

A U-Net-based self-stitching method for generating periodic grain structures

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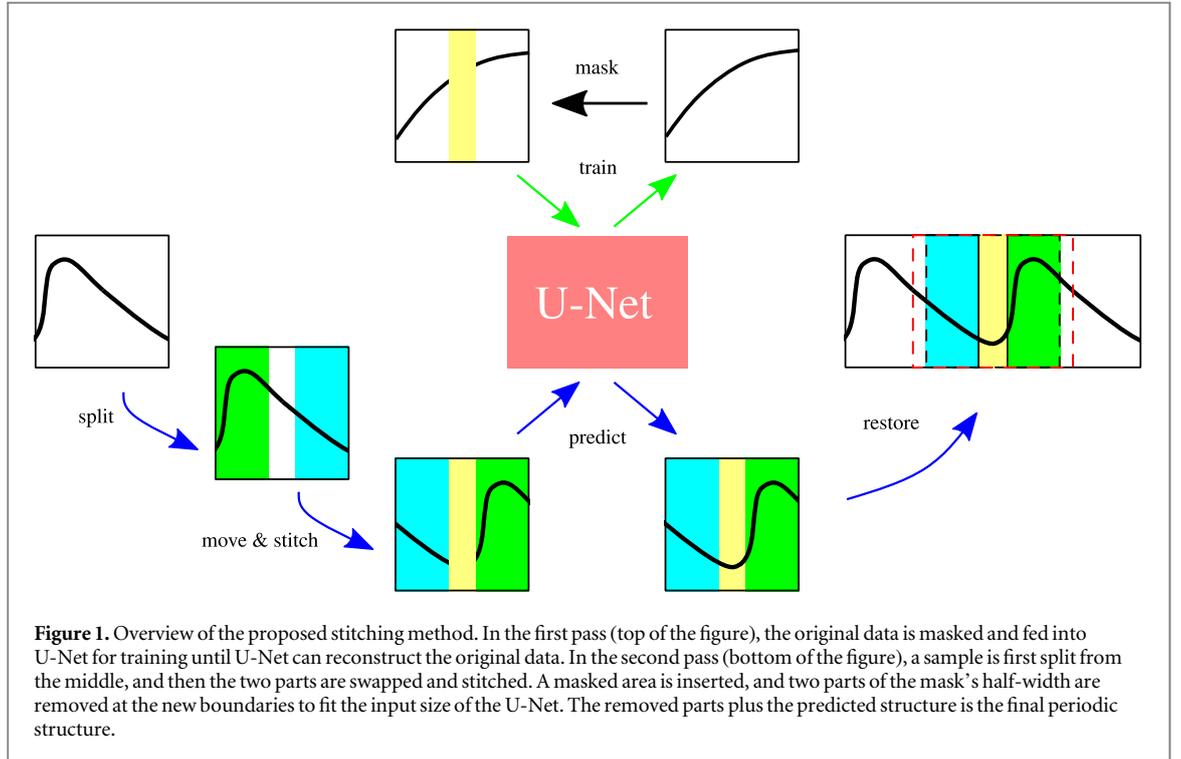
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Abstract

When modeling microstructures, the computational resource requirements increase rapidly as the simulation domain becomes larger. As a result, simulating a small representative fraction under periodic boundary conditions is often a necessary simplification. However, the truncated structures leave nonphysical boundaries, which are detrimental to numerical modeling. Here, we propose a self-stitching algorithm for generating periodic structures, demonstrated in a grain structure. The main idea of our algorithm is to artificially add structural information between mismatched boundary pairs, using the hierarchical spatial predictions of the U-Net. The model is trained with 20,000 grain sample pairs simulated from multiphase field simulations, resulting in the successful generation of periodic grain structures as expected. Furthermore, we employ an energy-based metric, the local curvature, to highlight the quality of the generated samples. Through this metric, we determine that the optimum value of the width of the mask is 1/16 of the sample width. The algorithm provides an automatic and unbiased way to obtain periodic boundaries in grain structures and can be applied to porous microstructures in a similar way.

