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Composition, mixing state and water affinity of meteoric smoke analogue nanoparticles produced in a non-thermal microwave plasma source

MSP analogues produced in a microwave plasma resonator

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Abstract: The article reports on the composition, mixing state and water affinity of iron silicate particles which were produced in a non-thermal low-pressure microwave plasma reactor. The particles are intended to be used as meteoric smoke particle analogues. We used the organometallic precursors ferrocene ($\text{Fe}(\text{C}_5\text{H}_5)_2$) and tetraethyl orthosilicate (TEOS, $\text{Si}(\text{OC}_2\text{H}_5)_4$) in various mixing ratios to produce nanoparticles with radii between 1 nm and 4 nm. The nanoparticles were deposited on sample grids and their stoichiometric composition was analyzed in an electron microscope using energy dispersive X-ray spectroscopy (EDS). We show that the pure silicon oxide and iron oxide particles consist of SiO_2 and Fe_2O_3 , respectively. For $\text{Fe}:(\text{Fe}+\text{Si})$ ratios between 0.2 and 0.8 our reactor produces (in contrast to other particle sources) mixed iron silicates with a stoichiometric composition according to $\text{Fe}_x\text{Si}_{(1-x)}\text{O}_3$ ($0 \leq x \leq 1$). This indicates that the particles are formed by polymerization of FeO_3 and SiO_3 and that rearrangement to the more stable silicates ferrosilite (FeSiO_3) and fayalite (Fe_2SiO_4) does not occur at these conditions. To investigate the internal mixing state of the particles, the H_2O surface desorption energy of the particles was measured. We found that the nanoparticles are internally mixed and that differential coating resulting in a core-shell structure does not occur.

Keywords: Microwave Plasma, Nanoparticles, EDS, Meteoric Smoke Particles, Iron Silicates

1 Introduction

Nanoparticles are of great importance for many chemical and physical applications and research fields. In planetary atmospheres, they serve as important condensation nuclei triggering the formation of clouds: Every day, about 40 tons of meteoric material enters the atmosphere of Earth (1, 2). This material ablates in the upper atmosphere with peak ablation heights of the major elemental components Fe, Si and Mg at a height between 80 km and 90 km (3). Oxides, carbonates and hydroxides are formed via oxidation by O_3 , O_2 , CO_2 and H_2O and then serve as building blocks for so called Meteoric Smoke Particles (MSP) (4). In the summer mesopause in Earth's atmosphere, the majority of MSPs is smaller than 2 nm in radius (5-10) and they are believed to serve as nuclei for the formation of noctilucent clouds (NLCs) (10-12). On Mars, the peak meteoric ablation height is between 60 km and 100 km (13-15), which is consistent with heights of CO_2 cloud observations (16-21) adverting the importance of MSPs as potential condensation nuclei. Indeed, Listowski and co-workers were only able to model realistic Martian mesospheric cloud patterns when assuming the presence of MSPs (22).

Due the small size of MSPs it is a challenging task to measure the MSP concentration and composition in the mesosphere which is subject of ongoing research. Current investigations point to an iron rich chemical formula (11, 23). Analogue materials for various kinds of cosmic dust have been produced by vapor condensation, sol-gel preparation or photo-oxidation (24-26). These methods do not mimic realistic conditions at which particles are formed in planetary atmospheres or the interstellar medium, but are used to produce particles with chemical compositions they might consist of. Here, we present a method to produce MSP analogues of realistic size using a non-thermal low-pressure mi-

49 microwave plasma reactor. This method does not mimic realistic conditions at which MSPs are produced
50 in the atmosphere as well, but allows creating particles which consist of the major elements expected
51 to be in MSPs. These particles thus serve as MSP analogues and are used to study cloud formation
52 processes at realistic mesospheric conditions of Mars and Earth with the recently introduced MICE-
53 TRAPS setup (27, 28).

