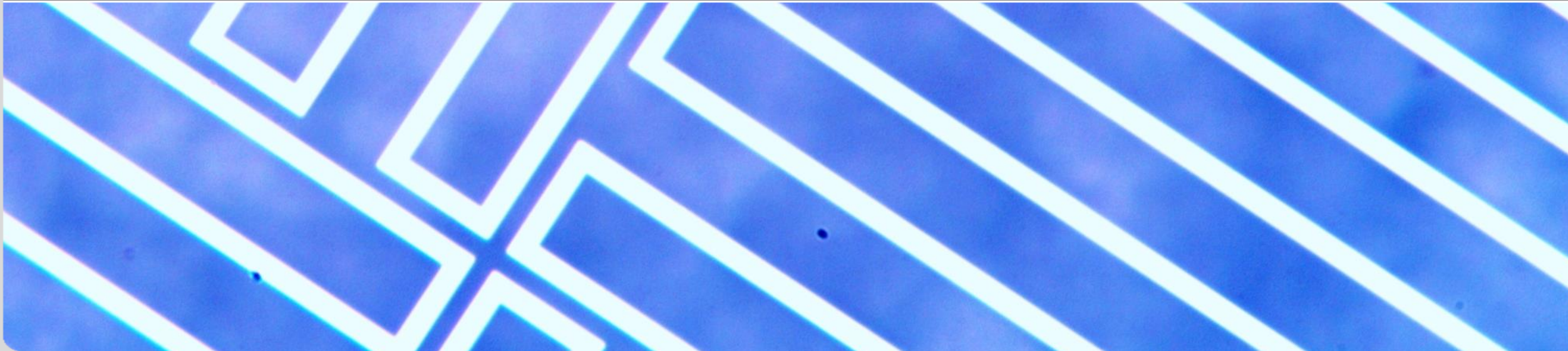


Development of diamond based KIDs

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Institute of Applied Materials (IAM-AWP), Karlsruhe Institute of Technology



Contents

1. Introduction

- Application
- Kinetic Inductance Detectors
- Why diamond?

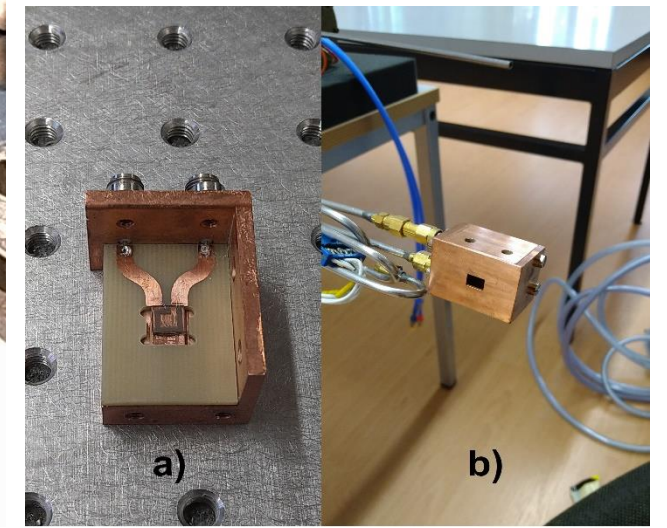
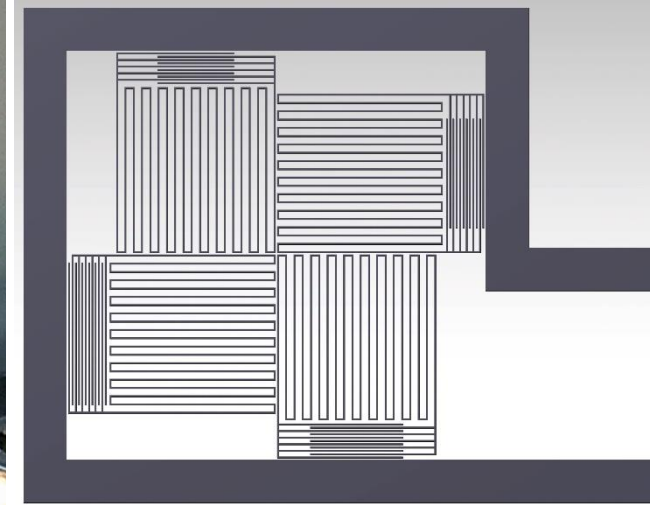
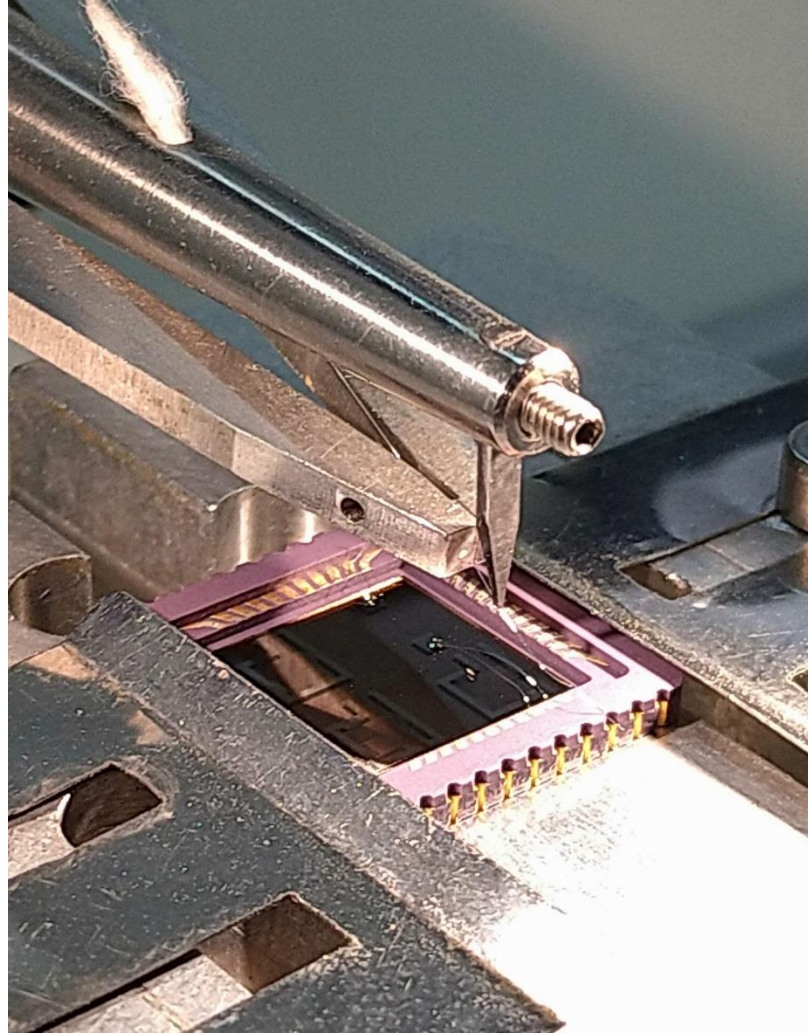
2. Simulations

- Frequency tuning
- Cross-Talk analysis
- Quality factors tuning
- Complete detector simulations

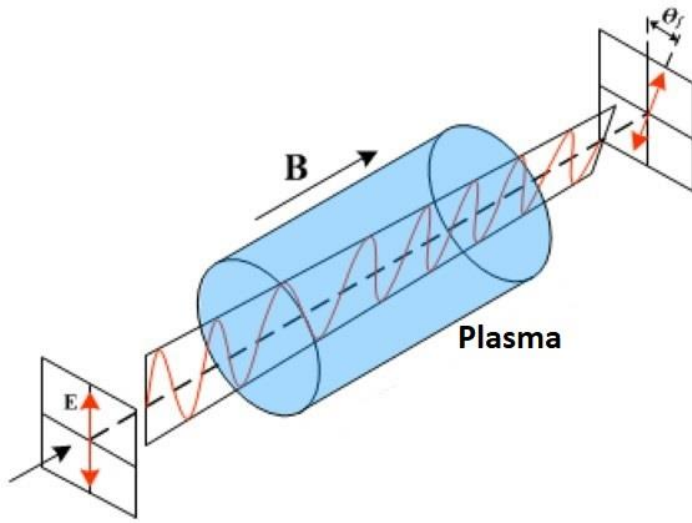
3. Experiments

- SC thin film characterization
- Microwave characterization
- THz characterization

4. Conclusions



Introduction – Application



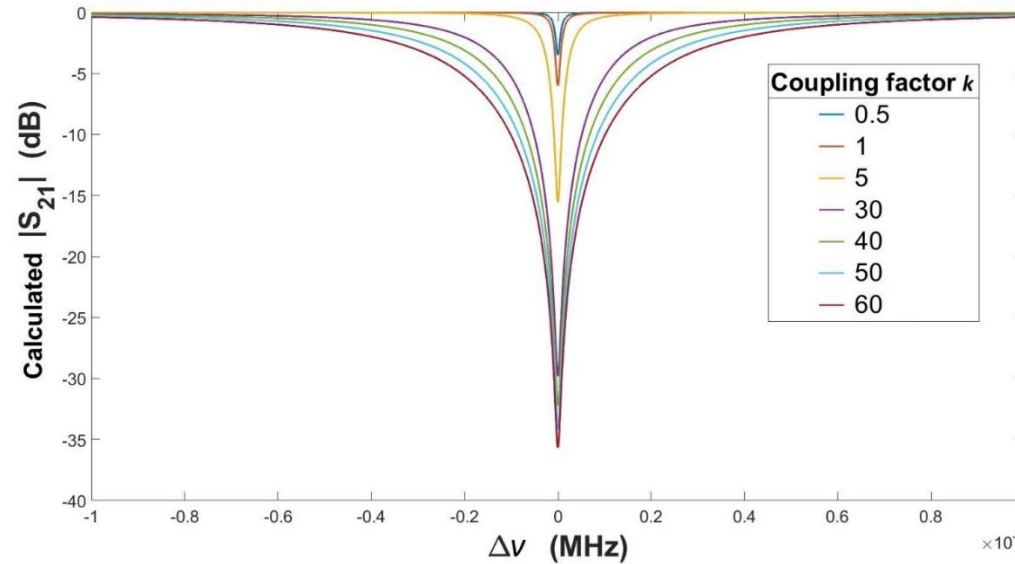
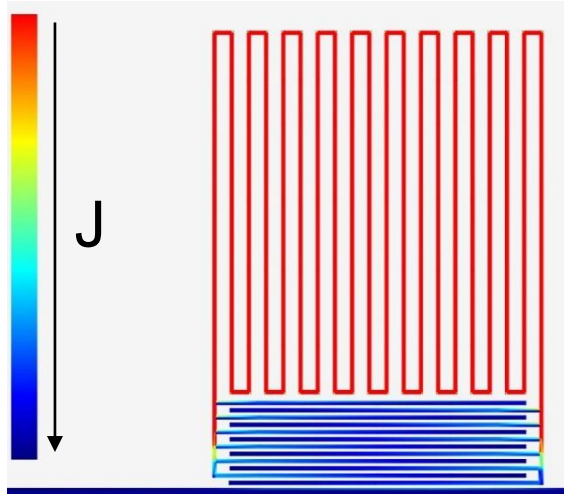
$$n_L \neq n_R \rightarrow \Delta\varphi \rightarrow \theta_f$$

