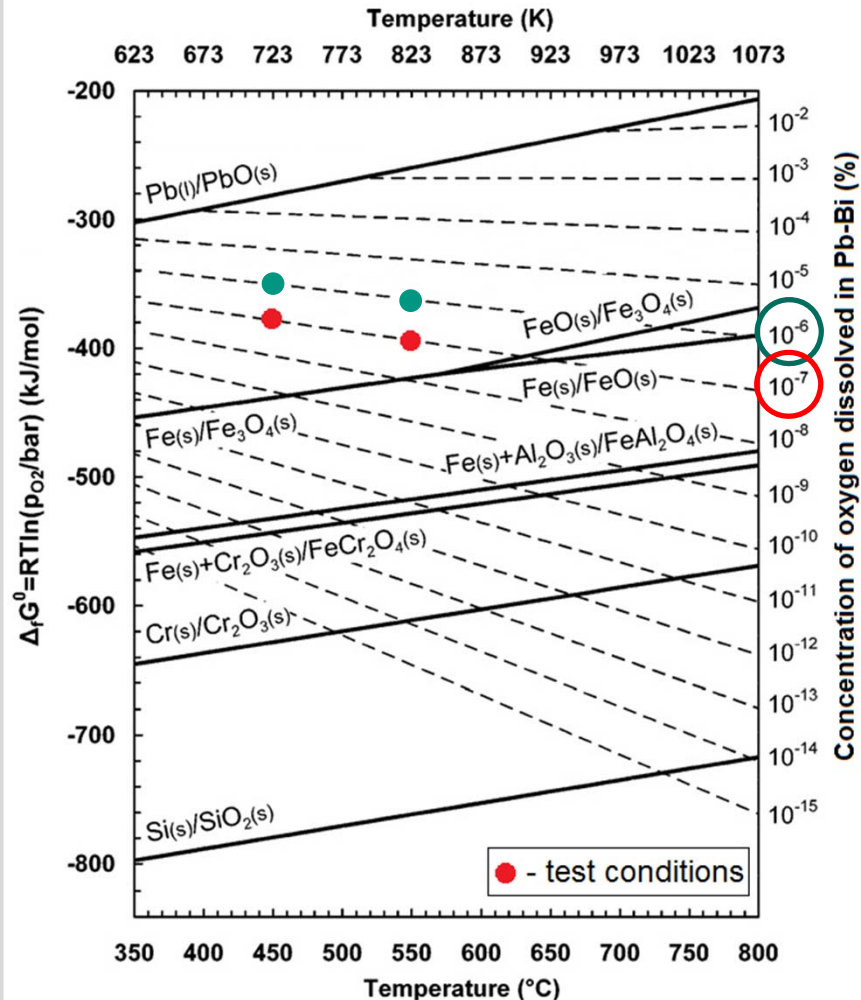
Earlier findings !

I.V. Gorynin et al. Met. Sci. Heat Treat. 41 (9) (1999) 384–388.



1 and 2 – austenitic steels of 316L type

Thermodynamic basis for in-situ addition of oxygen into liquid Pb-Bi



BASIS of Pb-Bi technology

- ❑ Pb-Bi dissolves and transports the oxygen;
- ❑ Components of steels (Si, Cr, Fe...) have high affinity to oxygen than Pb or Bi.
- ❑ **Main aim of the corrosion tests** is to determine the optimum temperature-oxygen concentration parameters for save and long-term operation of structural materials in contact with liquid Pb-Bi eutectic.

Previous test:

- $C_o = 10^{-6}$ mass%, $T = 450$ and 550°C

This work:

- $C_o = 10^{-7}$ mass%, $T = 450$ and 550°C

F/M steels tested in the CORRIDA loop

Concentration (in mass%) of alloying elements other than Fe

	Cr	Mo	W	V	Nb	Ta	Mn	Ni	Si	C
T91-A	9.44	0.850	<0.003	0.196	0.072	n.a.	0.588	0.100	0.272	0.075
T91-B	8.99	0.89	0.01	0.21	0.06	n.a.	0.38	0.11	0.22	0.1025
P92	8.99	0.49	1.75	0.20	0.06	-	0.43	0.12	0.26	0.11
E911*	8.50- 9.50	0.90- 1.10	0.90- 1.10	0.18- 0.25	0.06- 0.10	-	0.30- 0.60	0.10- 0.40	0.10- 0.50	0.09- 0.13
EUROFER	8.82	0.0010	1.09	0.20	n.a	0.13	0.47	0.020	0.040	0.11

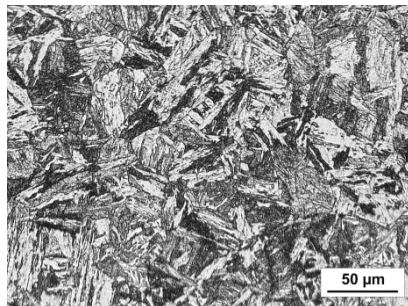
*nominal composition

Nominally 9 mass% Cr

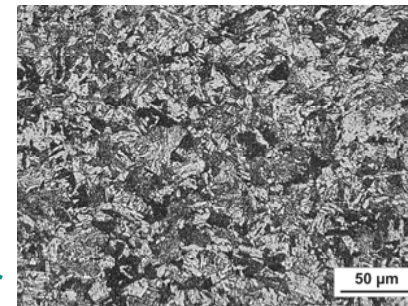
Element besides Cr that improves oxidation resistance