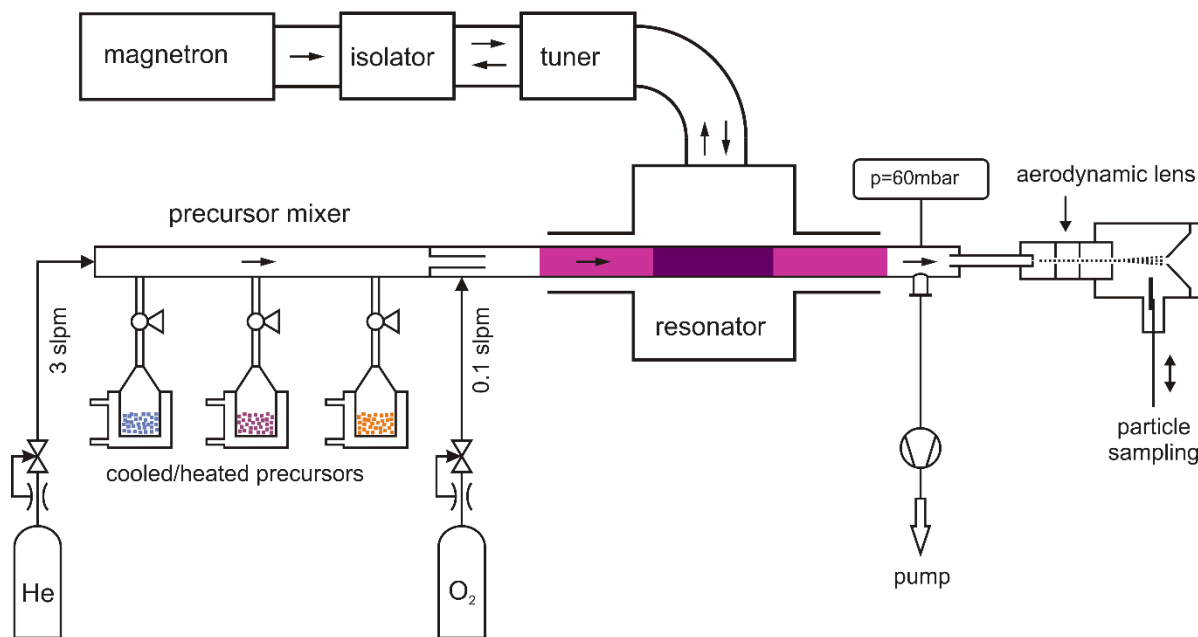
54 Understanding mesospheric cloud formation requires nucleation experiments conducted with the com-
55 plete composition range of MSPs, since the nucleation ability of the particles is likely to be material
56 dependent. In addition, MSPs might heat up in the low pressure environment of the mesosphere by
57 absorption of sunlight (29). The absorption ability of the particles is material dependent and is studied
58 in ongoing research.

59 In this work, we characterize mixed iron silicate nanoparticles produced in a non-thermal low pressure
60 microwave plasma reactor. The paper is organized as follows: In section 2, we introduce the used
61 experimental techniques which are Energy Dispersive Spectroscopy (EDS) and H₂O adsorption
62 measurements with MICE-TRAPS. We show in section 3 that pure iron oxide and silicon oxide parti-
63 cles are composed of Fe₂O₃ and SiO₂, respectively. Mixed iron-silicate particles, however, are com-
64 posed of a homogeneous mix of FeO₃ and SiO₃ molecules with their ratio being controlled by the iron
65 and silicon concentration in the plasma reactor. We conclude with a summary of the results and an
66 outlook in section 4.

67 2 Experimental

68 2.1 Nanoparticle generation

69 Metal oxide nanoparticles were produced in a non-thermal low-pressure (60 mbar) microwave plasma
70 reactor. The operating principle of such nanoparticle sources has been described previously in detail
71 (e.g. 30). The shape of metal oxide nanoparticles produced in similar experimental setups has been
72 shown to be compact and spherical with a marginal degree of agglomeration (31-34). The design of
73 the nanoparticle source employed in this experiment is depicted schematically in Figure 1. Compared
74 to other designs of this type the setup differs in the way of precursor vaporization and mixing with the
75 background gas flow which we will describe in more detail. Organometallic precursors are stored in
76 separate and independently temperature-controlled reservoirs. The amount of precursor vapor added
77 to the background flow in the mixer is determined solely by the temperature of the precursor reservoir.
78 This setup has the advantages that it is very simple and that liquid and solid precursors can be vapor-
79 ized using identical reservoirs without the need for additional flow controlled gas lines. In this study we
80 use tetraethyl orthosilicate (TEOS, Si(OC₂H₅)₄, Sigma Aldrich) and ferrocene (Fe(C₅H₅)₂, Sigma Al-
81 drich) as precursors for silicon and iron, respectively. The precursor vapors are mixed with a contin-
82 uous 3 slpm flow of Helium to which a 100 sccm flow of O₂ is added downstream. The addition of O₂ is
83 processed behind an orifice in flow direction to avoid premature oxidization and deposition of the low
84 volatile metal oxides, which has been observed to occur for some precursors. Pressure and flow con-
85 ditions ensure diffusive mixing of vapors and gases within a few centimeters. After mixing, the flow
86 enters a quartz glass tube of 2.2 cm inner diameter and 40 cm length. The Reynolds number of the
87 flow in this tube of Re~24 indicates a laminar regime. The tube is placed in the center of the micro-
88 wave resonator at which a microwave induced plasma of about 15 cm length is ignited (2.45 GHz, 350
89 W). The flow rate results in a retention time of about 60 ms in the plasma during which the precursor
90 molecules completely decompose causing the release of Fe and Si atoms. The presence of oxygen (3
91 mol%) in the gas flow assures oxidization of these atoms which then form particles. In addition, the
92 organic parts of the precursors are efficiently oxidized to volatile carbon dioxide and water which are
93 not included in the particle material. A precursor mixing ratio of 30 ppm (0.003 mol%) or less was
94 maintained in the reactor cell such that the ratio of O₂ per precursor molecule was always at or higher
95 than 1000. For comparison, to oxidize the organic parts of TEOS and ferrocene molecules requires 11
96 and 12.5 molecules of O₂, respectively. The bulk of the particle-laden flow behind the discharge region
97 is pumped off and discarded and less than 20% of the sample flow is extracted and fed to the nano-
98 particle characterization unit.



