# Single-ion conducting polymer electrolyte for $\mathrm{Li}\left|\mid \mathrm{LiNi} \mathrm{Ni}_{0.6} \mathrm{Mn}_{0.2} \mathrm{Co}_{0.2} \mathrm{O}_{2}\right.$ batteries-impact of the anodic cutoff voltage and ambient temperature 

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#### Abstract

Polymer-based electrolytes potentially enable enhanced safety and increased energy density of lithium-metal batteries employing high capacity, transition metal oxide-positive electrodes. Herein, we report the investigation of lithium-metal battery cells comprising $\mathrm{Li}\left[\mathrm{Ni}_{0.6} \mathrm{Mn}_{0.2} \mathrm{Co}_{0.2}\right] \mathrm{O}_{2}$ as active material for the positive electrode and a poly(arylene ether sulfone)-based singleion conductor as the electrolyte incorporating ethylene carbonate (EC) as selectively coordinating molecular transporter. The resulting lithium-metal battery cells provide very stable cycling for more than 300 cycles accompanied by excellent average Coulombic efficiency ( $99.95 \%$ ) at an anodic cutoff potential of 4.2 V . To further increase the achievable energy density, the stepwise increase to 4.3 V and 4.4 V is herein investigated, highlighting that the polymer electrolyte offers comparable cycling stability, at least, as common liquid organic electrolytes. Moreover, the impact of temperature and the EC content on the rate capability is evaluated, showing that the cells with a higher EC content offer a capacity retention at 2 C rate equal to $61 \%$ of the capacity recorded at 0.05 C at $60^{\circ} \mathrm{C}$.


Keywords Polymer electrolyte $\cdot$ Single-ion conductor $\cdot \mathrm{NMC}_{622} \cdot$ Cycling parameters $\cdot$ Lithium battery

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## Introduction

Rechargeable lithium-metal batteries are considered the next great leap forward toward higher energy densities [1]. Nevertheless, the severe risk of lithium dendrite formation, potentially causing a short circuit of the cell, and the continuous electrolyte decomposition at the electrode|electrolyte interface have so far hampered the commercial exploitation of such batteries-with one little exception: lithium-polymer batteries comprising an electrolyte based on poly(ethylene oxide) (PEO) [2]. This polymer electrolyte, however, suffers of two major drawbacks related to the operating temperature of about $60-80{ }^{\circ} \mathrm{C}$ (especially during charge, when Li plating occurs) and the choice of the active material for the positive electrode, which is limited to materials that are de-/lithiated below 4 V , such as $\mathrm{LiFePO}_{4}$ [2-6]. These two issues originate from the facts that the charge transport is coupled with the segmental motion of the polymer, which is faster the higher the temperature, and that the lithium-coordinating ether group and/or terminal hydroxyl groups are not sufficiently stable toward
oxidation beyond 4 V [4, 7-11]. Another issue is the potential reversed cell polarization at elevated current densities, leading to accelerated cell failure and favoring inhomogeneous (dendritic) lithium deposition [12-15].

Very recently, we have reported a new multi-block co-poly(arylene ether sulfone) electrolyte, which addresses these issues by covalently tethering the anionic function to the polymer backbone, stabilizing the ether group through adjacent electron-withdrawing groups, and introducing, e.g., ethylene carbonate (EC) as "molecular transporter" to actively facilitate $\mathrm{Li}^{+}$conduction rather than simply plasticizing the ionomer, the latter effect being facilitated by the selective coordination of EC to the ionophilic block (psi-PES) in which the charge transport occurs, while the ionophobic blocks (FPES) provide mechanical stability [16]. This nano-phase-separated, single-ion-conducting, multi-block copolymer electrolyte comprising EC allows for the stable cycling of $\mathrm{Li}\left|\mid \mathrm{Li}\left[\mathrm{Ni}_{1 / 3} \mathrm{Mn}_{1 / 3} \mathrm{Co}_{1 / 3}\right] \mathrm{O}_{2}\left(\mathrm{NMC}_{111}\right)\right.$ full cells with an anodic cutoff voltage of 4.2 V for more than 200 cycles at $40{ }^{\circ} \mathrm{C}$ with a specific capacity of about $100 \mathrm{mAh} \mathrm{g}^{-1}$ at $0.2 \mathrm{C}\left(32 \mathrm{~mA} \mathrm{~g}{ }^{-1}\right)$ [16].