$$\theta_F = \frac{e^3}{2\epsilon_0 m_e^2 c \omega^2} \int n_e(z) B_{\parallel} dz,$$

$$\int n_e(z) B_{\parallel} dz \approx J.$$

- Polarimetric diagnostic for nuclear fusion plasma
- THz spectral detection range
- Incident power: $\mu\text{W} \rightarrow$ a few mW
- Cross polarization rejection capabilities, multi-pixel design
- Radiation hardness
- Small footprint
- Reliability \rightarrow few moving parts

Introduction - Kinetic Inductance Detectors



$$\kappa = \frac{|S_{11,max}|}{|S_{21,min}|} = \frac{1 - |S_{21,min}|}{|S_{21,min}|}$$

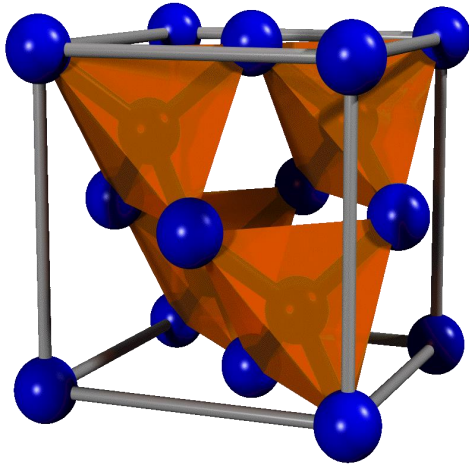
$$Q_L = \frac{\nu_{res}}{\Delta\nu_T}$$

$$Q_0 = Q_L(1 + \kappa) = \frac{Q_L}{|S_{21,min}|}$$

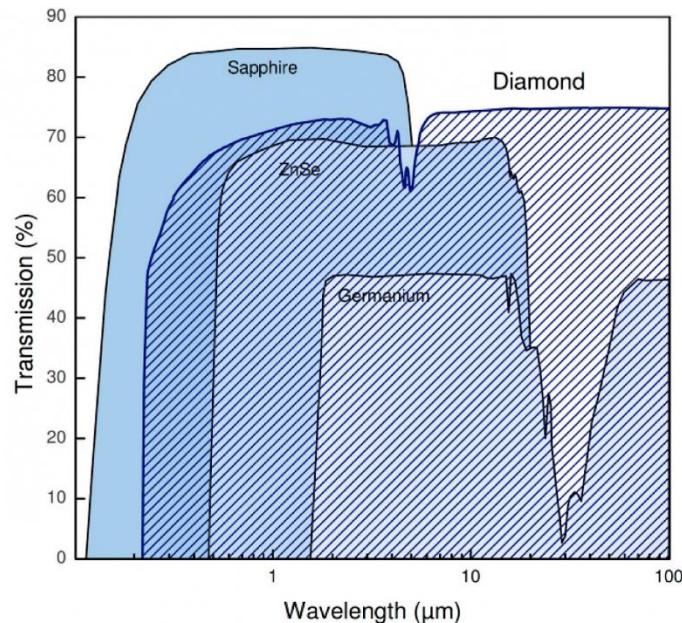
$$Q_C = \frac{Q_0}{\kappa} = \frac{Q_L}{1 - |S_{21,min}|}$$

- Superconducting (SC) thin film LC micro-resonators
- Photons \rightarrow Cooper pair breaking \rightarrow excess $e^- \rightarrow L_K \uparrow \rightarrow S_{21}$
 $\Delta\omega$ & $\Delta\phi$
- LEKID: discrete inter digital capacitor (IDC) and inductor \rightarrow easy freq. tuning (FDM) & inductor uniform current
- Depth and width of the resonances are strongly dependent on coupling k
- Design optimization: no saturation & no overlap \rightarrow evaluation of coupling strength/ Q_C
- SC of choice: Niobium Nitride (T_C (bulk) >16 K)

Introduction – Why diamond based detectors?



- Extremely low heat capacity C and extremely high thermal conductivity \rightarrow fast response (thermal detectors)
- Transparency \rightarrow no substrate events
- Low losses and low dielectric constant (especially at microwave frequencies) \rightarrow potential for very high internal quality factor
- Radiation hardness
- Mechanical stability \rightarrow thin substrates ($<20 \mu\text{m}$) possible
- High breakdown voltage \rightarrow no radiation generated charge carriers \rightarrow transparent at very high radiation intensities.



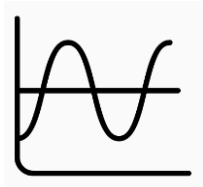
- Diamond and Related Materials, 3 (1994) 747-751
- IEEE TRANSACTIONS ON APPLIED SUPERCONDUCTIVITY, VOL. 3, NO.1, MARCH 1993

Detector design and optimization - simulations



SONNET Software

Analysis: Maxwell's equations, solving for J planar multilayer geometries
Parameters: L_k , ϵ_r , $\tan(\delta)$



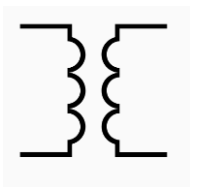
Frequency Tuning

Single pixel analysis, variation of IDC finger numbers and length



Cross Talk Analysis

2 identical pixels, bare / box / open box inter-pixel distance variation



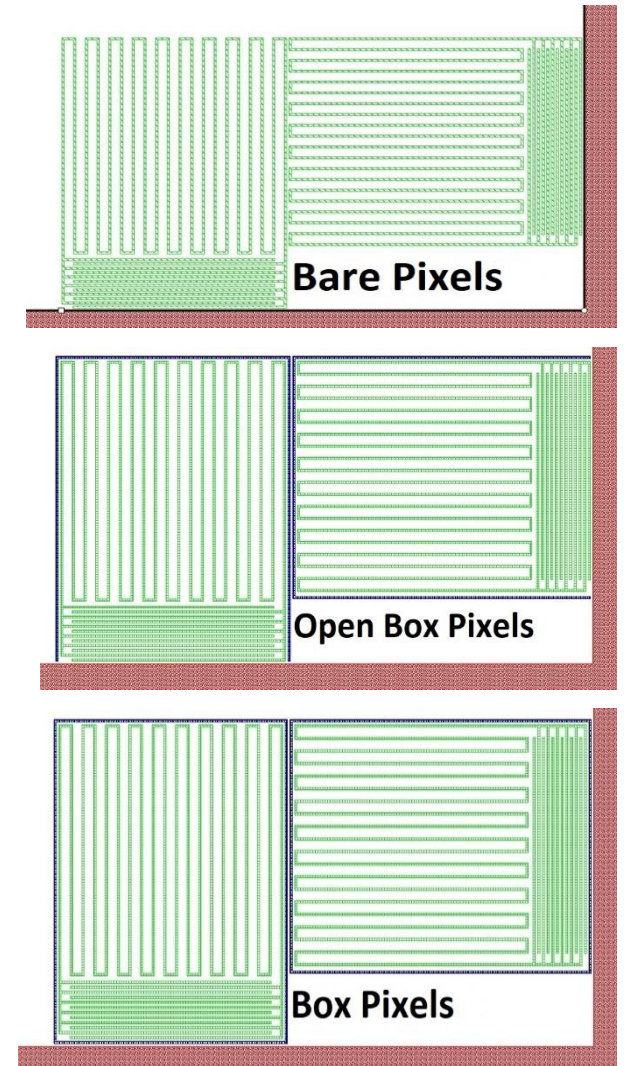
Q_c / t-line coupling

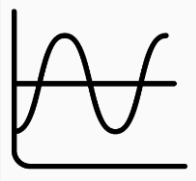
Single pixel, bare / box / open box feedline – pixel distance variation