Martensitic microstructure of F/M steels

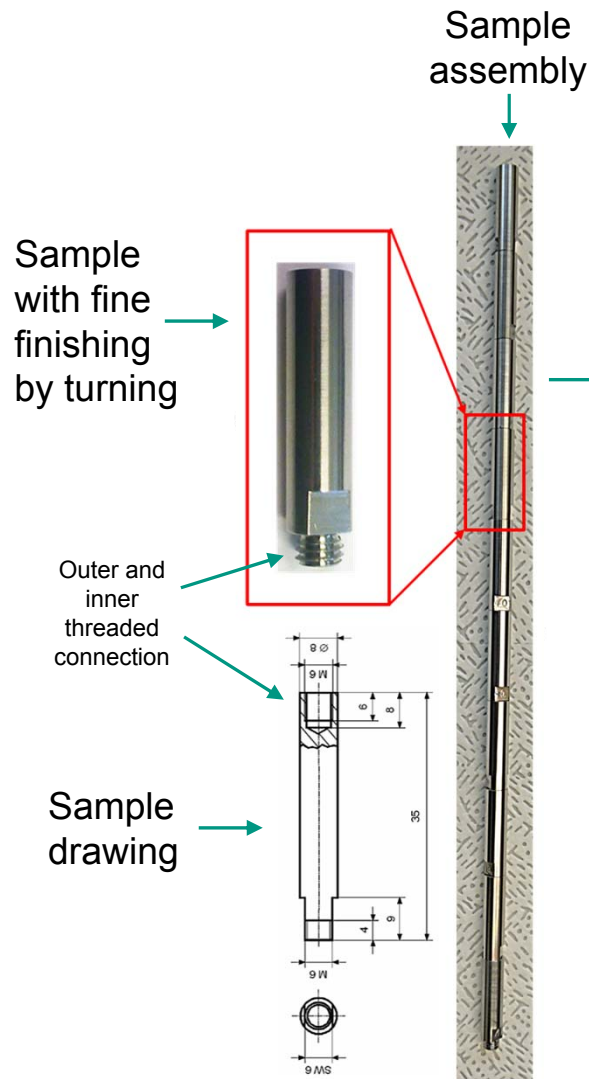
E911,
T91-A,
T91-B,
P92



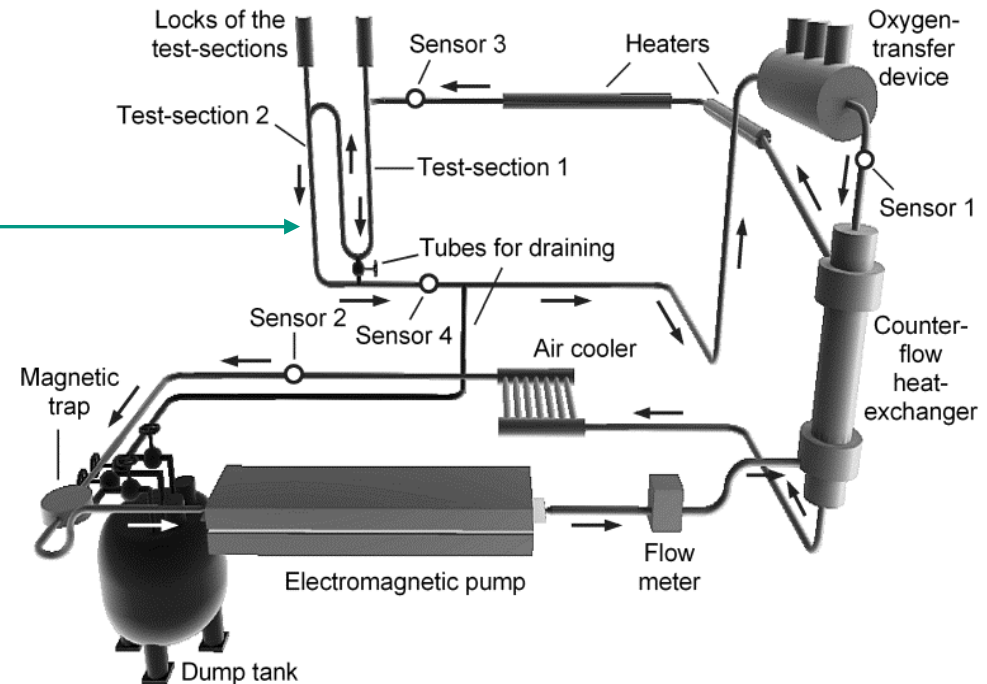
Grain size



EUROFER



CORROsion In Dynamic lead Alloys CORRIDA loop



The CORRIDA facility – a forced-convection loop made of austenitic stainless steel (1.4571) designed to expose material (steel) specimens to flowing (2 m/s) Pb-Bi eutectic (~1000 kg) with controlled oxygen concentration.

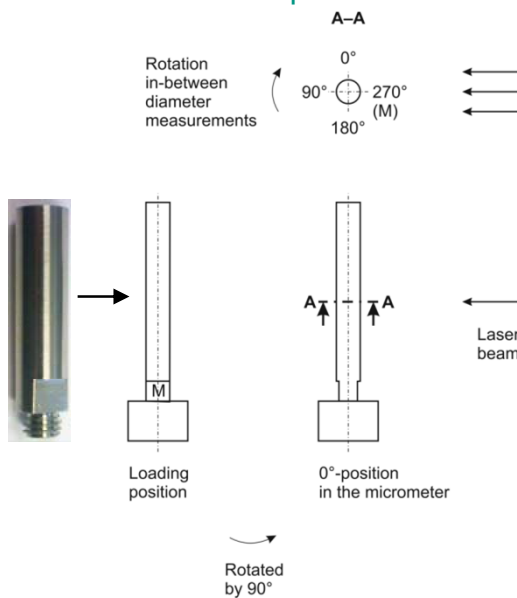
Quantification of corrosion loss

Goal of quantification

- Material loss, average of general corrosion and maximum of local corrosion
- Thickness of adherent (oxide) scale
- Overall change in dimensions, including the scale
- Amount of metals transferred to the liquid metal

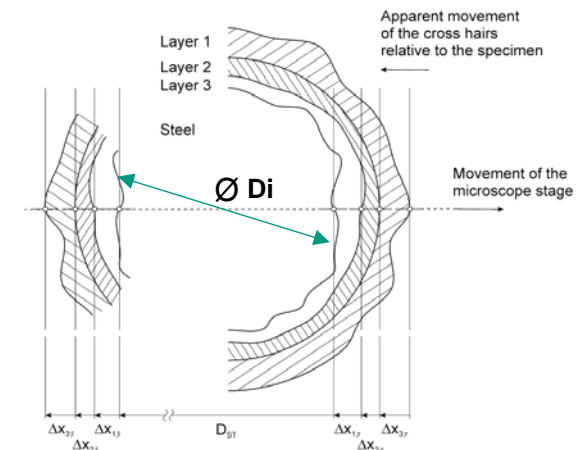
Metallographic method (cylindrical specimens)

Initial parameters - \emptyset



- Measurement of initial diameter in a laser micrometer with 0.1 μm resolution
- Diameter of unaffected material (12th measurements with rotation angle 15°) and thickness of corrosion zones determined in a microscope (LOM) with 1 μm resolution
- Corrosion modes on opposing sides of the re-measured diameter are evaluated (% of surface circumference)

Post-test examination



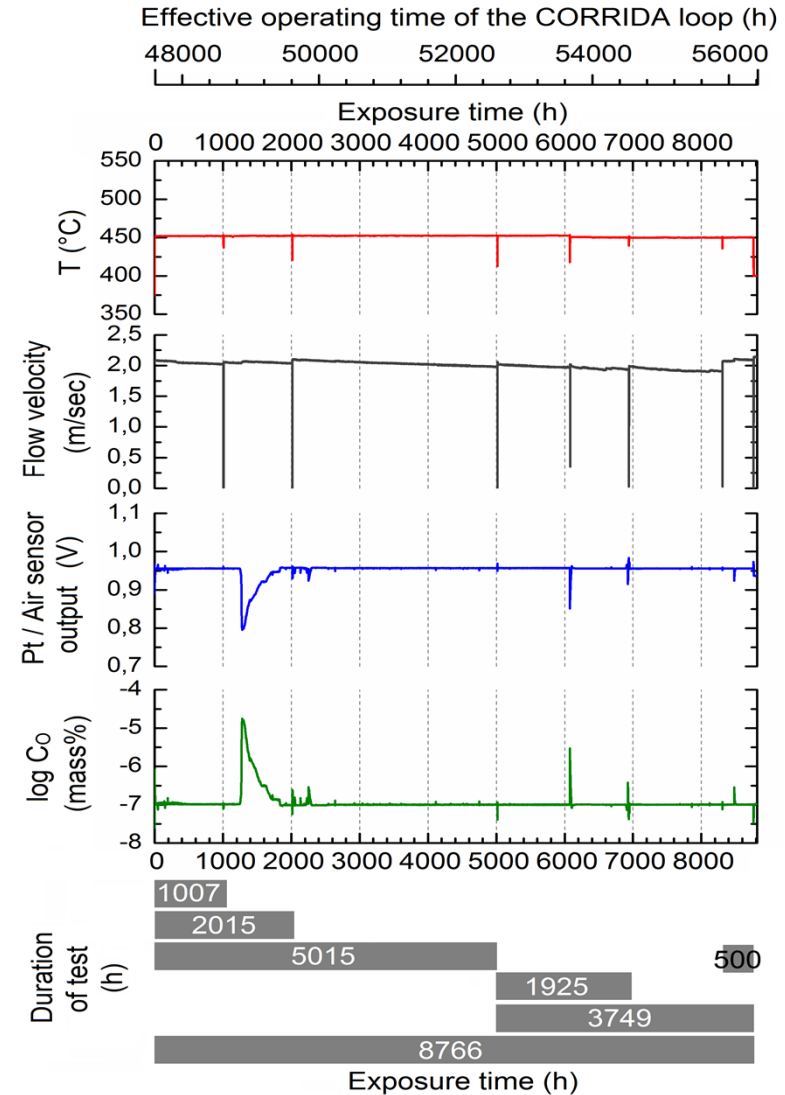
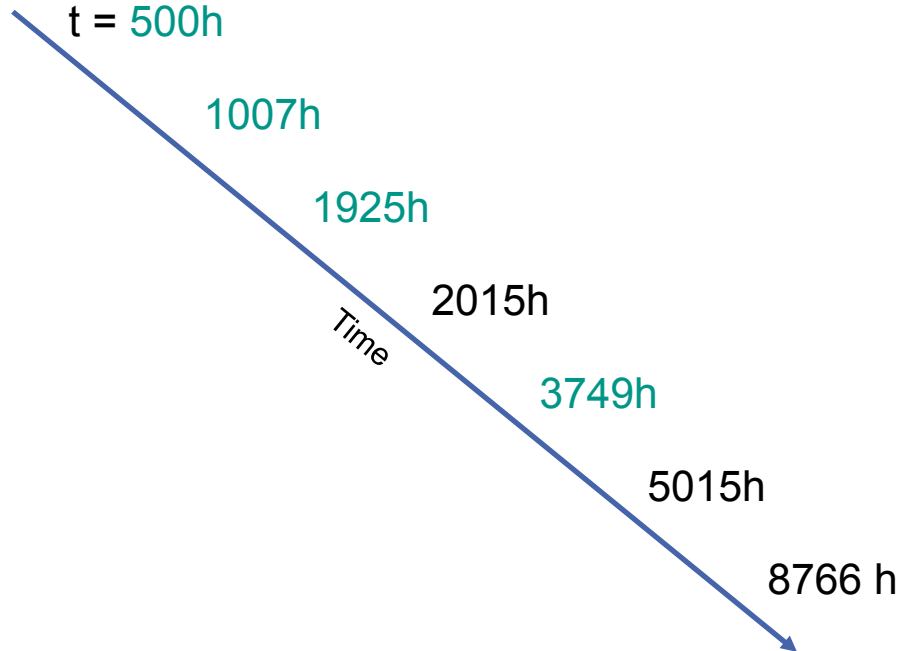
Transverse circular cross-section