Herein, we investigate the use of this multi-block copolymer electrolyte in high-energy $\mathrm{Li}\left|\mid \mathrm{Li}\left[\mathrm{Ni}_{0.6} \mathrm{Mn}_{0.2} \mathrm{Co}_{0.2}\right] \mathrm{O}_{2}\right.$ $\left(\mathrm{NMC}_{622}\right)$ cells, as schematized in Fig. 1, with particular regard to the impact of the stepwise increasing anodic cutoff voltage from 4.2 V to 4.3 V and 4.4 V . Moreover, we investigated the effect of increasing the EC concentration and the ambient temperature on the achievable rate capability, revealing a very good capacity retention even at 2 C thanks to an ionic conductivity well above $1 \mathrm{mS} \mathrm{cm}^{-1}$ under such conditions.

## Experimental

## Copolymer synthesis

The synthesis of the partially fluorinated multi-block poly(arylene ether sulfone) with covalently tethered lithium perfluorosulfonimide functions (herein referred to as SI), serving as electrolyte, and its characterization have been described in detail in Nguyen et al. [16]. Briefly, the block copolymer backbone (with block lengths of $15 \mathrm{~kg} \mathrm{~mol}^{-1}$ ) was synthesized via co-polycondensation and subsequently region-selectively brominated in order to allow for the covalent tethering of the sulfonimide anion using Ullman's coupling reaction [17] and lithium 1,1,2,2-tetrafluoro-2-(1,1,2,2-tetrafluoro-2iodoethoxy)ethanesulfonimide (I-psiLi) as precursor. EC was purchased from Merck or BASF and used as received. The copolymer was characterized via ${ }^{1} \mathrm{H}$ NMR and ${ }^{19} \mathrm{~F}$ NMR spectroscopy to confirm the molecular architecture. Sizeexclusion chromatography coupled with a multiangle light scattering detector revealed a weight-average molecular weight $\left(\mathrm{M}_{\mathrm{w}}\right)$ of $724 \mathrm{~kg} \mathrm{~mol}^{-1}$. The solvent content (SC), indicated as $x$ in SL $x \%$, is defined as the ratio between the mass of the EC-swollen membrane $\left(\mathrm{M}_{\mathrm{s}}\right)$ minus the mass of the dry membrane $\left(\mathrm{M}_{\mathrm{d}}\right)$ and $\mathrm{M}_{\mathrm{s}}$, multiplied by $100 \%$ [16]:
$\mathrm{SC}=\frac{\mathrm{Ms}-\mathrm{Md}}{\mathrm{Ms}} \times 100 \%$
The handling and processing of the ionomer membranes were conducted either in an argon-filled glove box or in the dry room to avoid a relevant impact of moisture.

Fig. 1 Molecular architecture of the multi-block copolymer electrolyte and schematic illustration of its nanophase separation including the EC molecules selectively coordinating the ionophilic blocks as well as the eventual $\mathrm{Li}\left|\mid \mathrm{NMC}_{622}\right.$ full cell incorporating such electrolyte


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## Electrode preparation and electrochemical characterization

The $\mathrm{NMC}_{622}$ electrode preparation was performed in the dry room as well. $\mathrm{NMC}_{622}$ (industrial source) was dispersed together with Super C65 (IMERYS) and poly(vinylidene difluoride) (PVdF, Solvay) in $N$-methyl-2-pyrrolidone (NMP, Aldrich) via magnetic stirring ( 3 h at 500 rpm ). The resulting slurry was cast on aluminum foil using a laboratory-scale doctor blade with a wet film thickness of $130 \mu \mathrm{~m}$. The resulting electrode sheets were dried at $60^{\circ} \mathrm{C}$ overnight. Disk-shaped electrodes $(\varnothing=12$ mm ) were punched from the thus pre-dried electrode sheets and subsequently dried at $120^{\circ} \mathrm{C}$ for 12 h under vacuum. Eventually, the electrodes were pressed at 10 t for 10 s . The total electrode composition was $88 \mathrm{wt} \% \mathrm{NMC}_{622}, 7 \mathrm{wt} \%$ Super C65, and $5 \mathrm{wt} \% \mathrm{PVdF}$. The active material mass loading (i.e., the mass loading of $\mathrm{NMC}_{622}$ ) was around $2.0 \pm 0.2 \mathrm{mg} \mathrm{cm}^{-2}$. For the electrochemical characterization, two-electrode, Swagelok-type cells were assembled sandwiching the ionomer electrolyte membrane between the $\mathrm{NMC}_{622}$ electrode and the lithium metal electrode (Honjo, battery grade). The whole cell was subjected to a pressure ( $4-5 \mathrm{t}$ ) for 3 min to infiltrate the ionomer electrolyte into the porous $\mathrm{NMC}_{622}$ electrode. Galvanostatic cycling was conducted using a Maccor 4000 battery tester. The temperature was controlled by placing the cells in climatic chambers (Binder). The cathodic cutoff voltage was kept constant at 2.8 V throughout all the tests reported herein, while the anodic cutoff voltage was varied from 4.2 to 4.3 V and 4.4 V . A dis-/charge rate of 1 C corresponds to a specific current of $160 \mathrm{~mA} \mathrm{~g}^{-1}$, with the mass in $g$ referring to the active material