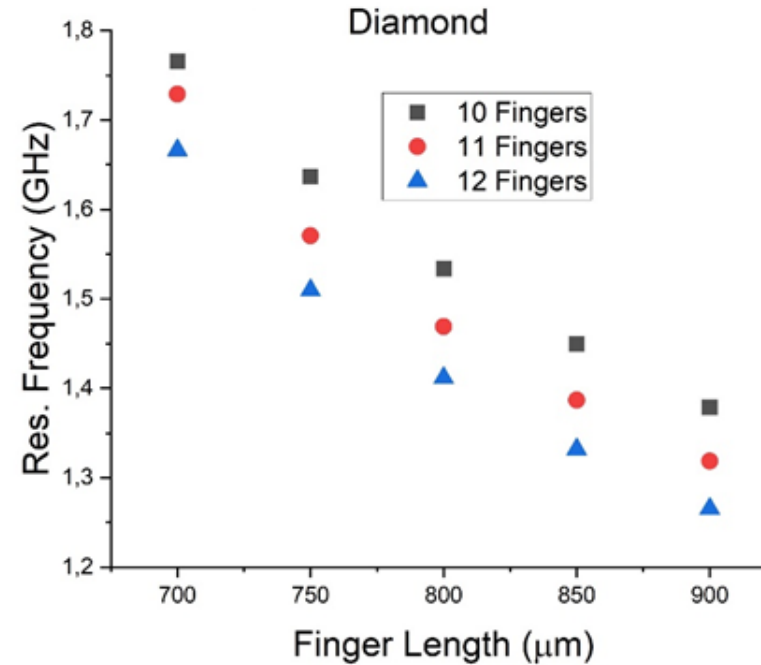
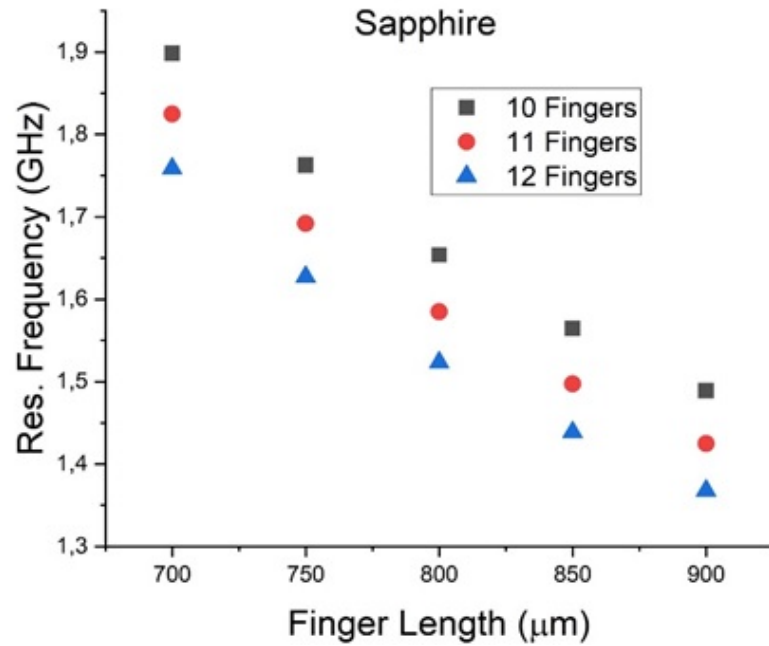
Substrate materials

Silicon (not shown), Sapphire, Diamond (poly)





Frequency Tuning

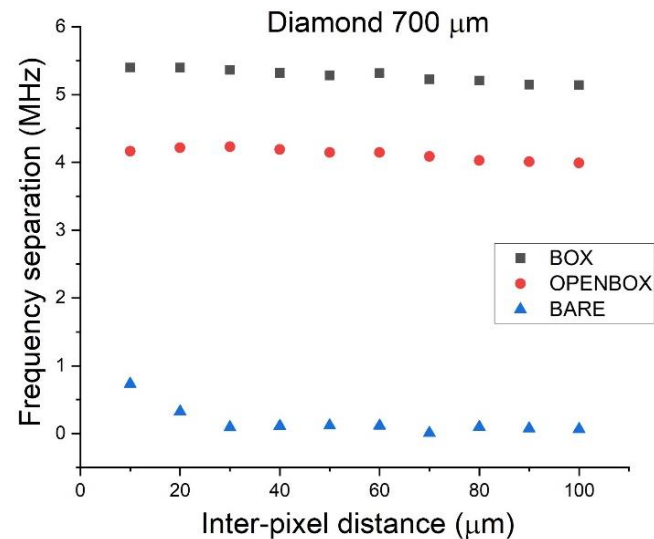
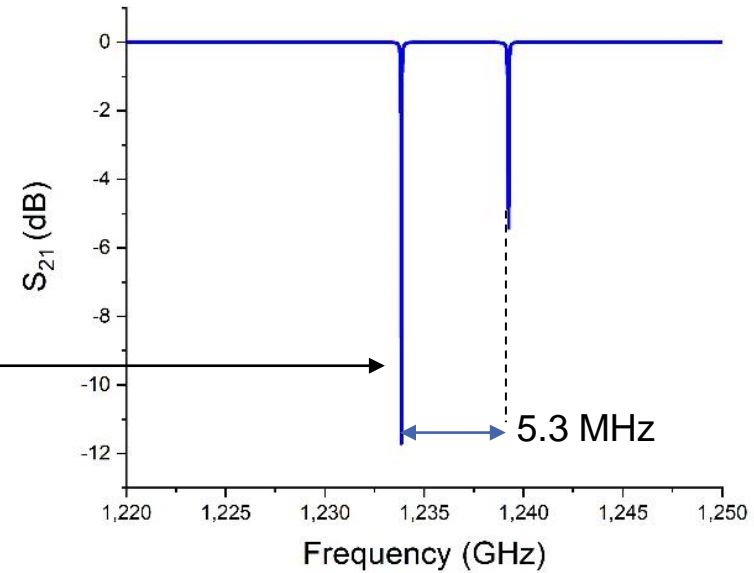
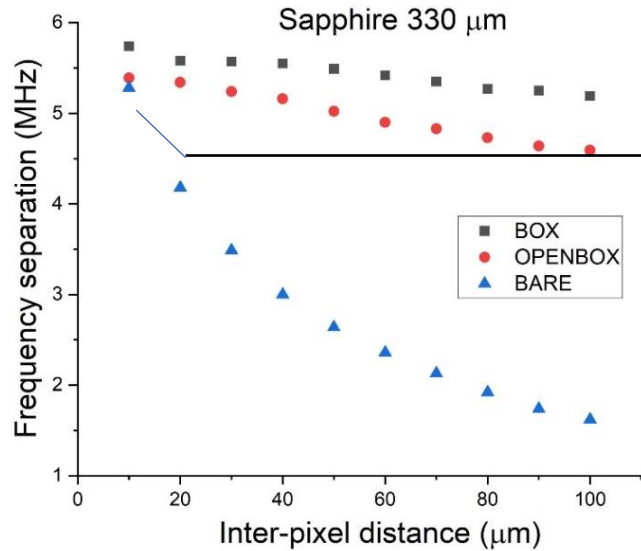


Substrate	Thickness	ϵ_r	$\tan(\delta)$	L_k (NbN)	ν (range)	$\Delta\nu$
Sapphire	330 μm	10.06	$5 \cdot 10^{-6}$	7.28 pH/□	1.35 – 1.9 GHz	36.7 MHz
Diamond	700 μm	5.67	10^{-5}	22.77 pH/□	1.25 – 1.75 GHz	33.3 MHz

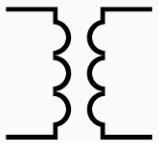
- Diamond lower ϵ_r , compensated by thicker substrate → similar frequencies
- 3 different combination of equally spaced resonances selected for production



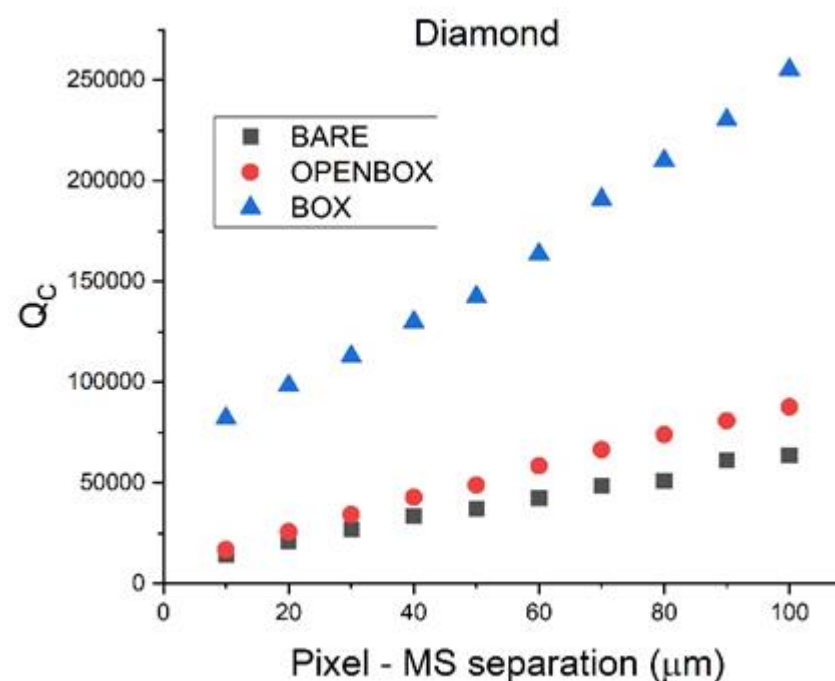
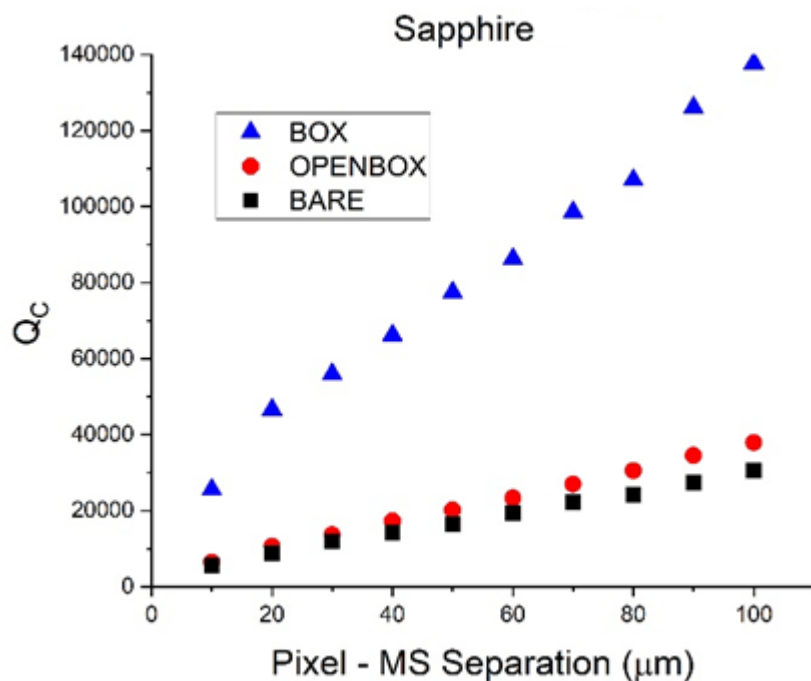
Cross-talk analysis