$$\Delta X_{ST} = \frac{1}{2} \left(D_0 - \left(\frac{\sum D_i}{i} \right) \right)$$

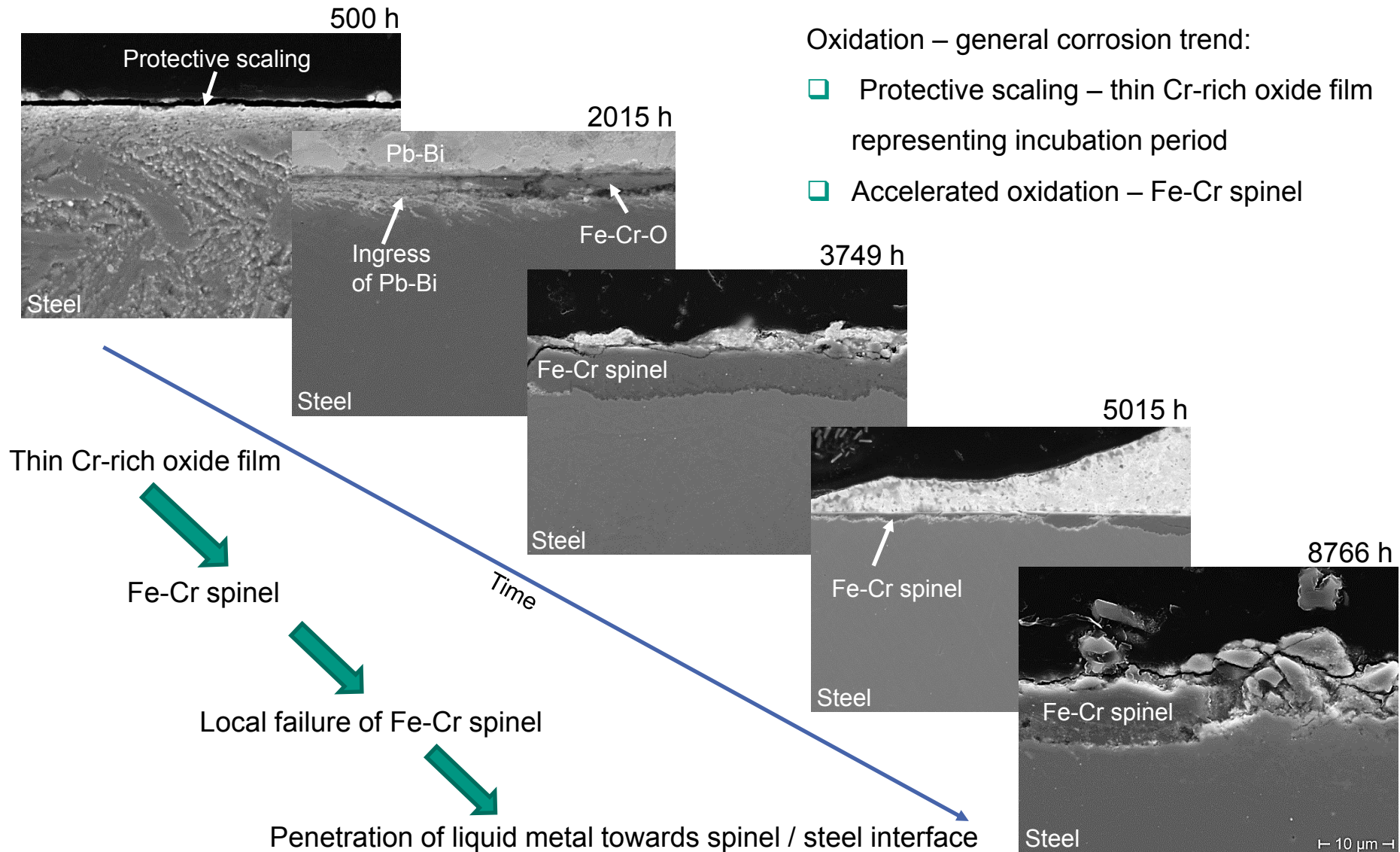
Corrosion test at 450°C in flowing Pb-Bi (2 m/s) with 10^{-7} mass% dissolved oxygen



- $T = 450(+5)^{\circ}\text{C}$
- $T_{\text{min}} \approx 350^{\circ}\text{C}$
- $c_{\text{O}} = 10^{-7}$ mass%, excursion to 10^{-5} mass%
- $v = 2(+/-0.2)$ m/s
- $t = 500\text{h}$



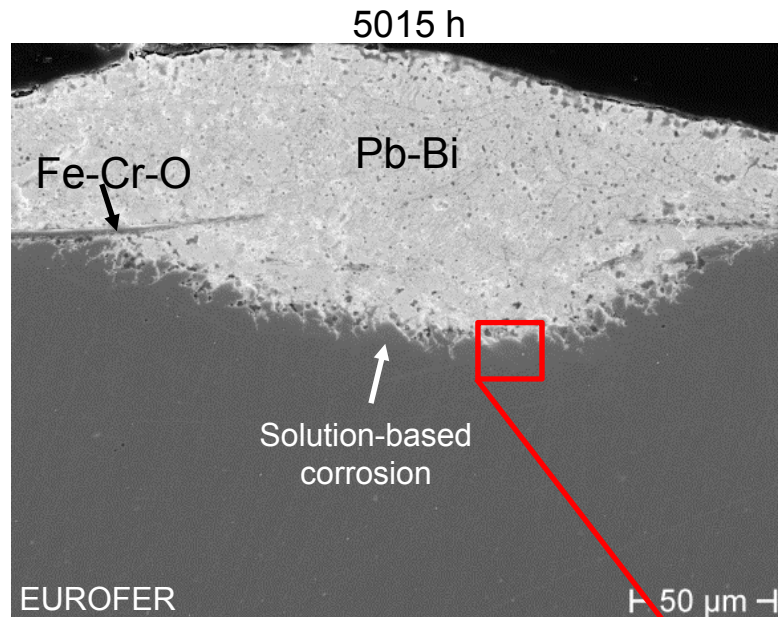
Corrosion appearances on surface of 9%Cr steels after test at 450°C in flowing Pb-Bi and $10^{-7}\%$ dissolved oxygen