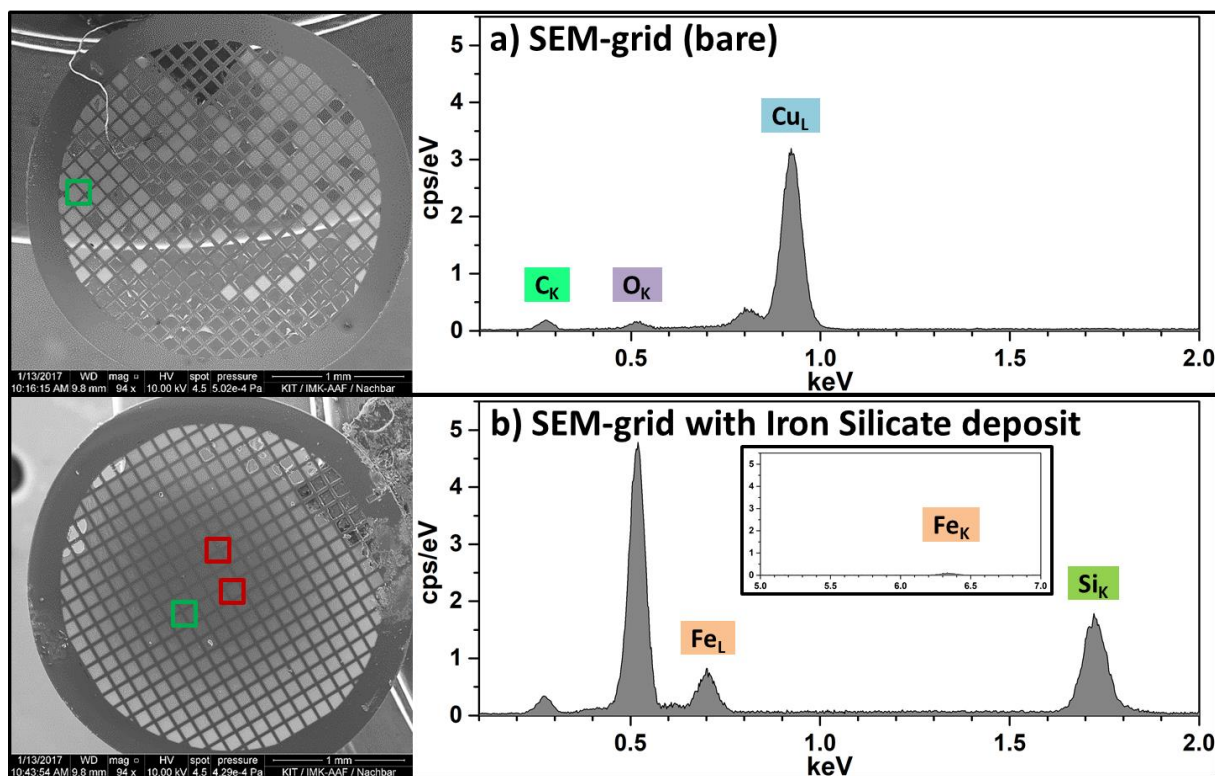
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100 **Fig. 1:** Schematic representation of the microwave nanoparticle source.

101 To study nanoparticle properties using the MICE-TRAPS apparatus the nanoparticles were separated
 102 from the residual gas of the nanoparticle source using an aerodynamic lens (ADL) and a differential
 103 pumping stage which has been described in detail previously (28). After the ADL, the particle beam
 104 enters the main vacuum chamber through a skimmer. At this point, a target plate was inserted to de-
 105 posit the particles on standard transmission electron microscope (TEM) sample grids (\varnothing 3mm, copper
 106 mesh, coated with carbon film). The particle mass distribution at the output of the nanoparticle source
 107 depends on pressure, flow conditions, precursor concentration in the plasma reactor and retention
 108 time. It was found, that for the conditions employed in this experiment using only a single precursor,
 109 particles with diameter ranging from 2 to 8 nm were produced. The charge state of the particles de-
 110 pends on various parameters such as precursor composition and concentration, microwave power,
 111 retention time and other parameters of the microwave induced plasma. Although the exact particle
 112 charge distribution could not be measured, we assume that about 50% of the particles are neutral and
 113 50% are singly charged with a preference towards carrying a positive charge. The exact values, how-
 114 ever, may differ substantially depending on the above mentioned parameters.

115 2.2 Nanoparticle composition

116 After particle deposition, the TEM sample grids were analyzed in a scanning electron microscope
 117 (SEM, FEI Quanta 650 FEG) using an energy dispersive X-ray spectrometer (EDS, Bruker Quantax
 118 XFlash 5010). The sample grids loaded with nanoparticles were placed perpendicular to the electron
 119 beam. EDS spectra were obtained with a 10 keV beam scanning a square area size of 200x200 μ m on
 120 the sample surface. Exemplary SEM images and EDS spectra of a bare TEM grid (a) and a grid with
 121 iron-silicate particle deposit (b) are shown in Fig. 2 for photon energies between 0.1keV and 2keV.
 122 The insert in the lower right part shows the EDS spectra of the iron-silicate particle deposit for photon
 123 energies between 5 keV and 7 keV. The EDS spectrum of the bare grid shows the characteristic
 124 peaks associated with the copper bulk material of the grid (L-line, 0.93 keV). The spectrum also shows
 125 weak peaks of C (K-line, 0.277 keV) and O which are always observed by the EDS detector in this
 126 instrument independent of the sample. The EDS spectrum of an iron silicate deposit shows peaks
 127 which are attributed to the presence of O (K-line, 0.525 keV), Fe (L-line, 0.705 keV) and Si (K-line,
 128 1.74 keV). The carbon peak in the EDS spectra of the iron silicate deposit is only slightly increased
 129 compared to the bare grid. This confirms that no eminent amount of carbon is present in the particle
 130 material.
 131