- Larger separation = stronger cross-talk
- Diamond: lower $\epsilon_r \rightarrow$ smaller cross-talk
- Shielding structures *increase* the cross-talk (contrary to literature reports)
- Tilted 2x2 configuration \rightarrow ungrounded shields = coupling booster

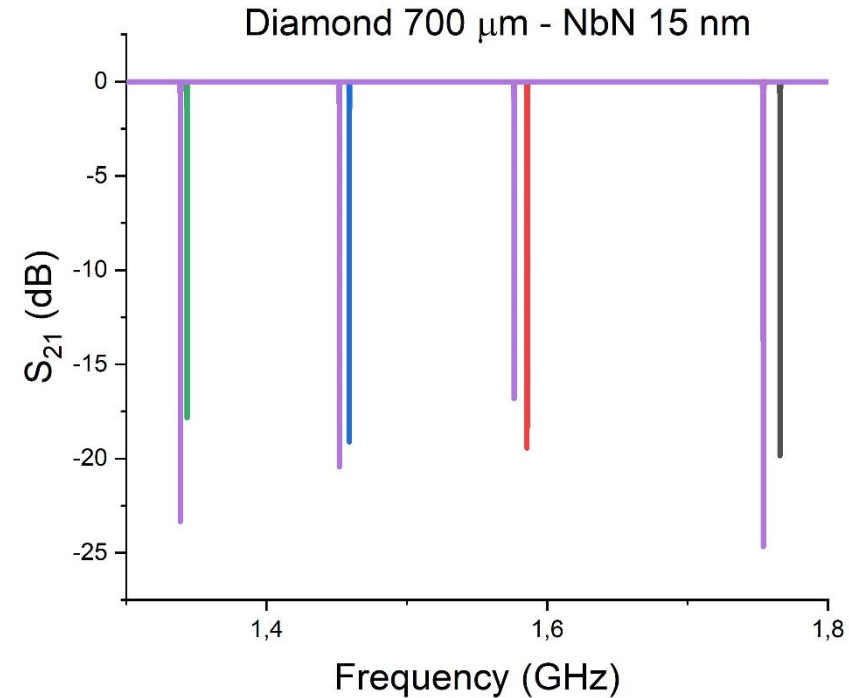
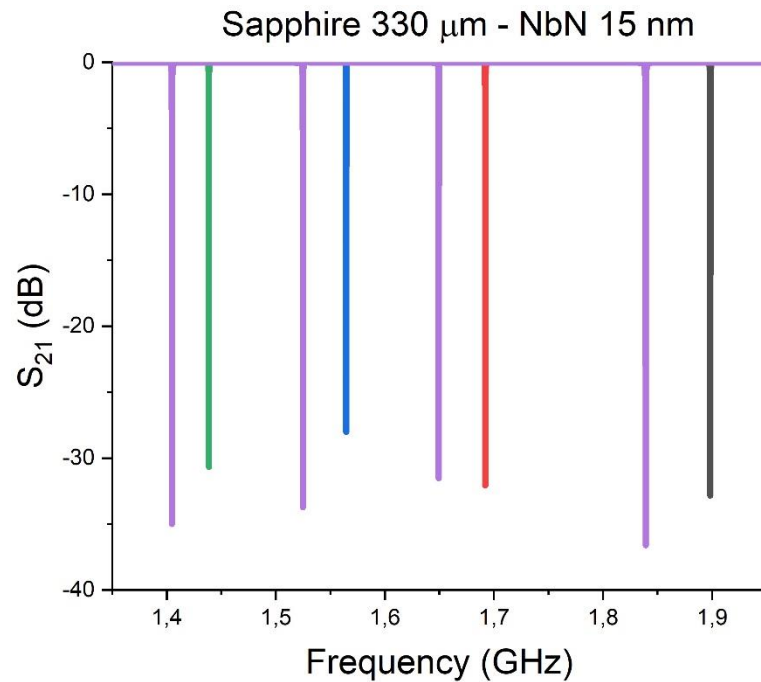


Q_C Evaluation



- Power intensive application \rightarrow prioritize stronger coupling (low Q_C) \rightarrow deeper resonances, no saturation
- Expected Q_0 under illumination $\approx 10^3$ - 10^4 , optimum coupling, max resp: $Q_C = Q_0 = 2Q_L$
- For the same configuration, diamond present lower coupling compared to sapphire
- Fully enclosed shielding pushes the Q_C toward values in the 10^5 range

Complete Detector Simulations



No cross-talk: adjacent pixels sufficiently separated in frequency



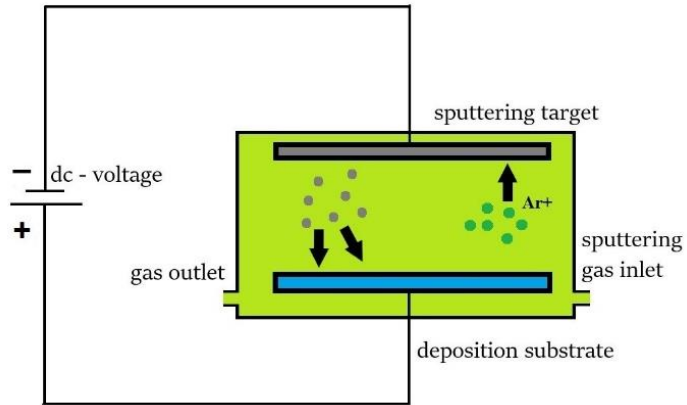
Additional capacitive effects \rightarrow Detector resonance frequency (purple) lower than single pixels (colored) \rightarrow low ϵ_r of diamond greatly limits this effect



Final design: No shielding, 3 Prototypes



Film deposition and characterization



DC reactive magnetron sputtering

- Sub. temp: 850°(Sa), 750° (Di)

- Deposition rate 0.065 nm/s

Photolithography and RIE

- Positive tone UV Photolith.
- RIE: 30 sccm SF₆ + 6 sccm O₂



Thin film Characterization

- Multi channel DC dipstick, 4 terminal probing

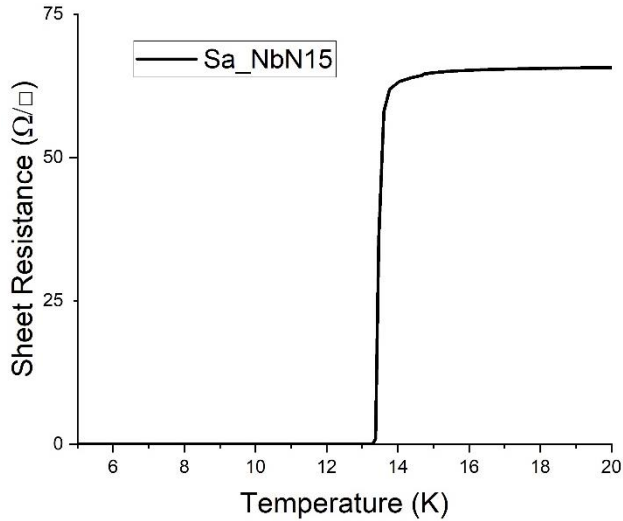
$$\Delta(T) \approx 1.74 \cdot \Delta(0) \left[1 - \frac{T}{T_c} \right]^{\frac{1}{2}}$$

- R/T curve →

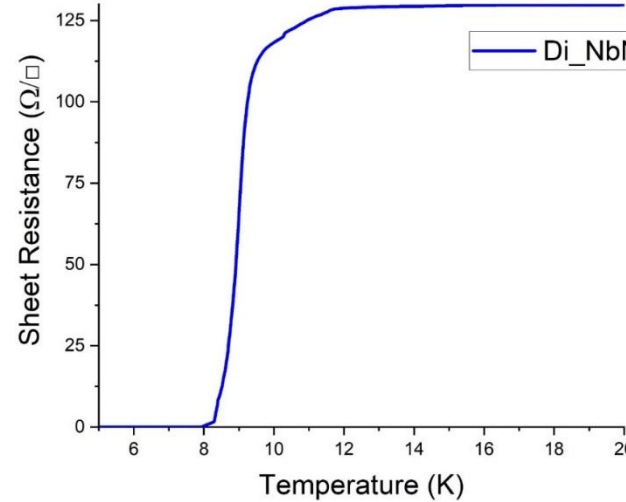
$$\Delta(0) = 1.76 \cdot k_B T_c \rightarrow L_k = \frac{\hbar R_s}{\pi \Delta}$$



SC thin film DC characterization



- $a = 4.785 \text{ \AA}$
- Mismatch: 0.4 \AA
- High T_C uniform, sharp transition \rightarrow high quality

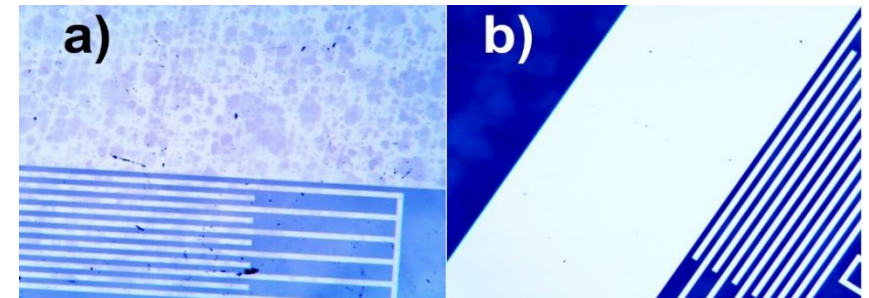


- $a = 3.567 \text{ \AA}$
- Mismatch: -0.81 \AA
- Low T_C , wide ΔT_C , multi step, high $R_{\square} \rightarrow$ lower quality