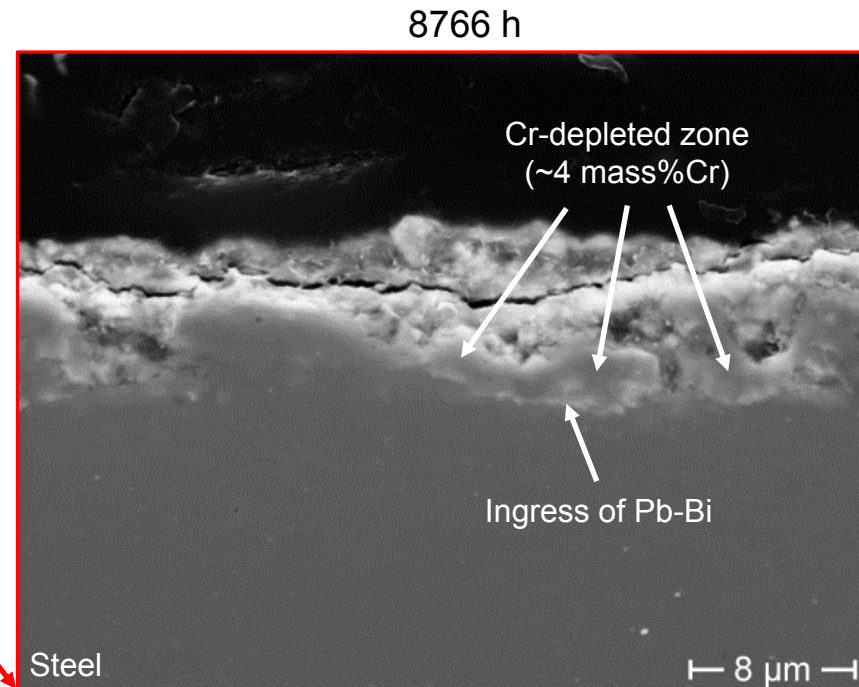
Oxidation – general corrosion trend:

- Protective scaling – thin Cr-rich oxide film representing incubation period
- Accelerated oxidation – Fe-Cr spinel

Solution-based attack – local corrosion trend

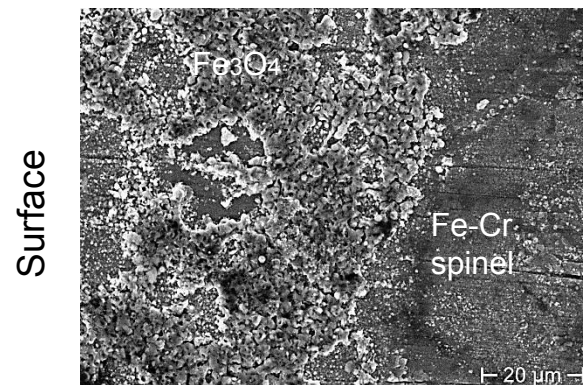
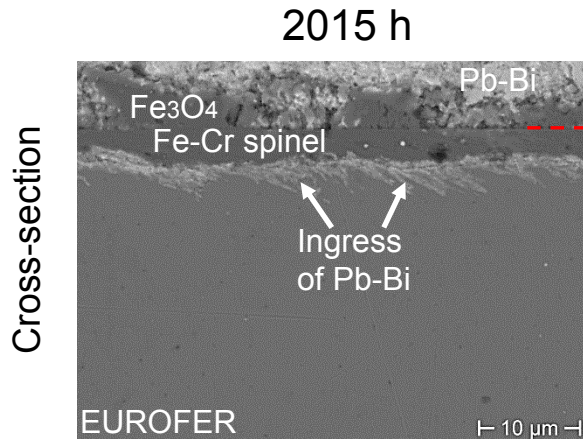


Pit-type corrosion attack



Detailed examination of near surface corrosion zone

Effect of increasing oxygen concentration



Magnetite crystals non-uniformly populate steel surface (Fe-Cr spinel surface)

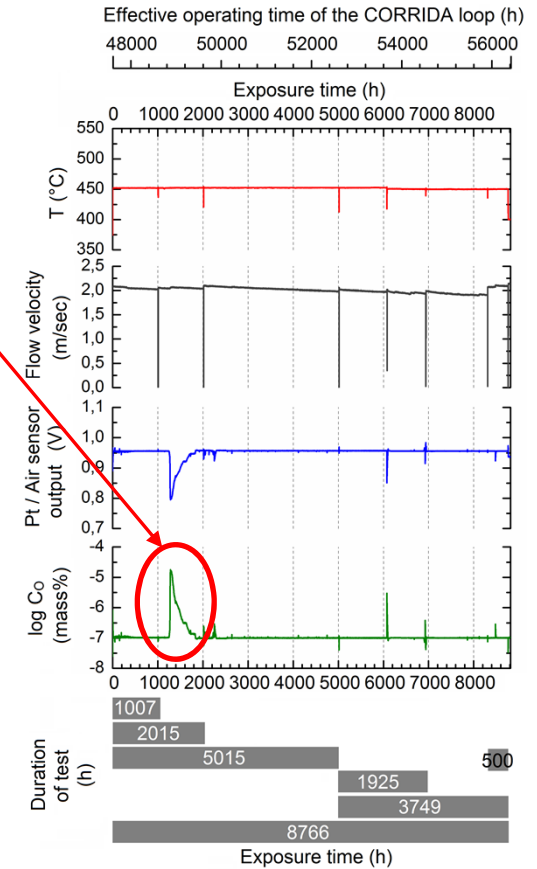
X=0 – initial interface steel / Pb-Bi

□ Magnetite formation

- Magnetite forms as a result of temporary increase in oxygen concentration from 10^{-7} to $\sim 10^{-5}$ mass%
- Magnetite is not observed after return to target 10^{-7} mass% O

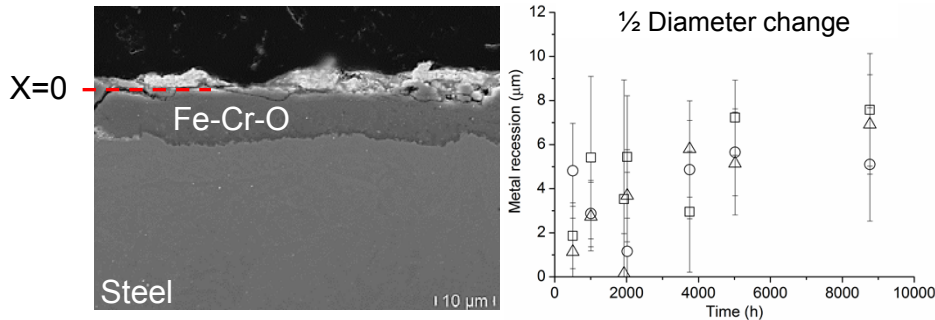


Magnetite dissolves!

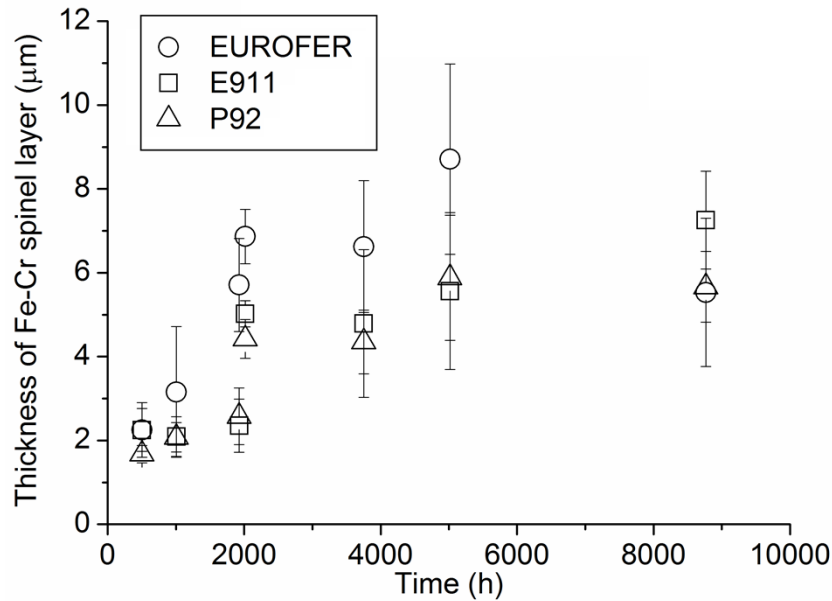
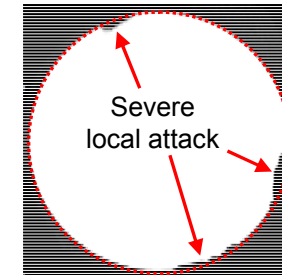