132

133 **Fig. 2:** Left panels: SEM images of a bare grid (a, top) and of a grid covered with iron-silicate nanoparticles (b, bottom) (scale
 134 bar 1 mm). Right panels: EDS spectra taken at the electron beam scanning locations indicated by the green boxes in the SEM
 135 images. Horizontal axis is photon energy (0.1 to 2keV) and vertical axis is photon count rate (0 to 5.5 cps/eV bin). Positions of
 136 the main emission peaks of C, O, Fe, Cu, and Si are labeled. The insert shows the high energy part of the spectrum (5 to 7keV)
 137 using the same vertical scale and illustrates the intensity of the Fe_K-line at low iron content.

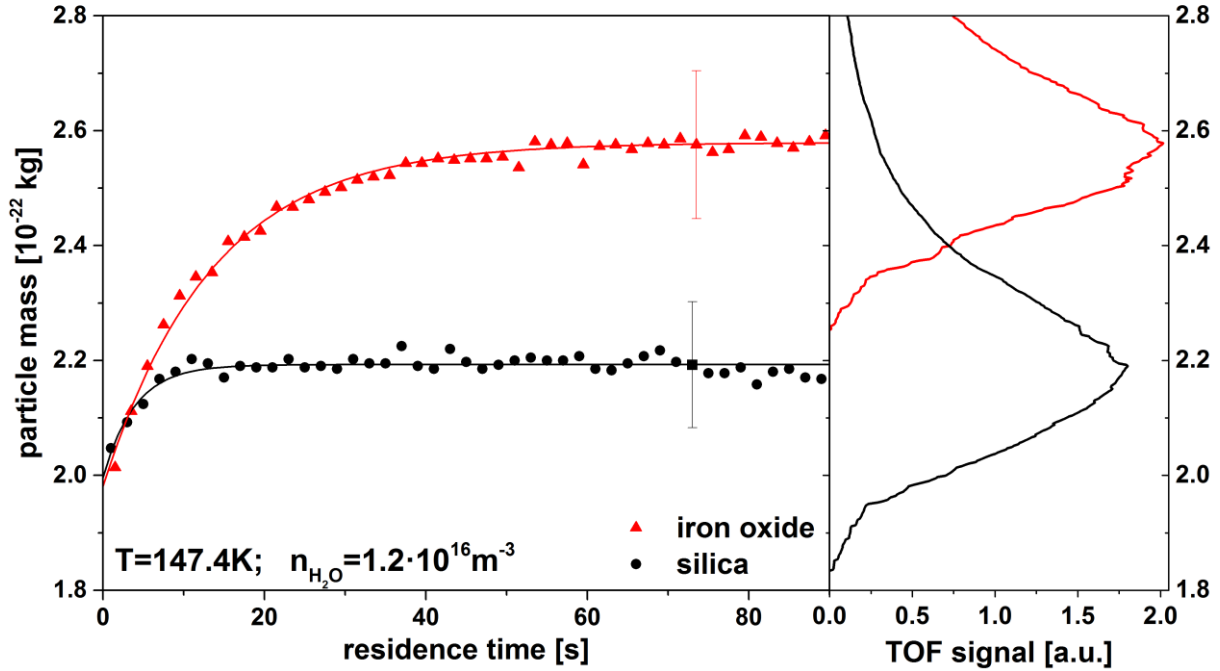
138 Quantitative analysis of the elemental composition was performed using the Esprit 1.9 software by
 139 Bruker. The analysis employed in the software is based on evaluating the peak to background ratios
 140 which are then input to a modified ZAF matrix correction which accounts for atomic number (Z), ab-
 141 sorption (A) and secondary fluorescence effects (F). A detailed description of the method can be found
 142 in (35-37). Data was only obtained for thick deposit layers, so that no copper signal from the TEM grid
 143 is present in the EDS spectra. At 10 keV excitation energy the Fe_K-line and Fe_L-line can be used for
 144 quantitative analysis. In general, analysis of the Fe_K-line usually provides more accurate results than
 145 the Fe_L-line as interference with the overlapping O_K-line and additional errors occurring due to a higher
 146 background and varying absorption effects in the low energy range can be avoided. However, for
 147 samples with a low iron content as presented in panel b) of Fig. 2 (Fe:(Fe+Si)=0.38) the intensity of
 148 the Fe_K-line (6.405keV) was so low (cf. insert) that analyzing the Fe_L-line resulted in a lower uncer-
 149 tainty determined by the analysis software. For analysis of the iron content we always used the line result-
 150 ing in the lower uncertainty. For each sample, EDS spectra were taken at three different scanning
 151 locations. It was found that for all samples the variance of analyzed elemental composition between
 152 each location was smaller than the uncertainty given by the analysis software for a single spectrum.
 153 Thus, we assume that the result is independent from the chosen scanning location.