Sub-Film.	T_C (K)	ΔT_C (K)	R_s (Ω_{\square})	Δ (meV)	L_k (pH/ \square)
Sa-NbN15	13.4	0.18	70.6	2.03	7.28

Sub-Film.	T_C (K)	ΔT_C (K)	R_s (Ω_{\square})	Δ (meV)	L_k (pH/ \square)
Di-NbN15	9.05	1.24	156.53	1.44	22.77

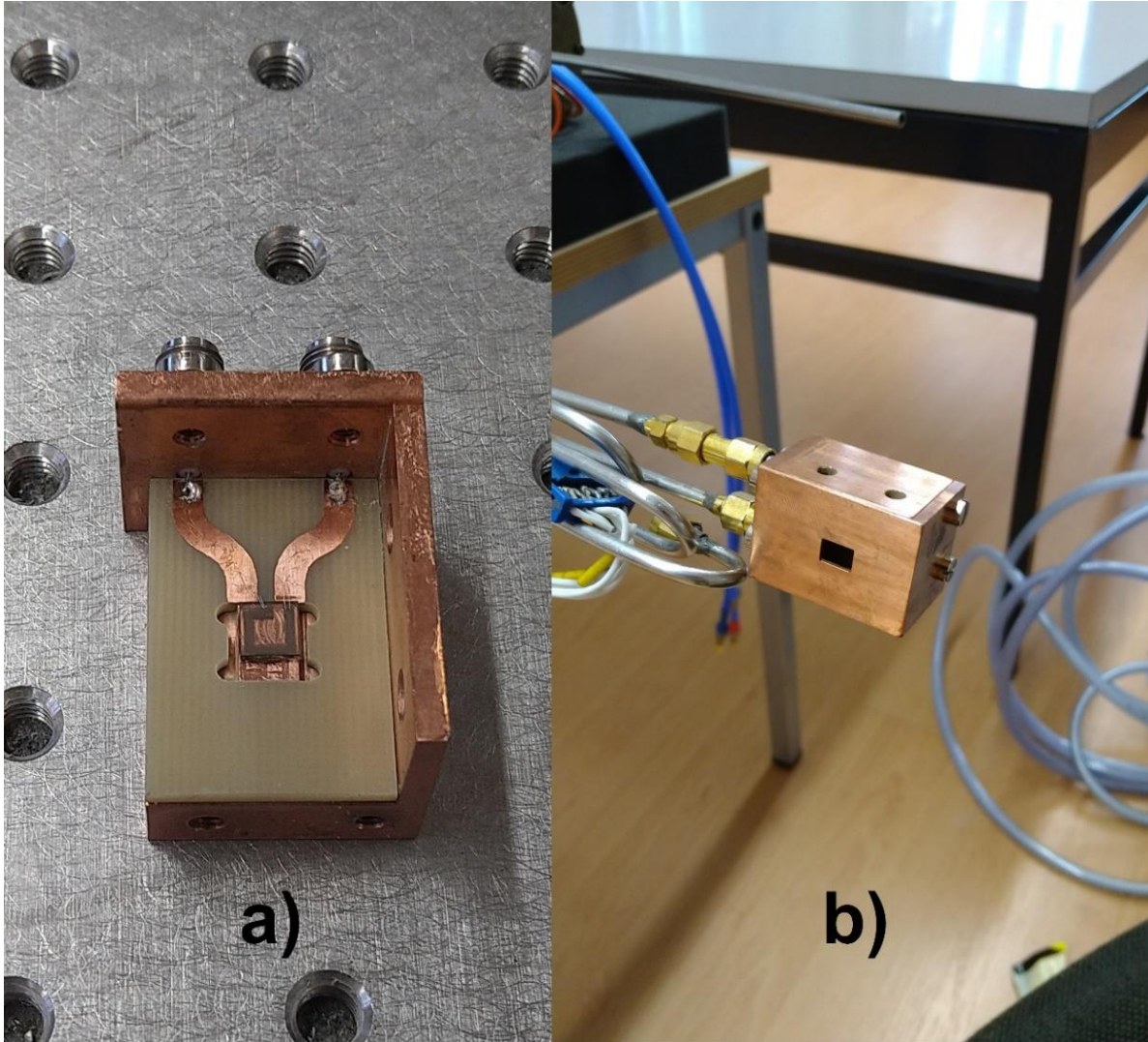
- Lattice and CTE mismatch between Di and NbN lead to low quality SC films
- Film degradation apparent under optical microscope, no flaking
- High L_K value for the film on diamond



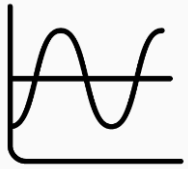
Diamond

Sapphire

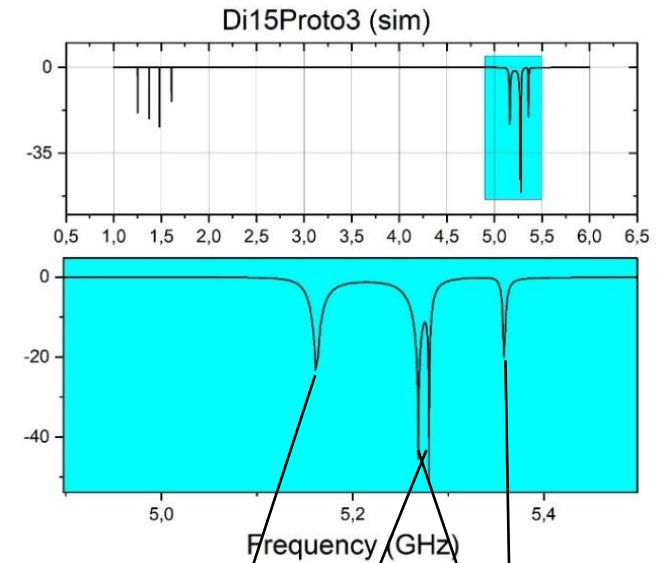
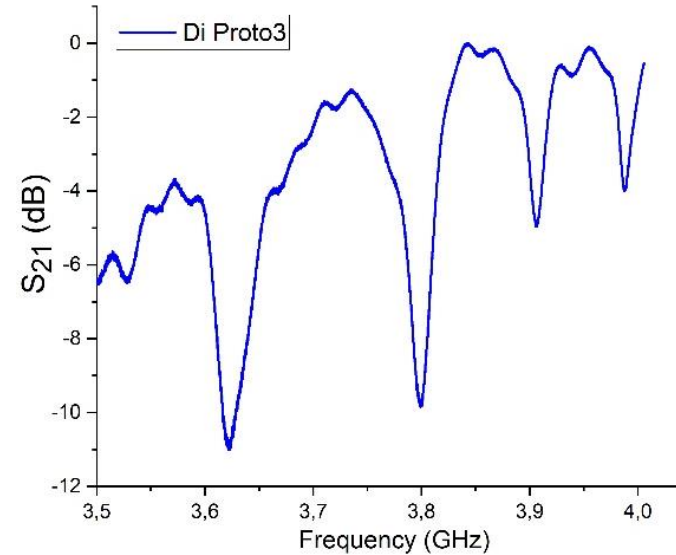
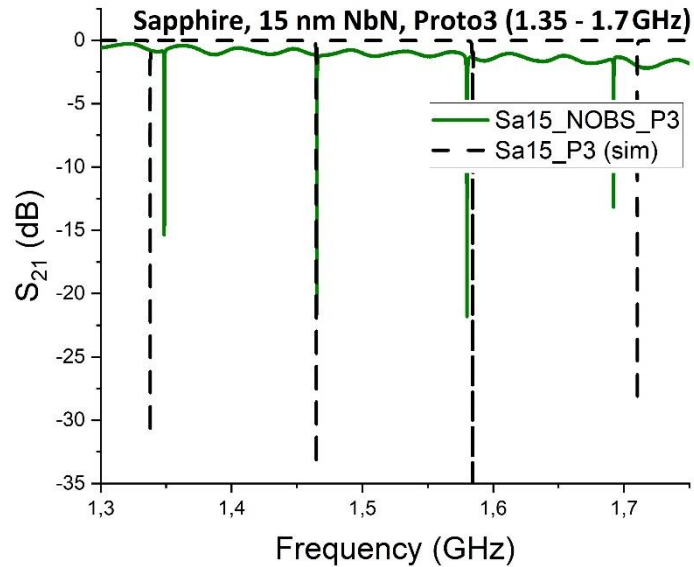
Microwave characterization setup



- OFHC Cu detector box + dipstick
- Direct connection with VNA
- -20 dBm injected power + 20 dB attenuator at port1
- Transition to SC marked by $-60 \rightarrow -40$ dB baseline jump