1. Introduction

Computational simulation, an economical and powerful research paradigm, plays an important role in explaining experiments, gaining insights into fundamental physical mechanisms towards designing materials and components. In materials science, one of the essential tasks for computational simulation is to reveal the structure-property linkage through homogenization approaches considering that full resolution models are extremely expensive to compute. In this regard, setting a representative geometry that is large enough to reflect the macroscopic properties is the first step. A larger simulation domain usually contains more information about the system, but at the same time involves higher computational costs that, in a power law [1], are scaled with the size of the computational domain, which rapidly becomes unaffordable again. Therefore, a representative volume element with periodic boundary conditions is a practical solution [2]. However, real microstructures do not always ensure periodicity, except for few cases, such as crystal structures. In particular, when extracting a representative element from reference experimental data [3–5], such as computer tomography images, the representative volume element is typically cropped from a larger network. This cropping process often results in the truncation of boundaries, introducing non-periodic features into the extracted element. This leads to an inconsistency between the corresponding boundaries and produces additional nonphysical effects. For example, the Pores Per Inch (PPI) of a metal foam is a widely used parameter in industrial applications to characterize porous materials. However, the pores on the boundaries are ignored



when determining the PPI [6]. Therefore, an approach that allows for the identification of boundary pores in a periodic structure will result in a more accurate PPI calculation.

There are many existing methods to generate periodic structures from scratch [7]. Among those, Voronoi-based method, metaheuristic approaches, and models built on physical laws are extensively discussed [8–10]. However, the generation of periodic boundaries comparable with reference structures is less visited. Mathematically, converting a non-periodic structure into a periodic structure is equivalent to interpolating the mismatching boundaries by a zone that preserves the intrinsic characteristics of the larger system. The desired interpolation should go beyond the direct truncation, the air layer [11], and the polynomial interpolation [12, 13]. Previous works try to rebuild microstructures based on geometrical rules [8] or by using meta-heuristic methods [9], requiring additional assumption of the properties of microstructure rather than depending on the intrinsic information contained by the sample them self. In comparison to traditional approaches, neural networks are successful and powerful tools that meet such requirements. The neural network is attracting considerable attention, as it has been shown to be a universal mapping tool that approximates any function [14]. This ability allows the extraction of the characteristic features of data, as well as the generation of a new structure based on the combination of learned features [15, 16]. Here, the interpolation between two boundaries can be seen as a problem of modifying existing structures. Therefore, it is promising to apply data-driven solution schemes to the current problem.

In this work, we propose a U-Net based self-stitching algorithm to generate a periodic grain structure from a non-periodic grain structure. The U-Net is popular for its encoder-decoder architecture [17] and its hierarchical structure which captures the characteristics of data at different scales [18]. U-Nets can achieve acceptable performance with less volumes of data. The validity of U-Net is verified in various applications, including the segmentation of 3D tomography images [19, 20] and the determination of the thickness of 2D van der Waals heterostructures from optical microscopy images [21]. As shown in figure 1, the algorithm consists of a training phase and a prediction phase. In the training phase, the U-Net is forced to reconstruct the masked structure, while in the prediction phase, the boundaries to be joined are first arranged in an individual structure and then connected by U-Net. A periodic version of the corresponding non-periodic structure can be obtained by a further step of simple manipulation. The proposed method is examined in one representative direction of a 2D grain structure. The U-Net model is established and trained using 20,000 physically plausible grain structure pairs, achieving a prediction accuracy of over 90%. Subsequently, periodic structures are successfully obtained. Furthermore, we utilize a quantitative metric, the local curvature, to assess the quality of the generated structures. Notably, the mask width, a crucial parameter in the algorithm, is found to have an optimum value of 1/16 of the sample width. Our results prove that the proposed algorithm is able to generate periodic boundaries in grain samples and paves the way for similar problems in a wide variety of systems.