154 2.3 Nanoparticle mixing state

155 To study the particle mixing state, the nanoparticle surface was investigated by measuring the desorp-
 156 tion energy of water molecules with the MICE-TRAPS apparatus. The device (38) and the method
 157 using CO₂ as the adsorbent (39) have been described in previous publications. In this work, H₂O has
 158 been used as the adsorbent. In brief, the nanoparticles are mass selected and trapped in MICE and
 159 are exposed to a flux of water molecules at a temperature of about 145K. Under such conditions the
 160 initially bare particle surface adsorbs water molecules until an equilibrium surface concentration is
 161 reached. The adsorption equilibrium is determined by the ratio of adsorption and desorption rate
 162 where the latter is described by the H₂O desorption energy, a surface property characteristic for the
 163 nanoparticle material. The desorption energy is expected to reflect the chemical composition of the
 164 nanoparticle surface and provides indications on the internal particle mixing state. The adsorption pro-

165 cess is monitored by measuring the particle mass with a time-of-flight mass spectrometer (TOF-MS) at
 166 regular time intervals. This is achieved by extracting small samples of the trapped particle cloud from
 167 MICE through the electrostatic exit lens of the ion trap and guiding the particles into the acceleration
 168 zone of the TOF-MS. The left panel of Fig. 3 shows exemplary measurements of the particle mass as
 169 a function of the trapping time in MICE for pure iron oxide and silica nanoparticles.

170



171

172 **Fig. 3:** Left panel: Nanoparticle mass during H₂O adsorption process as function of the residence time in MICE for iron oxide
 173 (red triangles) and silica (black dots) nanoparticles with initial mass $m_0 = 2 \cdot 10^{-22}$ kg. The solid lines are exponential decay fits to
 174 determine the adsorbed water mass for each measurement. Right panel: Smoothed time-of-flight mass spectra for two selected
 175 points marked with error bars in the left panel at $t=73$ s. The particle mass is shown as a function of the TOF signal. The position
 176 of the peak maximum is used as the mean particle mass plotted in the left panel.

177 Starting with an initial mass of $m_0=2 \cdot 10^{-22}$ kg ($=1.2 \cdot 10^5$ u) for both materials the particles adsorb water
 178 vapor and quickly reach an equilibrium state. Iron oxide exhibits a higher desorption energy than silica
 179 and adsorbs more water on the particle surface. The mass of adsorbed water m_{ads} is determined from
 180 the measurement by fitting a single exponential decay of the form $m(t) = m_0 + m_{ads}(1 - e^{-t/\tau})$ to the
 181 data. The adsorbed water mass relates to the desorption energy F_{des} by the following equation (40):

$$182 \quad m_{ads} = \frac{n_{H_2O} \cdot A_0}{f_{vib} \sqrt{2\pi}} \cdot \sqrt{m_{H_2O} kT} \cdot \exp\left(\frac{F_{des}}{kT}\right) \quad (1)$$

183 Here, n_{H_2O} is the water vapor density in MICE, m_{H_2O} is the mass of a water molecule, k is the Boltz-
 184 mann constant, T is the temperature, f_{vib} is the vibrational frequency (10^{13} Hz for H₂O), and $A_0 =$
 185 $(6\sqrt{\pi} m_0 / \rho)^{2/3}$ is the initial particle surface area which is calculated from the initial particle mass m_0
 186 and particle density ρ assuming spherical particles. For iron-silicates of various compositions no tabu-
 187 lated values for the bulk densities are available. We assume a density that is a linear combination of
 188 the densities of iron oxide ($\rho \sim 5.2$ g/cm³) and silica ($\rho \sim 2.3$ g/cm³) according to the elemental ratio of iron
 189 to silicon. Equation (1) can be rearranged to yield the desorption energy as function of particle size
 190 and adsorbed water mass.