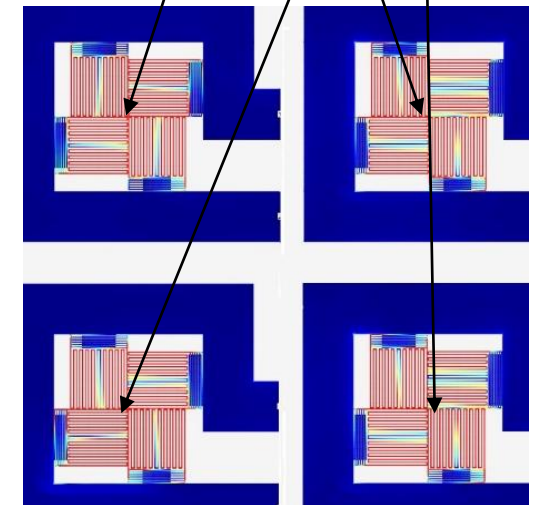


Resonance measurements

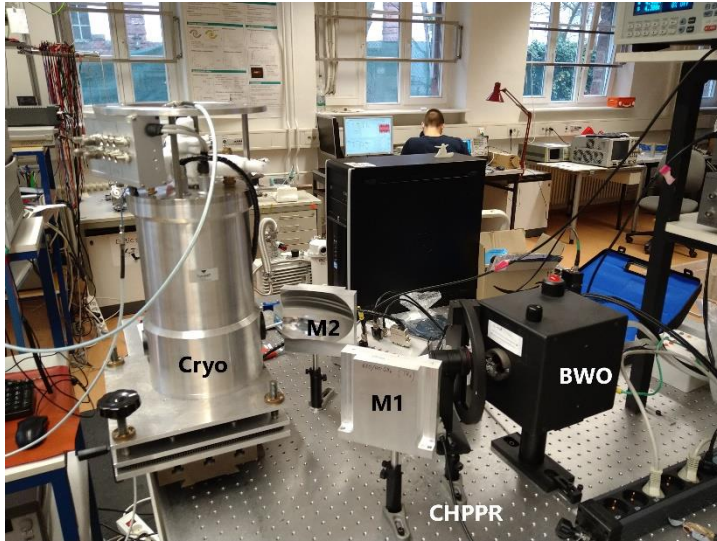


- Deep and sharp resonances
- Measured $Q_C \approx 3-5 * 10^3 \sim 2/3$ Simulated Q_C
- Simulations \approx measurements ($\Delta L_K \sim 10-15\%$)

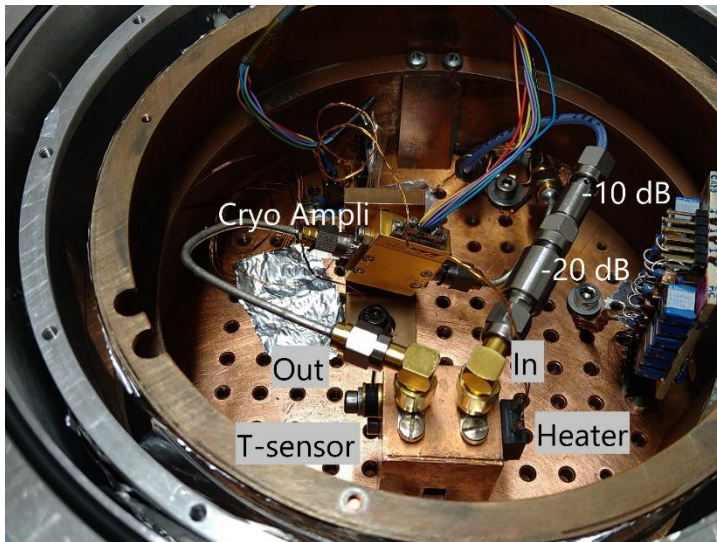
- Simulations: first order (1.25 - 1.55 GHz) + higher order (5.15 - 5.35 GHz)
- Measured: higher order, collective resonances only (3.6 \rightarrow 4 GHz), $Q \sim 10^2$
- High resonator internal losses



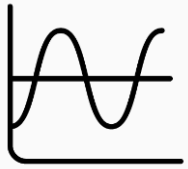
Detector THz response: measurement setup



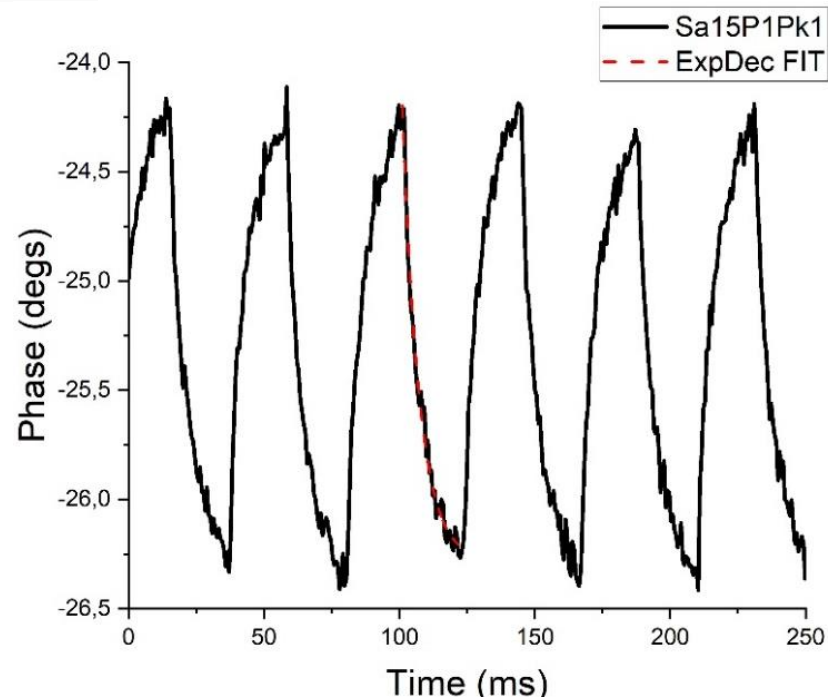
- BWO source @ 900 GHz, 20 μ W on detector
- 23 Hz chopper
- LHe bath cryostat



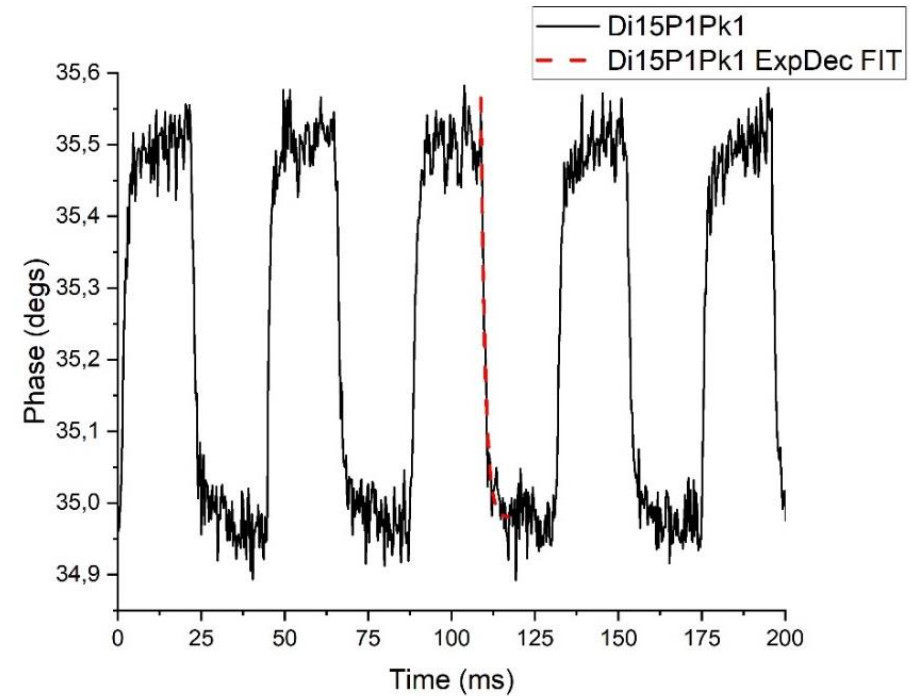
- Detector box with temp. sensor and heater
- 27 dB gain, 1-6 GHz bw cryo amplifier + external RT amplifier
- -10 and -20 dB attenuators



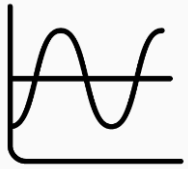
Detector THz response



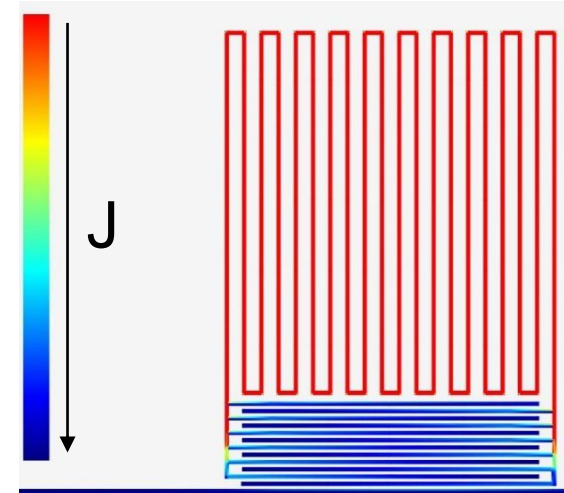
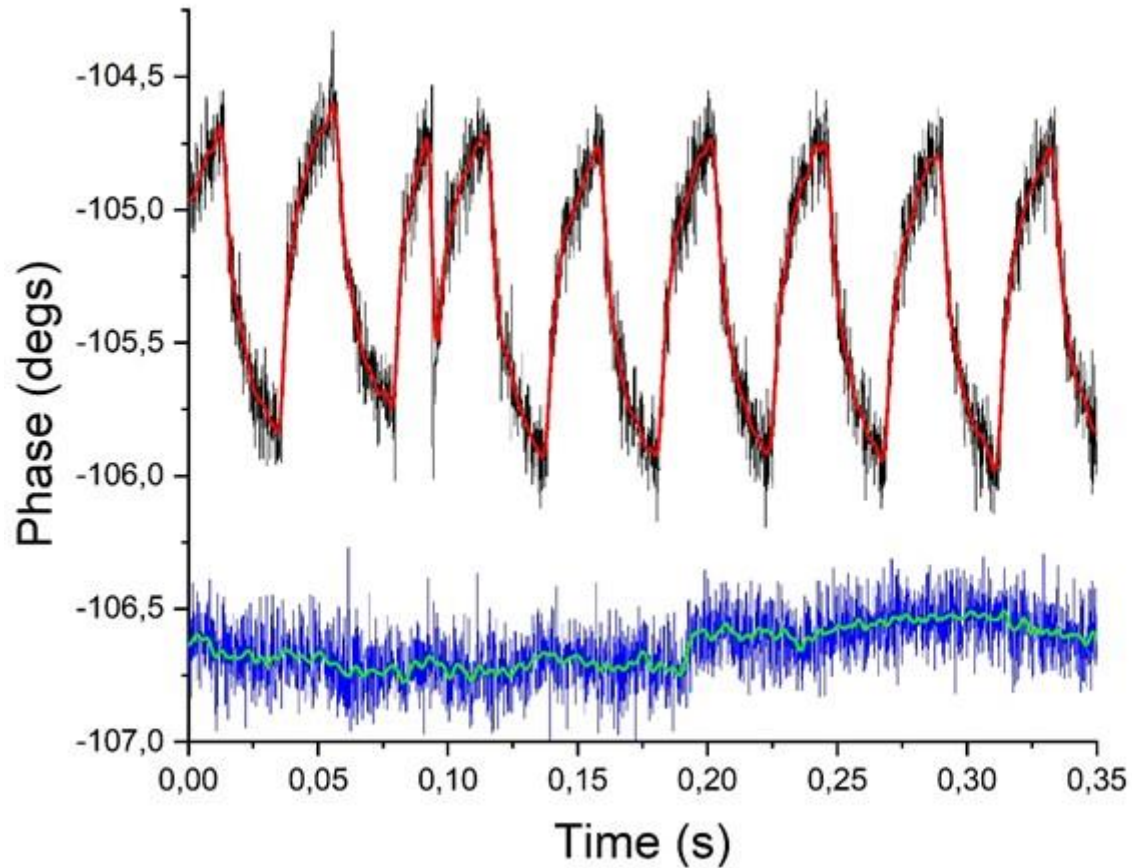
- Measurement Temperature: 4.5 K
- $\tau_{Sa} = 6 \text{ ms} \rightarrow$ bolometric response
- Responsivity: 0.1 deg./ μW , noise avg. $\sim 3 \cdot 10^{-8} \text{ W}/\sqrt{\text{Hz}}$ (high T + RT ampli)