Quantification of corrosion loss on 9%Cr F/M steels after exposure to flowing Pb-Bi at 450°C, 2 m/s, 10⁻⁷ mass% O

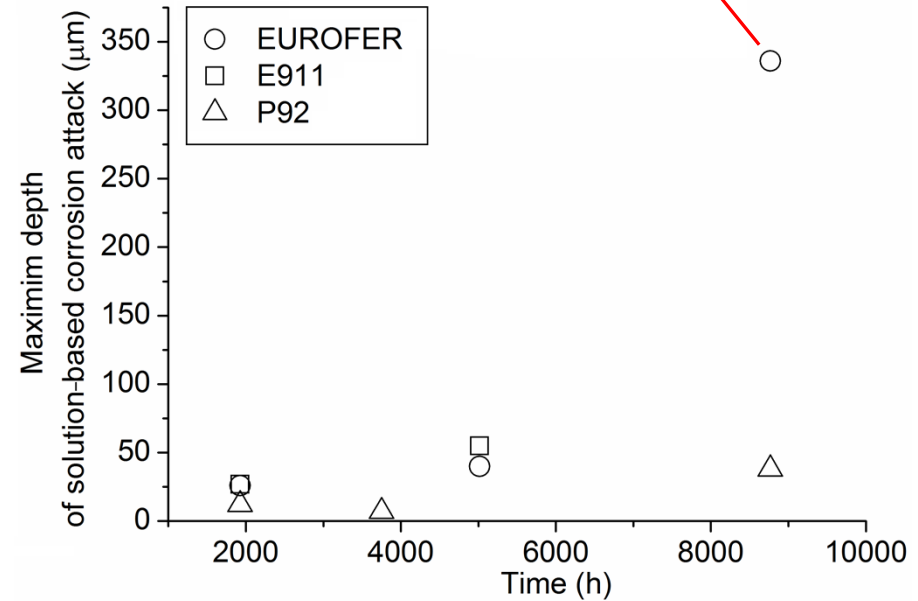
Accelerated oxidation General corrosion trend



Solution-based leaching of steel constituents (Fe, Cr) Local corrosion trend



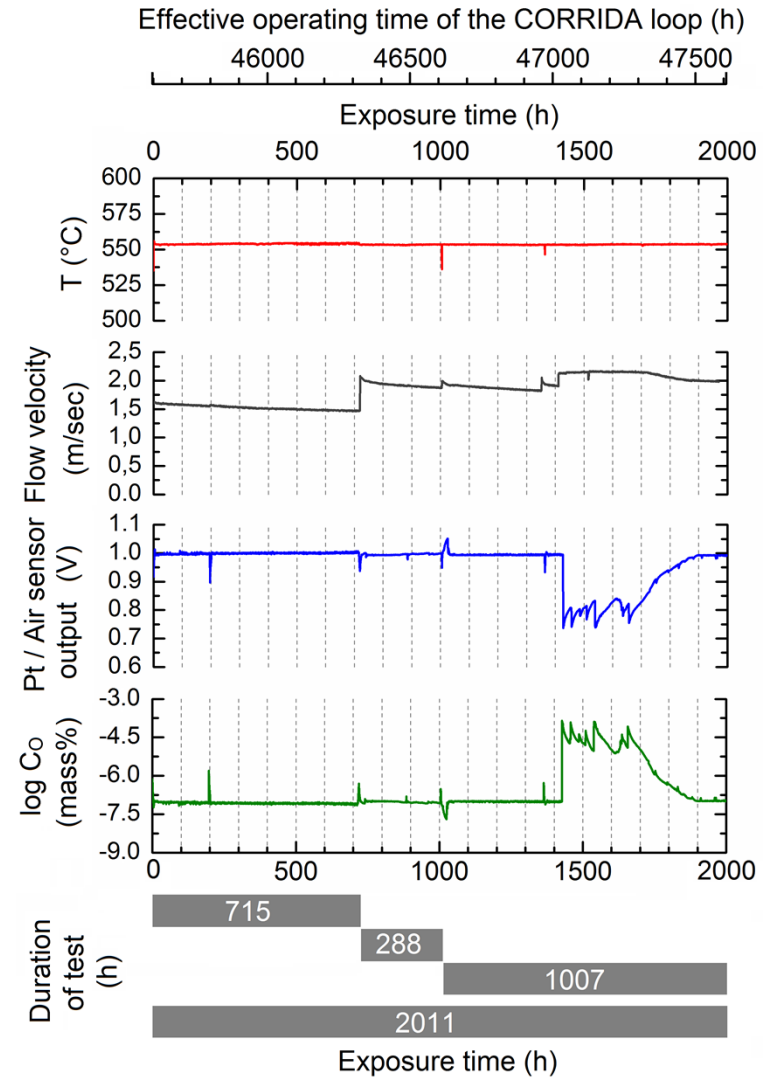
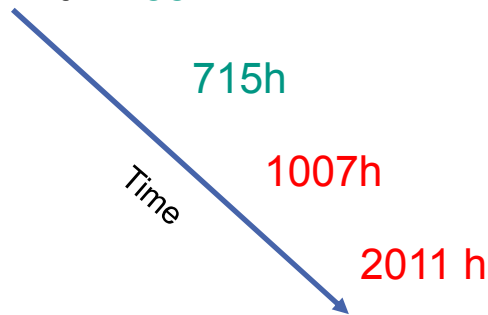
Incubation time 500-2000h



Corrosion test at 550°C in flowing Pb-Bi (2 m/s) with and 10⁻⁷ mass% dissolved oxygen



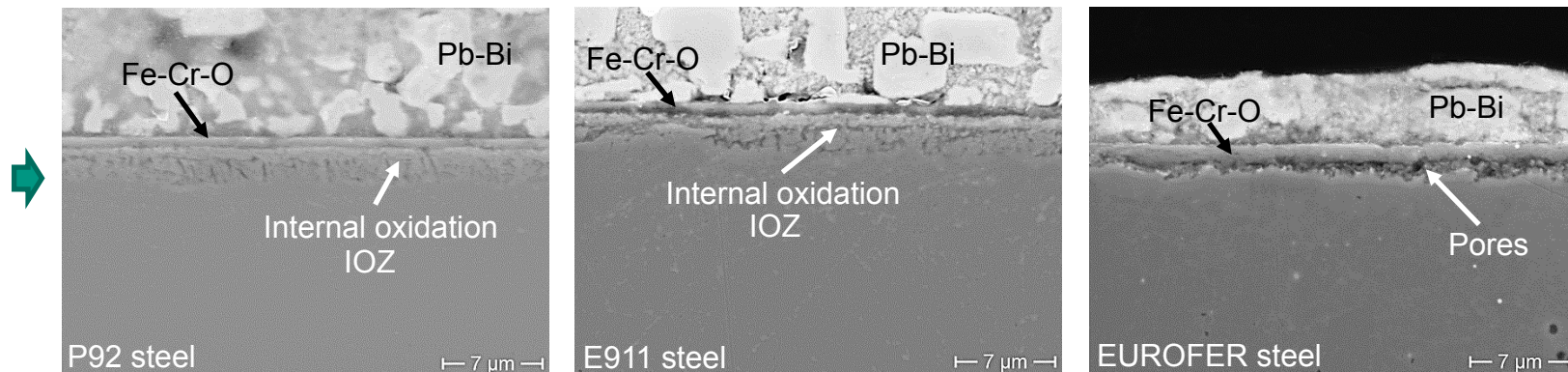
- T = 550(+5)°C
- T_{min} ≈ 385°C
- c_O = 10⁻⁷ mass%, excursion to 10⁻⁴–10⁻⁵ mass%
- v = 2(+/-0.2) m/s, initially 1.5–1.6 m/s
- t = 288h