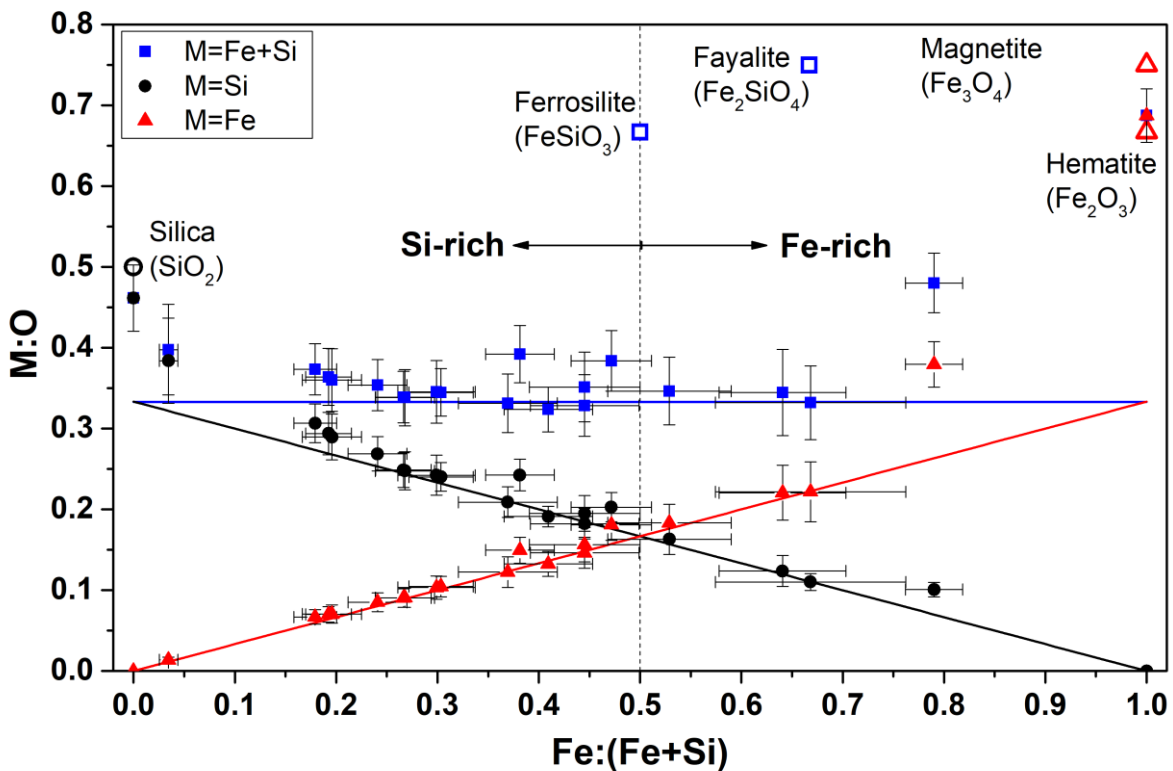
191 The data points in the left panel of Fig. 3 were obtained from the peak position of the time-of-flight
 192 mass spectra shown in the right panel exemplary for the two data points marked with error bars. The
 193 width of the peaks is specific for the TOF instrument and detector design. The width of the particle
 194 mass distribution trapped in MICE is typically much smaller. The peak position, however, is a very
 195 precise indicator of the mean particles mass. The difference in adsorbed water mass between both
 196 particle materials can easily be resolved.

197 To investigate the particle mixing state we analyze F_{des} as function of particle composition determined
 198 from EDS. For mixed particles containing both Fe and Si, we consider three possible scenarios: 1:
 199 Two independent classes of particles are produced in the particle source, one pure Fe-containing and
 200 one pure Si-containing class. 2: The particles created in the particle source exhibit a core-shell struc-
 201 ture, where the core contains only one metal oxide which is then covered with a shell of the other met-
 202 al oxide. This scenario could be realized if differential break-up of the precursors occurs in the flow
 203 reactor. 3: The particles are completely internally mixed, i.e. the chemical composition at the surface
 204 and in the bulk are identical. For case 1 the mass spectra for mixed particles should exhibit two distinct
 205 peaks, as both particle classes would adsorb water according to the measurements shown in Fig. 3.
 206 For case 2 the spectra should exhibit one distinct peak with the desorption energy being the same as
 207 either pure Fe- or pure Si-oxide particles. For case 3 one distinct TOF peak resulting in an intermedi-
 208 ate value for the desorption energy would be anticipated. While changing the Fe content of the parti-
 209 cles a continuous transition of the desorption energy from pure Si to pure Fe oxide would be expected.

210 3 Results and Discussion

211 3.1 EDS analysis

212 Mixed iron silicates were produced by varying the temperature of the precursors. For TEOS, tempera-
 213 tures were varied between 2°C and 30°C and for Ferrocene between 45°C and 90°C. The particles
 214 were deposited on sample grids whereupon the iron, silicon and oxygen content was analyzed using
 215 EDS. The iron to silicon ratio of the particle material was proportional to the vapor pressure ratio of
 216 both precursors at the applied temperatures. Fig. 4 shows the element to oxygen ratio for iron and
 217 silicon containing nanoparticles produced in the microwave nanoparticle source and analyzed using
 218 EDS as described above.