- Measurement Temperature: 5.5 K
- $\tau_{Di} = 1 \text{ ms} \rightarrow$ bolometric response, faster dynamics
- Responsivity: 0.03 deg./ μW , noise avg. $\sim 3 \cdot 10^{-8} \text{ W}/\sqrt{\text{Hz}}$ (high T + RT ampli)



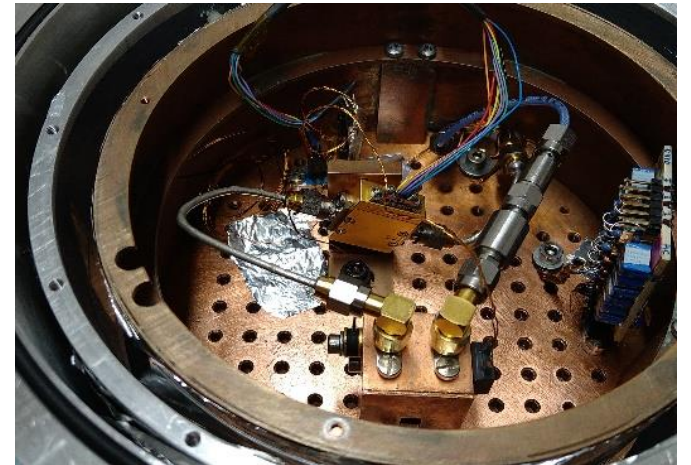
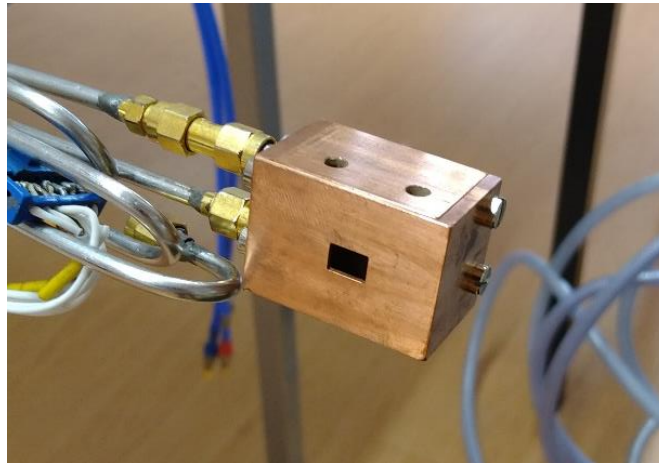
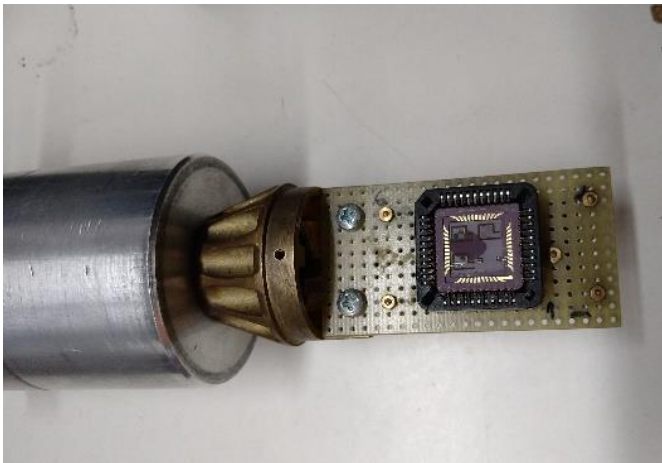
Cross polarization rejection



- Polarization selector: waveplate + linear polarizer
- Capacitor contribution negligible (low J)
- **Detector design effective in rejecting cross-polarization**

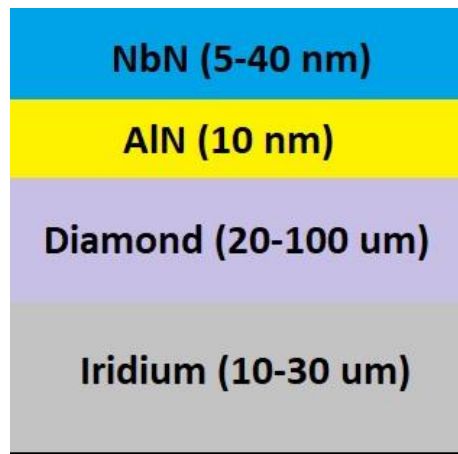
Summary

- Polarization sensitive KIDs based on diamond substrates have been studied and developed for fusion plasma applications
- Design finalized through simulation study for multiplexing, cross talking and coupling strength
- NbN films on diamond suffer from low quality, caused by the lattice and CTE mismatch
- Bolometric response, $\tau_{\text{Di}} < \tau_{\text{Sa}}$
- First investigation of diamond as substrate for LTSC detectors → first step toward the optimization of this technology
- Next steps: film quality improvement with AlN buffer layer, single crystalline diamond epitaxial growth on iridium



BACKUP

Future outlook - Heteroepitaxial KID

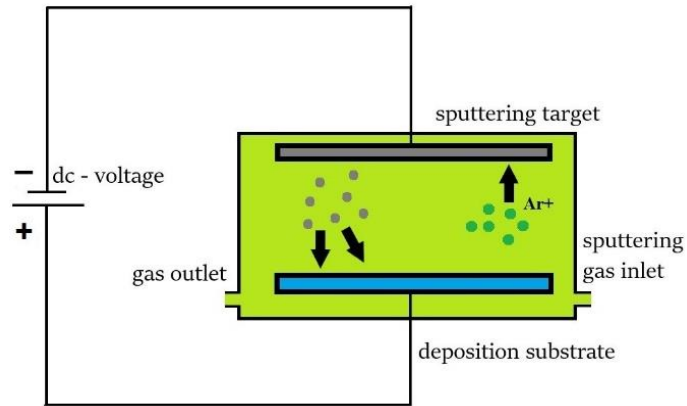


- Iridium is the best substrate for growth of single crystal diamond
- It has the ability to promote single crystal growth even when used as interlayer with a very rough main substrate, with large lattice mismatch.
- The diamond films grown on Ir have highly improved alignment and extremely low mosaicity → above a certain thickness, the films are single cryst.
- MBE and PLD deposition techniques
- Extreme corrosion resistance
- Iridium could additionally work as backshort / ground plane for a microstrip based KID
- AlN buffer layer for increased film quality

- Appl. Phys. Lett. **78**, 192 (2001)
- Appl. Phys. Lett. **91**, 061501 (2007);



Film deposition and characterization



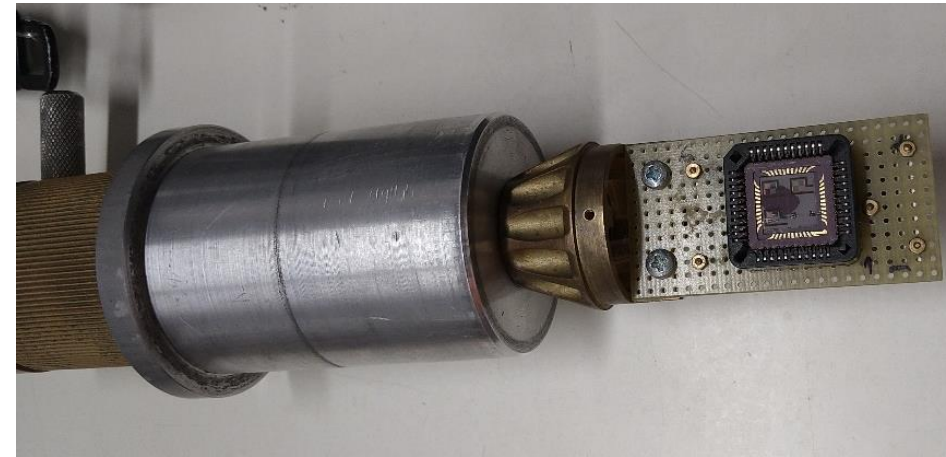
DC reactive magnetron sputtering

- Sub. temp: 850°(Sa), 750° (Di)