Fig. 2 Galvanostatic cycling of Li|SI55\% $\mid \mathrm{NMC}_{622}$ cells with an anodic cutoff voltage of 4.2 V . a Dis-/charge profiles of the initial five cycles at 0.05 C . b Plot of the specific discharge capacity and Coulombic efficiency vs. the cycle number with the C rate being increased to 0.5 C after the first five cycles (temperature $40^{\circ} \mathrm{C}$ )
mass loading, i.e., $\mathrm{NMC}_{622}$. Accordingly, all capacity values reported herein refer to the active material mass loading as well. We have chosen this specific current to define a dis-/charge rate of 1 C in order to allow for a direct comparison with our previous work [16], despite the fact that this is not the theoretical capacity of $\mathrm{NMC}_{622}$ as active material for the positive electrode. For comparison, also cells with a liquid organic electrolyte ( $1 \mathrm{M} \mathrm{LiPF}_{6}$ in EC-DMC, 1:1 by weight; UBE) were assembled and tested. For the determination of the limiting current density, two-electrode pouch cells were assembled with the ionomer electrolyte membranes sandwiched between two lithium foils. The cells were kept in a climatic chamber at the corresponding temperature for 6 h prior to the application of a sweep rate of $0.025 \mathrm{mV} \mathrm{s}^{-1}$ using a Solatron 1400 CellTest system.

## Results and discussion

To start with, $\mathrm{Li}|\mathrm{SI} 55 \%| \mathrm{NMC}_{622}$ cells were subjected to galvanostatic cycling at $40^{\circ} \mathrm{C}$, setting the anodic cutoff voltage to 4.2 V in order to have a direct comparison with the results reported earlier for $\mathrm{Li}|\mathrm{SI} 55 \%| \mathrm{NMC}_{111}$ cells [16]. Figure 2 a shows the dis-/charge profiles for the initial five formation


Fig. 3 a Rate capability and cycling stability test for $\mathrm{Li}|\mathrm{SI} 55 \%| \mathrm{NMC}_{622}$ cells with an anodic cutoff voltage of 4.3 V (orange) and 4.4 V (turquoise). The dis-/charge rate was increased initially every five cycles from 0.05 C to $0.1 \mathrm{C}, 0.2 \mathrm{C}, 0.5 \mathrm{C}, 1 \mathrm{C}$, and 2 C before subjecting the cells to constant current cycling at 0.5 C for additional 200 cycles (in the case of 4.3 V ) and 100 cycles (in the case of 4.4 V ). In both cases, only the (reversible) discharge capacity is shown. The temperature was set to 40 ${ }^{\circ} \mathrm{C}$. b Comparison of the rate capability and cycling stability of $\mathrm{Li}|\mathrm{S} 55 \%| \mathrm{NMC}_{622}$ cells (same as in (a)) and Li| $\mathrm{NMC}_{622}$ cells comprising a liquid organic electrolyte ( $1 \mathrm{M} \mathrm{LiPF}_{6}$ in $\mathrm{EC} / \mathrm{DMC}, 1: 1$ by weight). For the latter test, the temperature was set to $20^{\circ} \mathrm{C}$
cycles at 0.05 C . The cell delivers a specific capacity of 157 $\mathrm{mAh} \mathrm{g}^{-1}$, while the first cycle Coulombic efficiency is about $85 \%$. The subsequent long-term cycling ( 300 cycles performed at 0.5 C ) is presented in Fig. 2b. The delivered capacity during the first cycles is $93 \mathrm{mAh}^{-1}$, which slightly decreased upon cycling to stabilize at about $85 \mathrm{mAh} \mathrm{g}^{-1}$. As a result, the capacity retention was $91.1 \%, 87.2 \%$, and $81.1 \%$ after 100,200 , and 300 cycles, respectively. This impressive cycling stability considering the use of the Li metal electrode is accompanied by a very high average Coulombic efficiency of $99.95 \%$, which is remarkably high as a result of the excellent compatibility of the ionomer electrolyte with Li metal.