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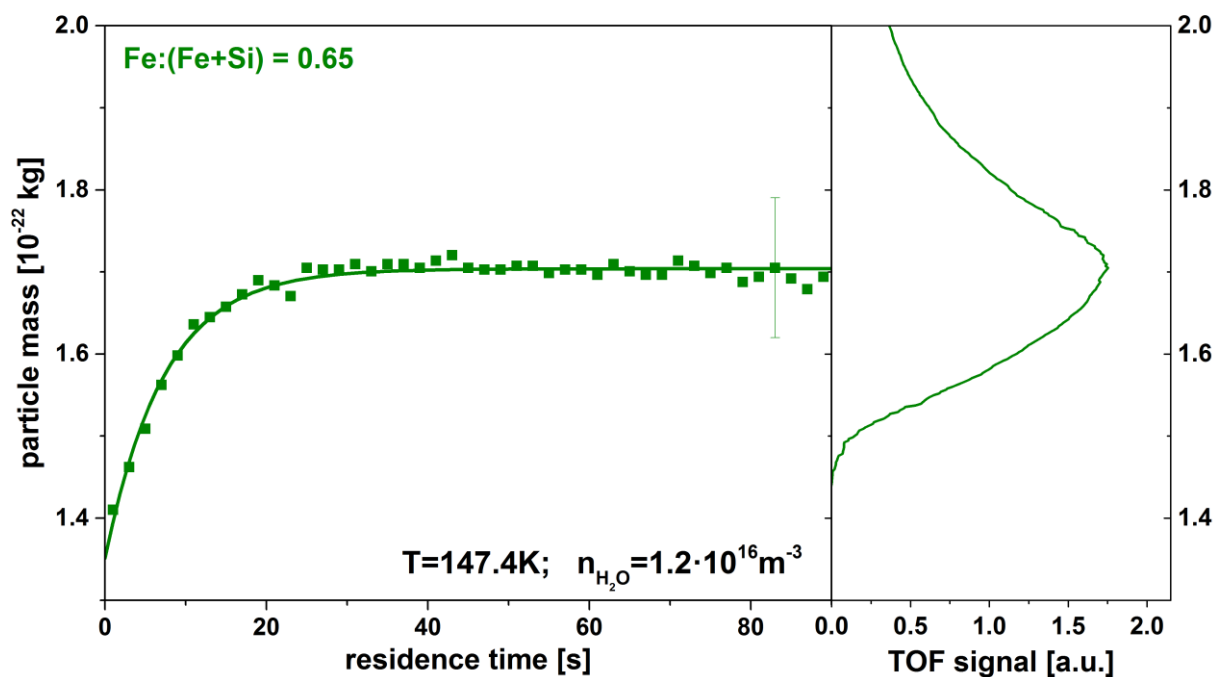
220 **Fig. 4:** Element to oxygen ratio M:O of iron-silicate nanoparticles plotted against the relative content of Fe. Open symbols indi-
 221 cate elemental ratios of natural iron and silicate minerals.

222 The relative abundances of Fe (red triangles), Si (black dots) and Fe+Si combined (blue squares) with
 223 respect to oxygen against the relative Fe content in the particles are shown. Additionally, the plot con-

224 tains the relative abundances of Fe, Si, and Fe+Si for the natural occurring minerals silica (SiO_2),
 225 magnetite (Fe_3O_4), hematite (Fe_2O_3), ferrosilite (FeSiO_3), and fayalite (Fe_2SiO_4) according to their the-
 226 oretical stoichiometric composition shown as open symbols. At zero iron content, the nanoparticle
 227 composition coincides with the composition of silica, which agrees with a previous finding in a similar
 228 experimental arrangement (41). At zero silicon content, the nanoparticle composition matches the iron
 229 to oxygen ratio of Fe_2O_3 . This compares well with studies in which microwave plasma particle sources
 230 were used to produce iron oxide nanoparticles and which identified the particle material as Fe_2O_3 us-
 231 ing XRD, Raman spectroscopy and Mössbauer spectroscopy (42, 43). At intermediate iron content
 232 ($0.2 < \text{Fe}:(\text{Fe}+\text{Si}) < 0.8$), the particle composition does not show a linear transition between SiO_2 and
 233 Fe_2O_3 . Instead, the particle composition follows a lower M:O ratio and Fe appears to be continuously
 234 interchangeable with Si. In fact, the measured composition is well represented by the system $\text{Fe}_x\text{Si}_{(1-x)}\text{O}_3$
 235 ($0 \leq x \leq 1$) which is plotted in Fig 4 as solid lines. The continuous exchange of Fe and Si indicates
 236 that the particle formation occurs by polymerization of gas phase FeO_3 and SiO_3 . In contrast to the
 237 production of iron oxide and silica particles, further rearrangement to more stable silicates does not
 238 occur. In addition, we observed a 10 to 100 times higher particle mass production rate when mixed
 239 iron silicates were produced compared to only using one precursor material. These observations show
 240 that the chemistry involved in particle formation is altered when producing mixed iron silicates. Photo-
 241 oxidization in the presence of O_2 and O_3 at room temperature and atmospheric pressure has shown to
 242 produce a different particle material (25, 26): Here, independent of the applied gas phase concentra-
 243 tions of the iron and silicon precursors, only an iron to silicon ratio of 2 was produced. This reaction
 244 product corresponds to the mineral fayalite (Fe_2SiO_4) and indicates a different reaction pathway.
 245 However, the composition as determined by EDS does not necessarily reflect the composition of indi-
 246 vidual particles as the scan area of the electron beam ($200 \times 200 \mu\text{m}$) always covers many deposited
 247 nanoparticles. The EDS results are an average of the composition of the particles deposited on the
 248 sample grids. To elucidate the particle mixing state, additional information was provided by measuring
 249 the surface sensitive adsorption and desorption of H_2O on the nanoparticles.