Phenomena observed on 9%Cr steels after test at 550°C in flowing Pb-Bi and 10⁻⁷% dissolved oxygen

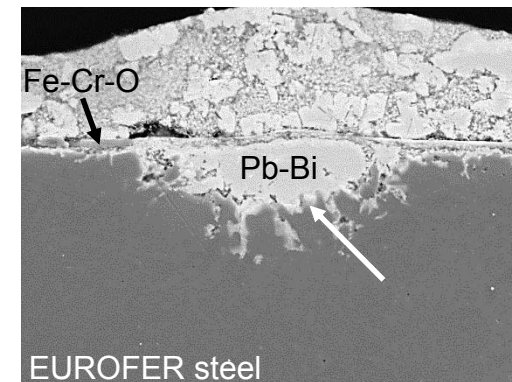
288 h

General corrosion trend



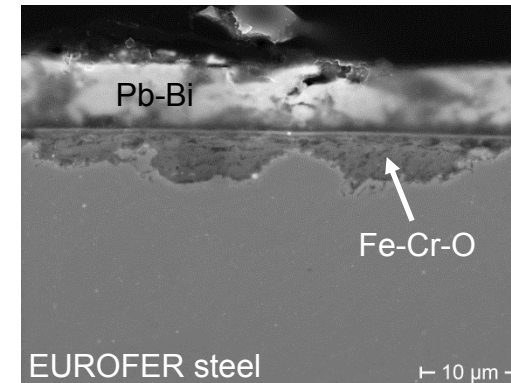
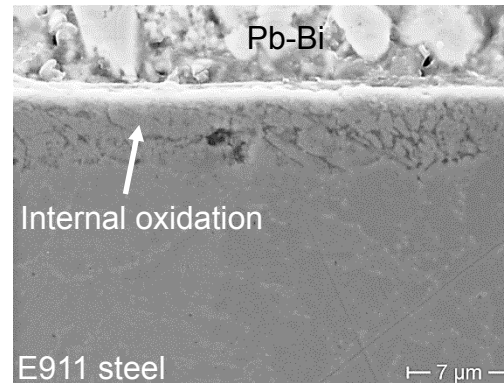
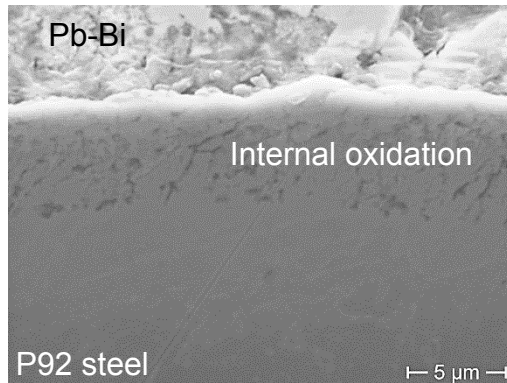
- Oxidation (Fe-Cr spinel + IOZ) – general corrosion trend
 - IOZ – Cr-based oxides in non-oxidized Fe-matrix
- Pit-type solution-based attack – local corrosion trend

Local corrosion trend

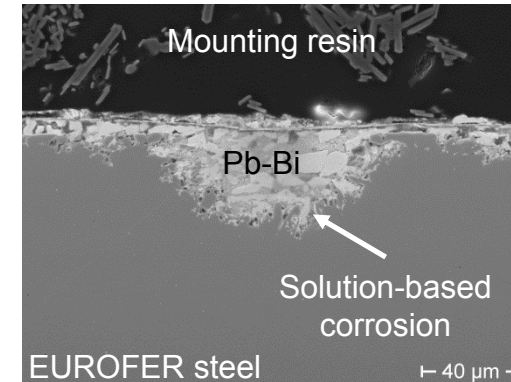
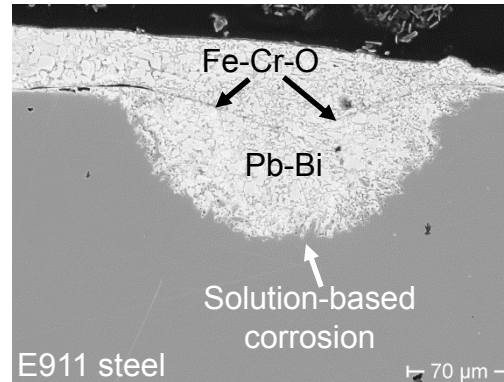
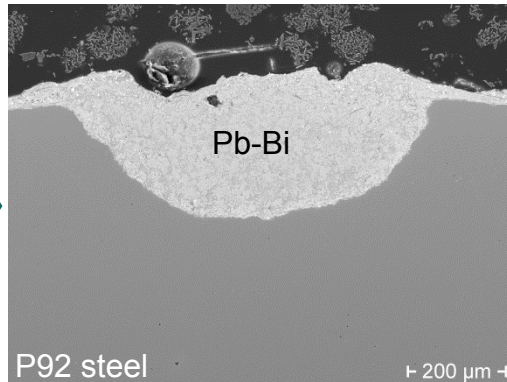


715 h

General corrosion trend

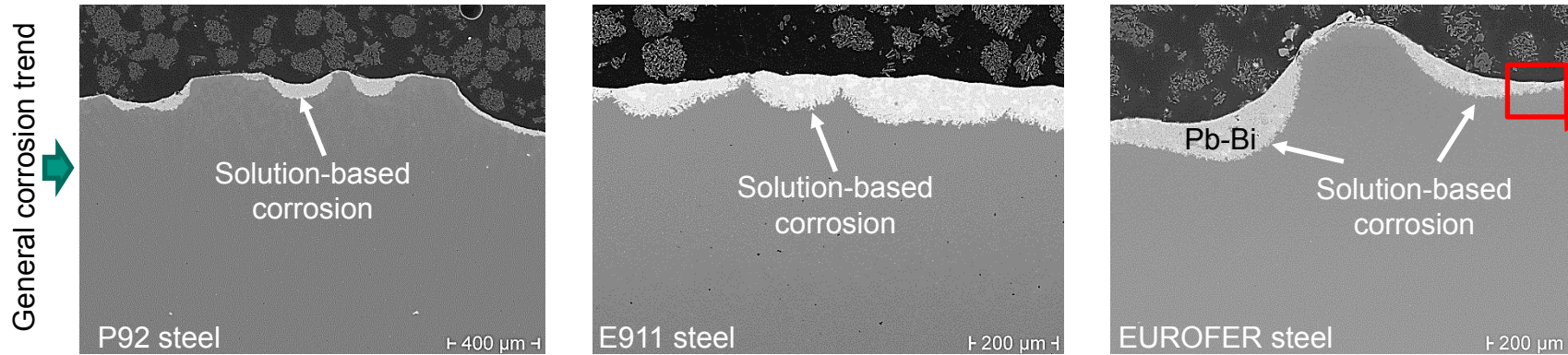


Local corrosion trend



- ❑ Oxidation (Fe-Cr spinel + IOZ) – general corrosion trend
- ❑ Pit-type solution-based attack – local corrosion trend

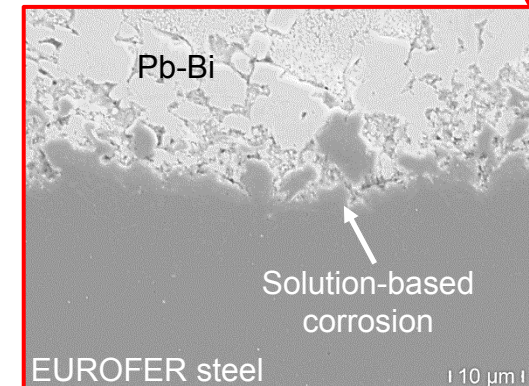
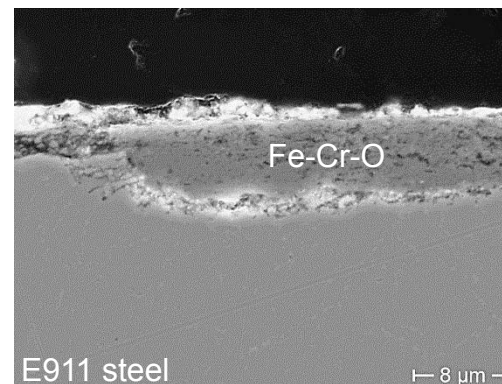
1007 h