In a next step, the cells were tested using higher upper cutoff voltage ( 4.3 V and 4.4 V : see Fig. 3a). The dis-/charge rate was initially varied from 0.05 up to 2 C to investigate the rate performance and then kept constant at 0.5 C in order to study the cycling stability. At 0.05 C , the reversible specific capacity, i.e., the capacity obtained upon lithiation (discharge), increases from $157 \mathrm{mAh} \mathrm{g}^{-1}$ (@ 4.2 V ) to 162 and $173 \mathrm{mAh} \mathrm{g}^{-1}$ when elevating the anodic cutoff voltage to 4.3 and 4.4 V , respectively. This trend is maintained increasing the C rate to 0.1 C with slightly higher capacities for the $\mathrm{Li}|\mathrm{SI} 55 \%| \mathrm{NMC}_{622}$ cycled with an anodic cutoff of 4.4 V . When further increasing the dis-/charge rate to 0.2 C , however, very similar capacities were obtained. Finally, at even higher C rates, the cells cycled with an anodic cutoff of 4.3 V showed higher capacities than the cell cycled with an anodic cutoff of
4.4 V. Precisely, specific discharge capacities of about 108, 86 , and $49 \mathrm{mAh}^{-1}$ were obtained at $0.5 \mathrm{C}, 1 \mathrm{C}$, and 2 C , respectively, for an anodic cutoff of 4.3 V , while setting the anodic cutoff to 4.4 V led to capacities of around 103,78 , and $36 \mathrm{mAh} \mathrm{g}^{-1}$ at $0.5 \mathrm{C}, 1 \mathrm{C}$, and 2 C , respectively. After this rate capability test, the dis-/charge rate was set constantly to 0.5 C to evaluate the general cycling stability. For an anodic cutoff of 4.3 V , the cells provided very stable cycling with about $73.8 \%$ capacity retention after 200 cycles (i.e., about 80 $\mathrm{mAh} \mathrm{g}^{-1}$ ). A very similar capacity retention of $74.4 \%$ was obtained for an anodic cutoff of 4.4 V , but after 100 cycles. These results suggest that the detrimental reactions occurring at the interface between the cathode and the ionomer electrolyte are more pronounced when elevating the anodic cutoff voltage to 4.4 V and that this effect outweighs the initially beneficial impact on the achievable specific capacity. This is in good agreement with a recent study on a very similar ionomer electrolyte system [18]. Remarkably, though, the capacity retention is slightly higher than for $\mathrm{Li}\left|\mid \mathrm{NMC}_{622}\right.$ cells comprising a common liquid organic electrolyte $\left(1 \mathrm{M} \mathrm{LiPF}_{6}\right.$ in EC/DMC), i.e., $74.4 \%$ vs. $74.0 \%$, as depicted in Fig. 3bdespite the generally higher specific capacity across all dis-/ charge rates, which is assigned to the substantially higher ionic conductivity ( $>10 \mathrm{mS} \mathrm{cm}^{-1}$ at $20^{\circ} \mathrm{C}^{2}$ vs. $<1 \mathrm{mS}$ $\mathrm{cm}^{-1}$ at $40^{\circ} \mathrm{C}$ [16]) and potentially a better penetration of the electrolyte into the electrode's pores. This result indicates that the fading is, at least partially, related to the performance


Fig. 4 Investigation of the impact of the EC content and ambient temperature on the rate capability of $\mathrm{Li} \| \mathrm{NMC}_{622}$ cells. a $\mathrm{Li}|\mathrm{S} 155 \%| \mathrm{NMC}_{622}$ at $40{ }^{\circ} \mathrm{C}$ (see also Fig. 3a). b Li|SI65\% $/ \mathrm{NMC}_{622}$ at
of the cathode-active material itself when setting the upper cutoff to 4.4 V and/or that the ionomer electrolyte offers the same oxidation stability of organic carbonate-based liquid electrolytes, in spite of the reportedly poor stability of ethylene carbonate at elevated potentials [19].