250 3.2 H_2O desorption energy

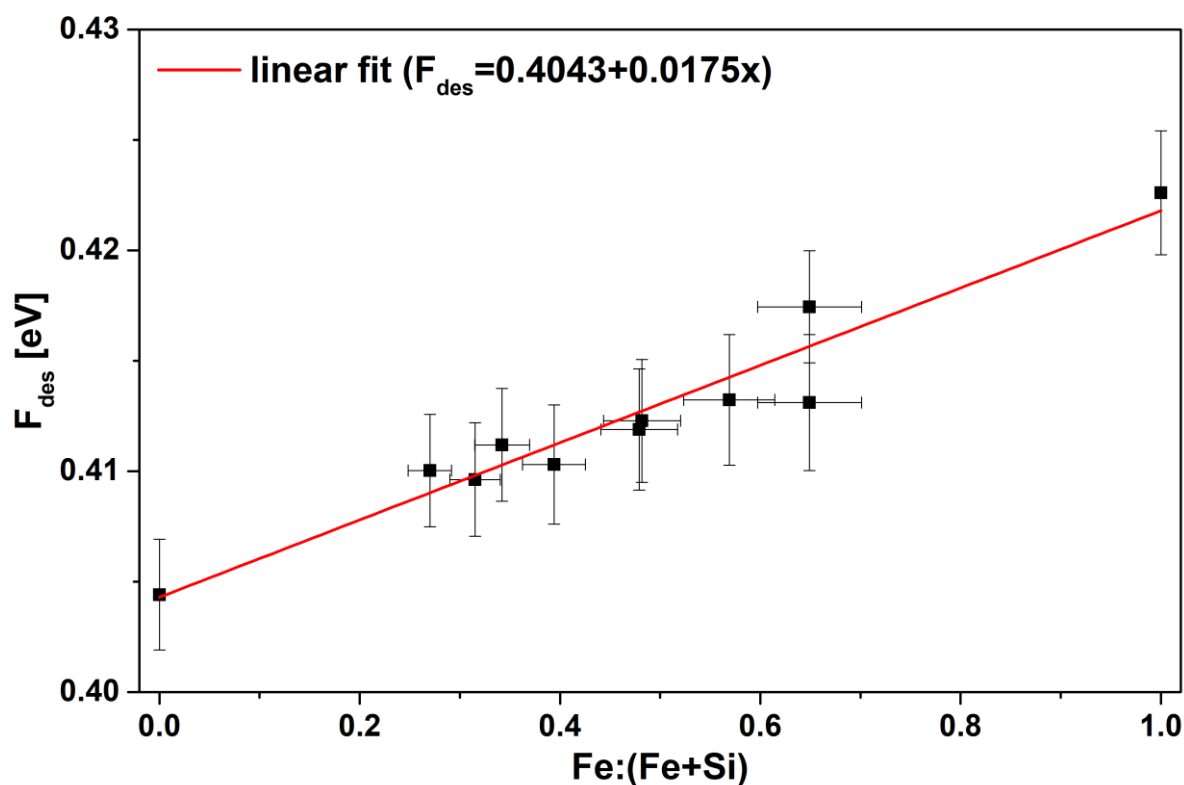
251 The water vapor surface desorption energy of oxide nanoparticles with varying Fe and Si content has
 252 been measured using the method described above. One example measurement is shown in the left
 253 panel of Fig. 5 for a mixing ratio of $\text{Fe}:(\text{Fe}+\text{Si}) = 0.65$.



254

255 **Fig. 5:** Left panel: Nanoparticle mass against residence time in MICE for mixed iron and silicon containing particles. Right panel:
 256 Smoothed TOF mass spectrum corresponding to the point marked with error bars at $t=83\text{s}$.

257 The right panel depicts the TOF spectrum corresponding to the single data point marked with error
 258 bars at $t=83$ s residence time. The TOF spectrum shows a single peak only with a peak width which is
 259 identical to the peak width for pure nanoparticles. As we have argued above, this indicates that all
 260 particles of the trapped particle population indeed adsorb the same amount of water molecules and
 261 thus have the same desorption energy. This finding eliminates the possibility of two or more particle
 262 classes with different desorption energies present in the trapped particle cloud.
 263 To further investigate the particle mixing state, a series of desorption energy measurements were per-
 264 formed with particles of 2 nm in radius and varying Fe:(Fe+Si) particle mixing ratios and the results are
 265 shown in Fig. 6. The data shows a linear transition of the desorption energy from pure silica to pure
 266 iron-oxide particles. This indicates that the surface composition also follows a smooth transition and
 267 that the stoichiometric composition which has been determined by EDS measurements directly reflects
 268 the surface properties of the nanoparticles. We conclude that the nanoparticles are completely inter-
 269 nally mixed and that differential coating to a core-shell structure does not occur.



270

271 **Fig. 6:** H₂O desorption energy for nanoparticles ($r = 2\text{nm} \pm 0.2\text{nm}$) of varying Fe and Si content. The solid line represents a
 272 linear fit to the data.

273 4 Summary and Outlook

274 In this study, we characterized the composition of MSP analogue nanoparticles produced in a non-
 275 thermal low-pressure microwave plasma source. EDS measurements show that silicon oxide particles
 276 consist of silica (SiO₂) and iron oxide particles are composed of Fe₂O₃. In addition, mixed iron silicates
 277 with an Fe:(Fe+Si) ratio between 0.2 and 0.8 are present in the form of Fe_xSi_(1-x)O₃ ($0 \leq x \leq 1$) with the
 278 ratio of Fe to Si molecules reflecting the iron and silicon molecule concentration in the plasma reactor.
 279 H₂O adsorption measurements using MICE-TRAPS were used to conclude that only one particle class
 280 is produced, that the nanoparticles are internally mixed and that differential coating to a core-shell
 281 structure does not occur. In future work, we will add a magnesium precursor to the system and study
 282 the composition of mixed magnesium-iron-silicates. Measurements of the material dependent light
 283 absorption coefficient of these now well characterized particles as well as their H₂O nucleation ability
 284 will be part of upcoming publications.

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