- Deposition rate 0.065 nm/s

Photolithography and RIE

- Positive tone UV Photolith.
- RIE: 30 sccm SF₆ + 6 sccm O₂



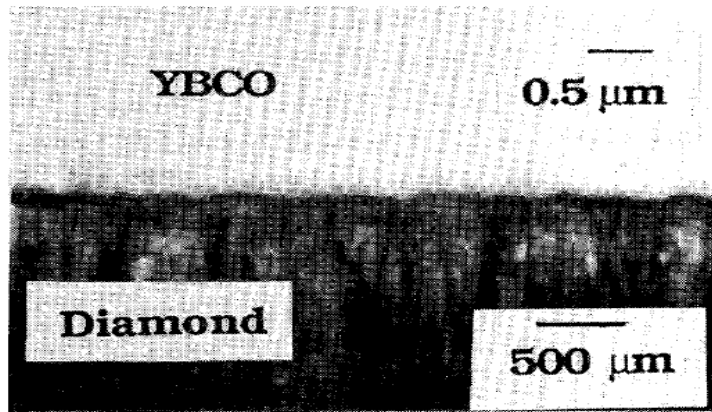
Thin film Characterization

- Multi channel DC dipstick, 4 terminal probing

• R/T curve →

$$\Delta(T) \approx 1.74 \cdot \Delta(0) \left[1 - \frac{T}{T_C} \right]^{\frac{1}{2}}$$
$$\Delta(0) = 1.76 \cdot k_B T_C \quad \rightarrow \quad L_k = \frac{\hbar R_s}{\pi \Delta}$$

YBCO Deposition on diamond



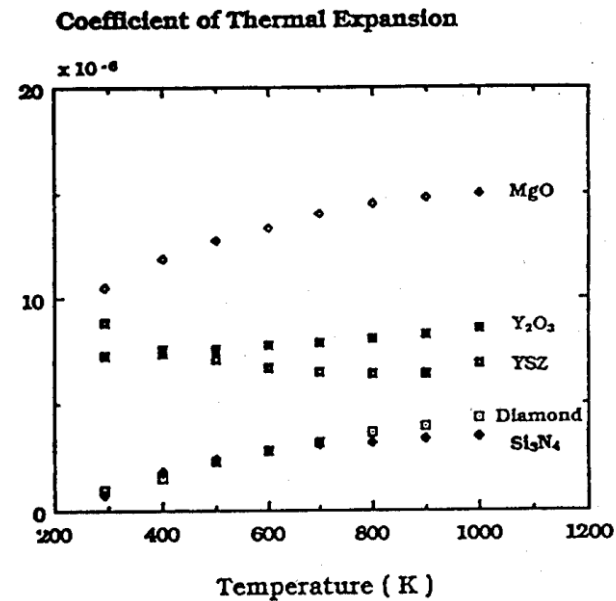
YBCO can be directly grown on poly-Di, but

- Very low quality film ($T_C = 17K$)
- Very difficult patterning
- No sharp separation of layers of SC and sub.

YBCO deposition is an O_2 dependent process → decomposition of Di

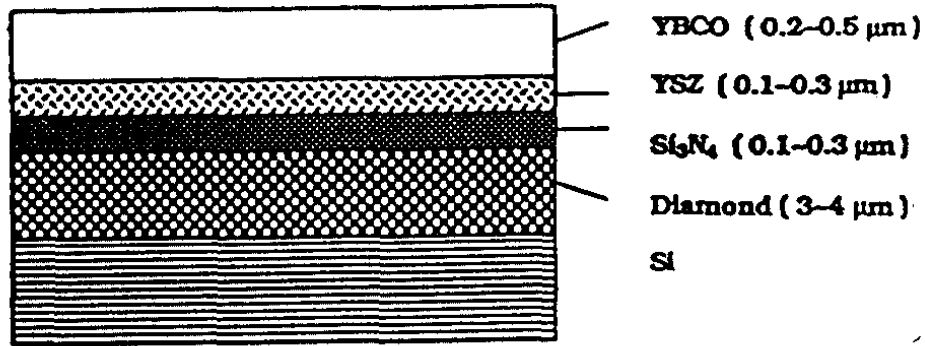
→ Si_3N_4 / YSZ Buffer layer system

- Coefficient of thermal expansion (CTE) and lattice matching → avoid film cool down cracking
- Yttrium Stabilized Zirconia: growth template for Di nucleation
- Si_3N_4 acts as a barrier for the diffusion of oxygen and carbon, protects Di during YBCO depo.

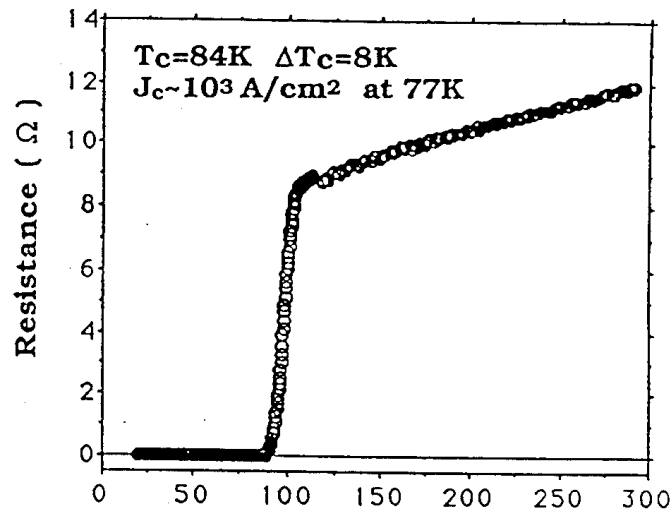


"M. Ece et al. / Superconducting films on diamond by PLD"

YBCO Deposition on diamond - state of the art films

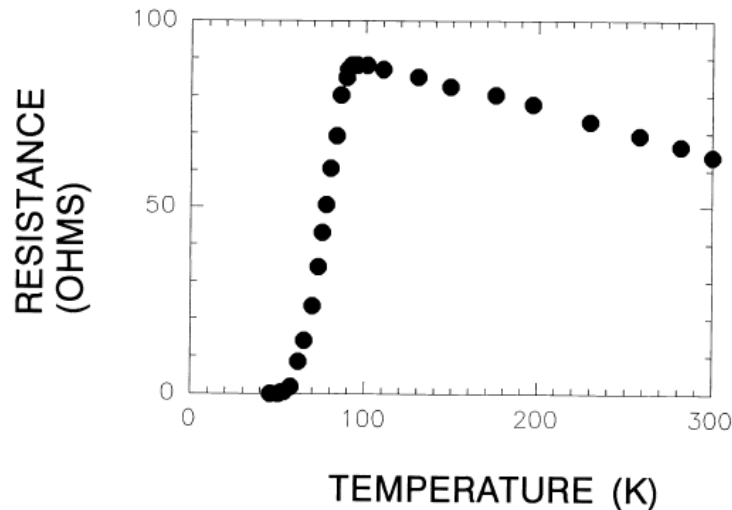
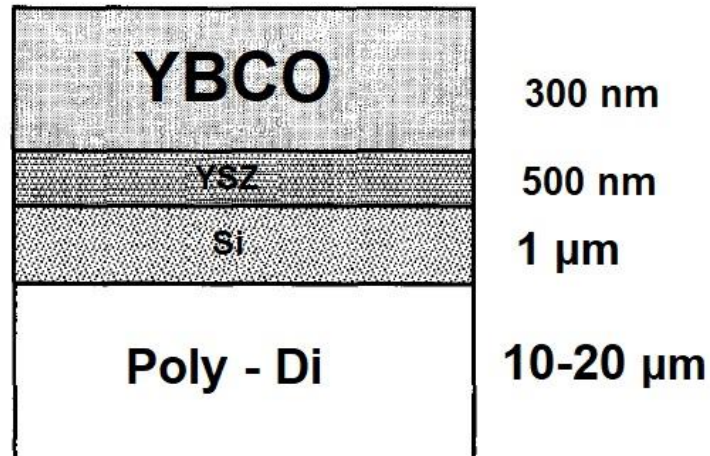


- 2-4 μm thick nanocrystalline (< 0.2 μm) diamond films, 500-1000 Å roughness, over Si wafers.
- Pre-annealing of diamond films at 680-800 C for 1-3 hours in high vacuum (2.6×10^{-6} Torr) – outgassing of H from diamond
- 0.1 – 0.3 μm Si₃N₄ deposited by RF-magnetron in 10-20 Torr of Ar @ 680-750 C
- 0.1 – 0.3 μm YSZ deposited by reactive RF-magnetron @ 680-750 C, in 10-30 Torr of mixed Ar/O₂
- YBCO grown in-situ, off-axis reactive RF magnetron @ 660-750 C, in 200-300 mTorr of mixed Ar/O₂
- In situ post deposition annealing, with O₂ venting (600-760 Torr), 30 min @ 500 C
- $T_C = 84$ K, $\Delta T_C = 8$ K, $J_C \sim 10^3$ A/cm²



“IEEE TRANSACTIONS ON APPLIED SUPERCONDUCTIVITY, VOL. 3, NO.1, MARCH 1993”

YBCO Deposition on diamond – lower quality example



- Si/YSZ buffer layer (polycrystalline)
- Plasma CVD deposition of 1 μm amorphous Si layer, vacuum annealing @ 700 C for crystallization
- YSZ and YBCO deposited by laser ablation, 650 C sub. temp., 200 mTorr O₂
- T_{C-zero} = 50 K, onset = 87 K → 25 K wide transition, low responsivity
- 200x200 μm² bolometric micro-bridge, direct laser writing
- 18 V/W responsivity, 6.3x10⁻⁹ W^{1/2}/Hz NEP

"Superconducting YBa₂Cu₃O₇ bolometer on polycrystalline diamond," Proc. SPIE 2159, High-Temperature Superconducting Detectors: Bolometric and Nonbolometric, (20 May 1994)"