Following these results, 4.3 V was chosen as the anodic cutoff voltage for the subsequent investigation of the impact of the EC content and ambient temperature on the rate capability of $\mathrm{Li}|\mathrm{SIx} \%| \mathrm{NMC}_{622}$ cells (Fig. 4). In Fig. 4a, the evaluation of the rate capability for $\mathrm{Li}|\mathrm{SI} 55 \%| \mathrm{NMC}_{622}$ at $40^{\circ} \mathrm{C}$ is shown again, serving as reference. The same test was subsequently applied for $\mathrm{Li}|\mathrm{SI} 65 \%| \mathrm{NMC}_{622}$ cells, i.e., employing a higher EC content in the ionomer membrane (Fig. 4b). This increase in EC concentration leads to higher capacities across all dis-/charge rates, particularly, at C rates above 0.5 C . At 1 C , for instance, the capacity increased from about 86 to 102 mAh $\mathrm{g}^{-1}$, and at 2 C , the capacity increased from around 49 to 78 $\mathrm{mAh} \mathrm{g}^{-1}$, corresponding to relative increases of about $19 \%$ and $58 \%$ at 1 C and 2C, respectively. This superior rate capability is assigned to the improved ionic conductivity for SI65\% compared with SI55\% (around $1.2 \mathrm{mS} \mathrm{cm}^{-1}$ vs. 0.5 $\mathrm{mS} \mathrm{cm}{ }^{-1}$ at $40{ }^{\circ} \mathrm{C}$ [16]) and the enhanced limiting current density ( $1.3 \mathrm{~mA} \mathrm{~cm}^{-2}$ at $40^{\circ} \mathrm{C}$ vs. $1.2 \mathrm{~mA} \mathrm{~cm}^{-2}$ at $50^{\circ} \mathrm{C}$ ). In fact, when increasing the ambient temperature to $60^{\circ} \mathrm{C}$ (Fig. 4c), a further rate capability improvement is observed-also at the lowest C rate of 0.05 C , resulting in a specific capacity of about $171 \mathrm{mAh}^{-1}$ vs. $160 \mathrm{mAh} \mathrm{g}^{-1}$ for the $\mathrm{Li}|\mathrm{SI} 65 \%| \mathrm{NMC}_{622}$ cells. At 2C, the specific capacity was still around $104 \mathrm{mAh} \mathrm{g}^{-1}$, which translates into a capacity retention of around $61 \%$ with regard to the capacity obtained at 0.05 C -or an increase in capacity by $34 \%$ compared with the $\mathrm{Li}|\mathrm{SI} 65 \%| \mathrm{NMC}_{622}$ cells run at $40^{\circ} \mathrm{C}$. As a matter of fact, the enhanced kinetics is reflected also in an increased limiting current density with an increased temperature of around $1.5 \mathrm{~mA} \mathrm{~cm}^{-2}$ (vs. $1.3 \mathrm{~mA} \mathrm{~cm}^{-2}$ at $40^{\circ} \mathrm{C}$ ). The overall comparison of the rate capability when varying the EC content and elevating the ambient temperature is summarized in Fig. 4d, highlighting the stepwise improvement for an increased EC concentration and testing temperature.

## Conclusions

$\mathrm{Li} \mid \mathrm{NMC}_{622}$ cells comprising single-ion-conducting SLx $\%$ as electrolyte provide excellent cycling stability for more than 300 cycles. Elevating the anodic cutoff voltage to 4.3 and 4.4 V leads to a slight decrease in cycling stability. However, the comparison with a common liquid organic electrolyte reveals that this decrease in cycling stability is related to either the cathode active material and/or the presence of EC, while the ionomer electrolyte is at least as stable as the liquid electrolyte at such elevated cutoff voltages. The rate capability, however, is lower for the $\mathrm{Li}\left|\mid \mathrm{NMC}_{622}\right.$ cells containing the
ionomer electrolyte, though this can be substantially enhanced when increasing the EC content and/or applying elevated temperatures due to the increased ionic conductivities and limiting current densities at such conditions.

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## Compliance with ethical standards

Conflict of interest The authors declare that they have no competing interests.

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