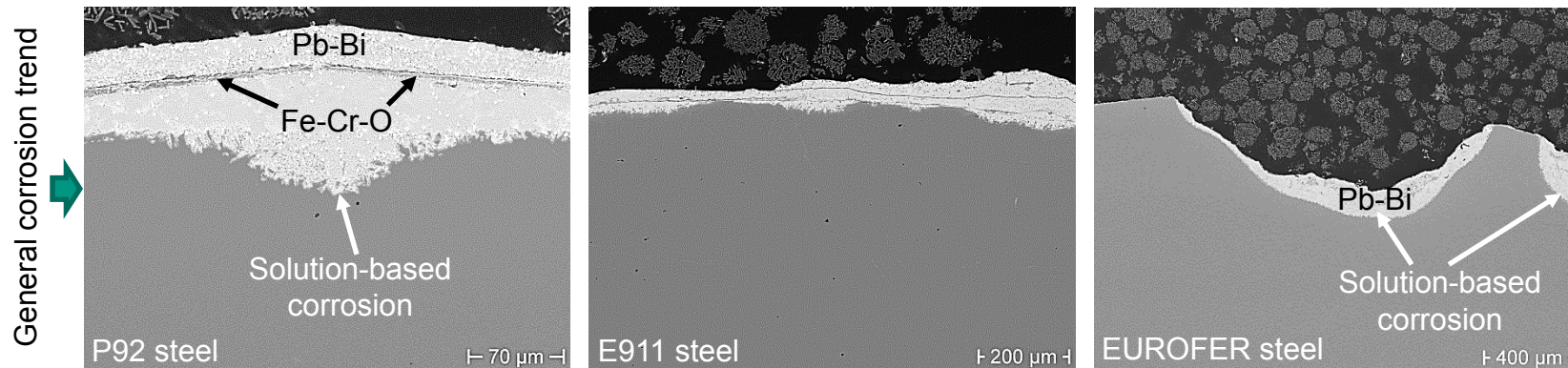
Severe material ablation due to solution-based attack –
general corrosion trend

Oxidation (Fe-Cr spinel) –
local corrosion trend

Local corrosion trend



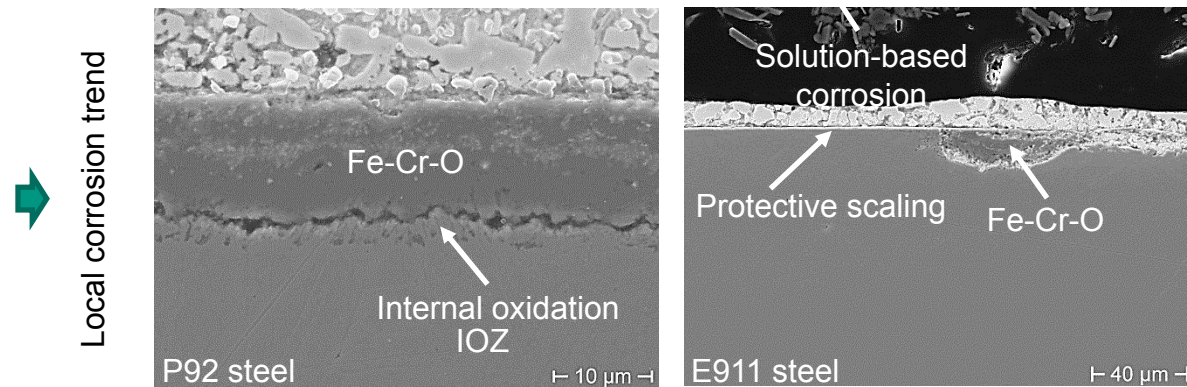
2011 h



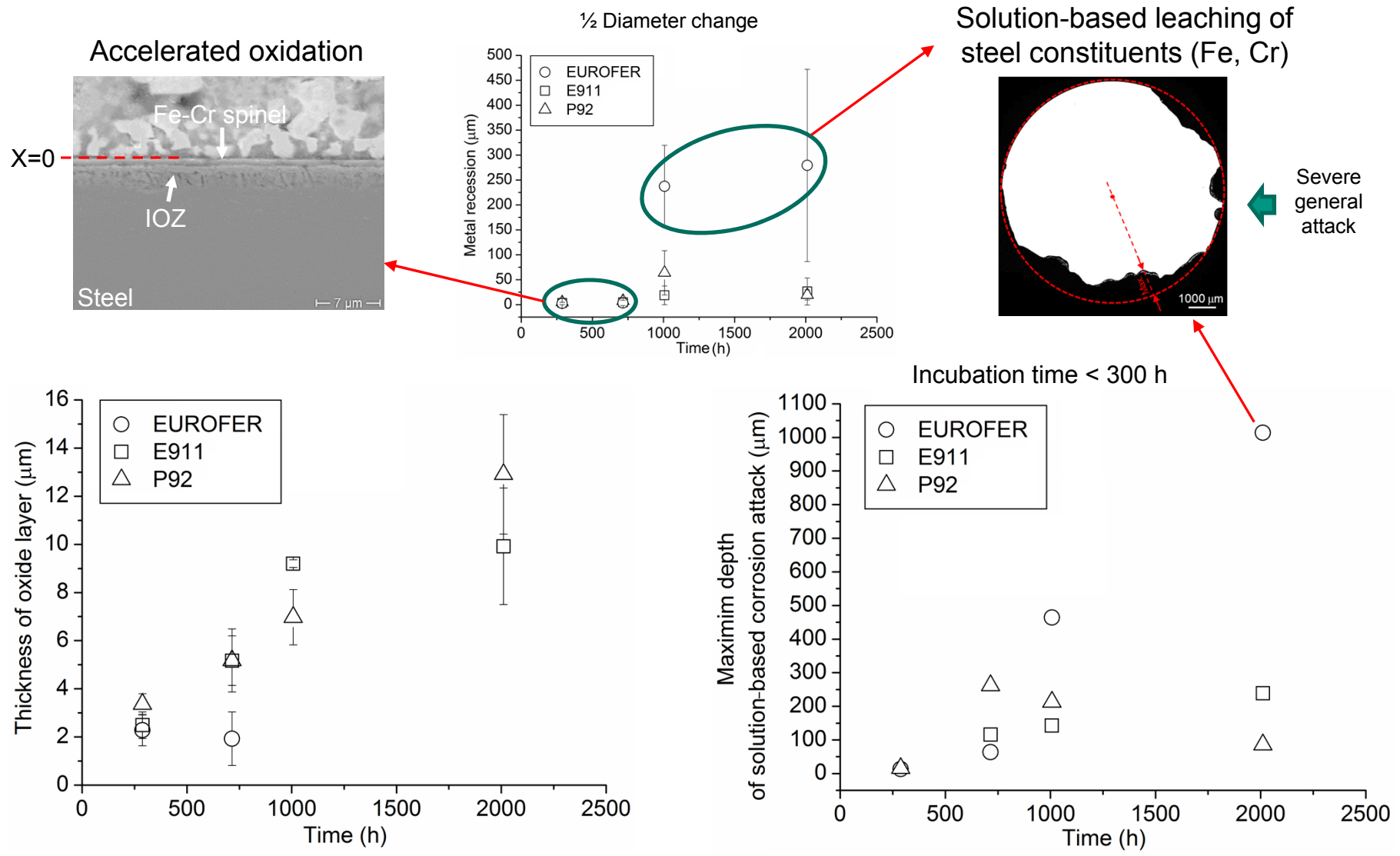
❑ Severe solution-based attack – general corrosion trend

❑ Oxidation – local corrosion trend:

- Fe-Cr spinel + IOZ
- Cr-rich film - protective scaling



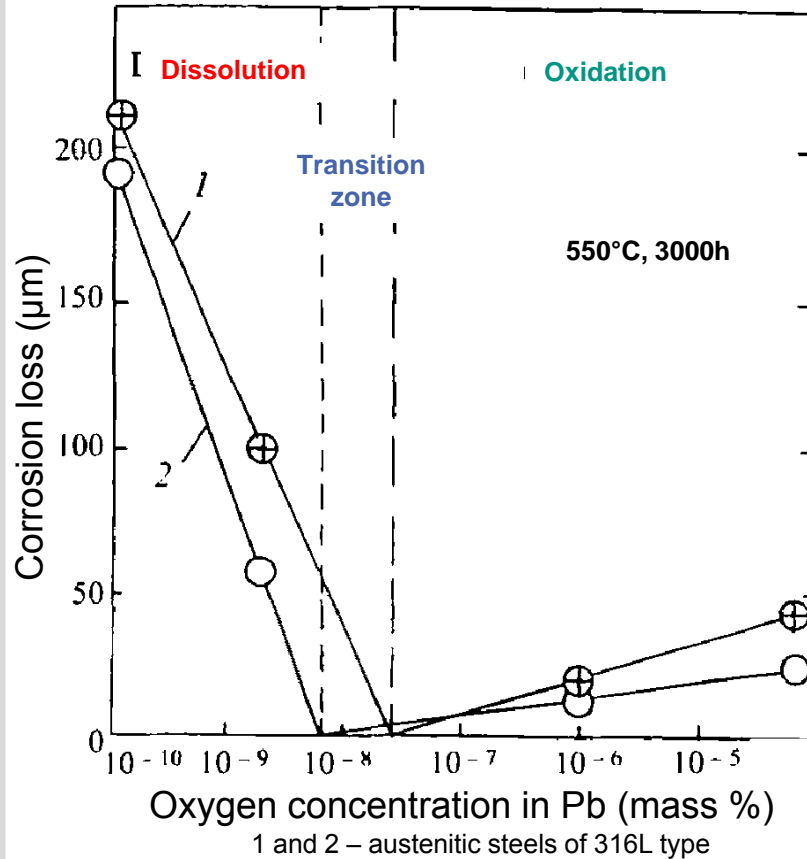
Quantification of corrosion loss on steels after exposure to flowing Pb-Bi at 550°C, 2 m/s, 10⁻⁷ mass% O



Comparison of earlier findings with last ones !

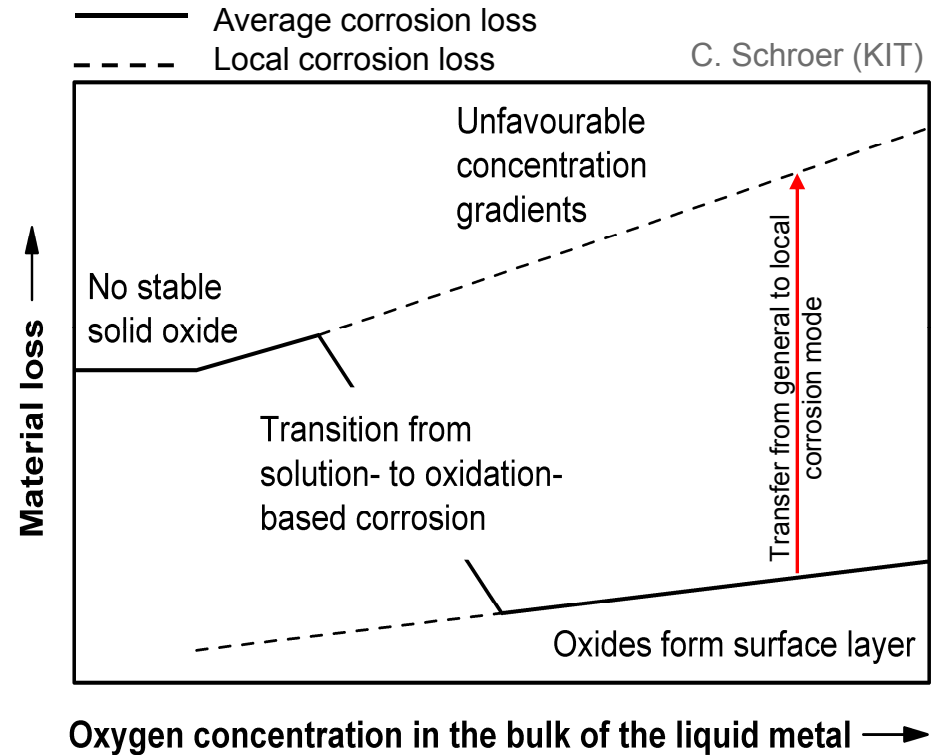
Earlier findings !

I.V. Gorynin et al. Met. Sci. Heat Treat. 41 (9) (1999) 384–388.



❑ In general correct, but in particular too idealistic !

Today's findings !



- ❑ In the oxide-protection regime the local failure of scale results in more severe solution-based corrosion attack instead of expected re-oxidation of steel surface!
- ❑ Task is to postpone the appearance of local attack!

Comparison of corrosion behaviour of 9%Cr steels in Pb-Bi depending on the oxygen concentration in the melt

10⁻⁷ mass%O

Increase in oxygen concentration

10⁻⁶ mass%O

450°C

- Oxidation
 - Protective scaling – thin **Cr-based film**
 - Accelerated oxidation – thicker **Fe-Cr spinel**
- Local solution-based attack as a result of spinel failure
 - Incubation time 500-2000 h

550°C

- Oxidation
 - Protective scaling – thin Cr-based film
 - Accelerated oxidation – thicker Fe-Cr spinel
- General solution-based attack as a result of spinel failure and Cr-based oxide film
 - Incubation time < 300 h

450°C

- Oxidation
 - Accelerated oxidation – **Bi-layer scale**
 $\text{Fe}_3\text{O}_4 / \text{Fe}(\text{Fe},\text{Cr})_2\text{O}_4$
- Local solution-based attack as a result of bi-layer scale failure
 - Incubation time > 8000 h

550°C

- Oxidation
 - Protective scaling – thin Cr-based film
 - Accelerated oxidation – thicker Fe-Cr spinel
- Local solution-based attack as a result of spinel failure and Cr-based oxide film
 - Incubation time 5000-15000 h

SUMMARY

- ❑ At 450 and 550°C, in flowing oxygen-containing LBE (2 m/s and 10^{-7} mass% O), **F/M steels with 9% nominal content of Cr (P92, E911, EUROFER) show following corrosion modes:**
 - ❑ Protective scaling – short term or local phenomenon (thin Cr-based oxide)
 - ❑ Accelerated oxidation – the general corrosion mode (thicker $\text{Fe}(\text{Fe}_x\text{Cr}_{1-x})_2\text{O}_4$ scale) resulted in metal recession at 450°C of $\sim 6 \mu\text{m}$ after one year
 - ❑ Solution-based corrosion
 - local at 450°C and ranged between 7-336 μm
 - general at 550°C and reached 13-1000 μm
- ❑ **EUROFER showed the largest corrosion loss** among the steels tested via accelerated oxidation and solution-based corrosion that might be caused by
 - less Si content, which normally improves protective properties of scales formed on P92 and E911 and prolongs incubation period
 - fine-grained structure that in combination with less protective oxide film might favor development of local solution-based corrosion attack after failure of scale
- ❑ **Comparison between 10^{-7} mass% O and 10^{-6} mass% O:**
 - ❑ Shorter incubation time for 10^{-7} mass% O
 - ❑ Slower accelerated oxidation for 10^{-7} mass% O in terms of metal recession but only at 550°C
 - ❑ Decrease in scale thickness for 10^{-7} mass% O at 450°C due to missing magnetite, but an equivalent amount of Fe is dissolved by liquid metal
 - ❑ The material loss caused by oxidation is generally lower at the lower oxygen concentration, but the risk of initiation of local solution-based corrosion attack increases;
 - ❑ 10^{-6} mass% is closer to the optimum oxygen content in LBE than 10^{-7} mass% at least for 9%Cr steels

Thank you for attention !!!

Acknowledgements

The construction and operation of the CORRIDA loop was financially supported by the Nuclear Safety Programme of KIT.

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