2. Methods

2.1. The multi-phase field method for grain generation

The phase field method is used to computationally generate a large number of grain samples. Following Nestler *et al* [10], a general multiphase, multicomponent phase-field model can be established on the free energy description:

$$F(T, c, \phi) = \int_{\Omega} (f(T, c, \phi) + (\epsilon a(\phi, \nabla\phi) + \frac{1}{\epsilon}w(\phi))) dx \quad (1)$$

which states that the total free energy is an integral of the local free energy density, contributed by the bulk energy f , depending on temperature T , the concentration c , and the phase variable $\phi = \phi_{\alpha}, \alpha = 1, 2, 3, \dots, N$. Meanwhile, the interfacial free energy, on the other hand, can be formulated as the gradient-dependent term $\epsilon a(\phi, \nabla\phi)$ plus a potential well term $\frac{1}{\epsilon}w(\phi)$, where ϵ is a measure of the thickness of the diffusive boundary layers.

In this study, constant temperature is assumed and the energy difference between different phases is not considered, so the bulk energy f disappears at all. Equation (1) could be reduced to

$$F(\phi) = \int_{\Omega} (\epsilon a(\phi, \nabla\phi) + \frac{1}{\epsilon}w(\phi)) dx. \quad (2)$$

Through the Euler–Lagrange formula, the corresponding evolution equation for each phase ϕ_{α} can be derived as

$$\tau\epsilon \frac{\partial\phi_{\alpha}}{\partial t} = \epsilon [\nabla \cdot a_{,\nabla\phi_{\alpha}}(\phi, \nabla\phi) - a_{,\phi_{\alpha}}(\phi, \nabla\phi)] - \frac{1}{\epsilon}w_{,\phi_{\alpha}}(\phi) - \lambda \quad (3)$$

where the subscript commas denote partial derivatives. τ is a kinetic factor, which is set to a constant scalar 1. λ is a Lagrange multiplier that imposes the constraint $\sum_{\alpha=1}^N \phi_{\alpha} = 1$, which takes the form:

$$\lambda = \frac{1}{N} \sum_{\alpha=1}^N \left[\epsilon [\nabla \cdot a_{,\nabla\phi_{\alpha}}(\phi, \nabla\phi) - a_{,\phi_{\alpha}}(\phi, \nabla\phi)] - \frac{1}{\epsilon}w_{,\phi_{\alpha}}(\phi) \right]. \quad (4)$$

The definition of each energy function is given below. First, the gradient-dependent term is

$$a(\phi, \nabla\phi) = \sum_{\alpha<\beta} \gamma_{\alpha\beta} [a_{\alpha\beta}(q_{\alpha\beta})]^2 |q_{\alpha\beta}|^2 \quad (5)$$

where $q_{\alpha\beta} = \phi_{\alpha}\nabla\phi_{\beta} - \phi_{\beta}\nabla\phi_{\alpha}$, and $a_{\alpha\beta}$ is an anisotropic factor. For simplicity, anisotropy is neglected.

Then, the potential well term is assumed to be of multiobstacle type:

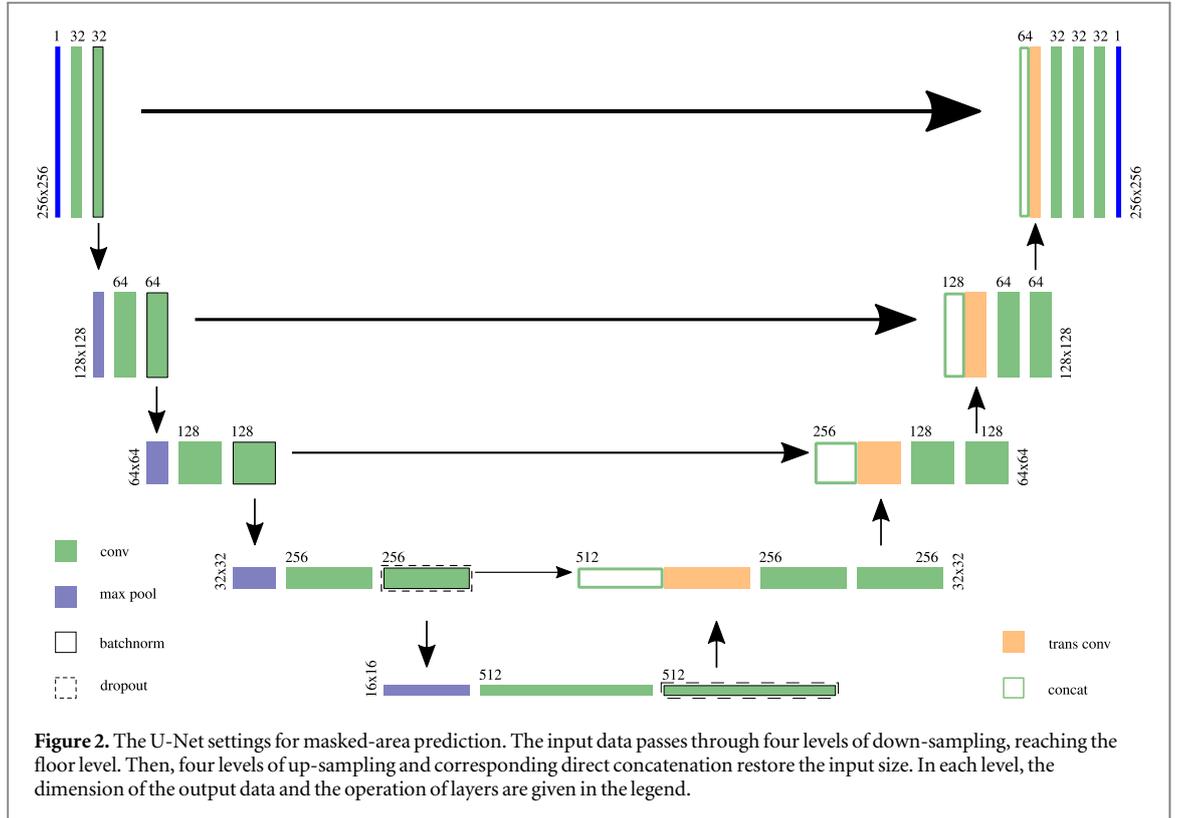
$$w(\phi) = \frac{16}{\pi^2} \sum_{\alpha<\beta} \gamma_{\alpha\beta} \phi_{\alpha} \phi_{\beta} + \sum_{\alpha<\beta} \gamma_{\alpha\beta\delta} \phi_{\alpha} \phi_{\beta} \phi_{\delta} \quad (6)$$

where $\gamma_{\alpha\beta}$ and $\gamma_{\alpha\beta\gamma}$ are coefficients.

The approach works in 2D and 3D and is capable to describe normal and abnormal grain growth for isotropic as well as anisotropic interface conditions and takes into account various physical influences such as heat treatment, mechanical load or diffusion. Grain growth processes are performed with the simulation framework PACE3D [22], which is developed and maintained in the research group of the IDM at the Karlsruhe University of Applied Sciences. The parameters incorporated in the simulation can be found in [23]. A simulation domain of 4000×4000 numerical cells is set up and 60 000 randomly distributed grains are initialized by a Voronoi-tessellation algorithm in a preprocessing step. After the simulation, 20 000 small patches are sampled from an intermediate time step. These grain samples are skeletonized for further data analysis applications. Results of phase-field simulations of grain structures serve as reference data sets for the developed U-Net approach.

2.2. The architecture of U-Net

The modules in U-Net are connected by a U-shape path, as shown in figure 2. U-Net is commonly used for the classification problem, but with slight modification, it could also solve our generative problem. Basically, we follow the architectural outline in the original U-Net paper, where two possible classes should be determined, namely the bulk phase and the boundary phase. In the output layer, the two classes are compressed into one channel. Along the ‘U’ path, the downsampling modules, where the dimension of data is reduced, come first, followed by up-sampling modules, where the data dimension is expanded. There are five levels in the downsampling and upsampling half-path, respectively, where the floor level is the connecting module. At each level, a bypass connection additionally starts from the downsampling module at each level, pointing to its upsampling counterpart. In each module, the solid blocks indicate the dimension of data after the manipulation specified by its color and edge style. As an example, we choose the first module, where the dimension of input



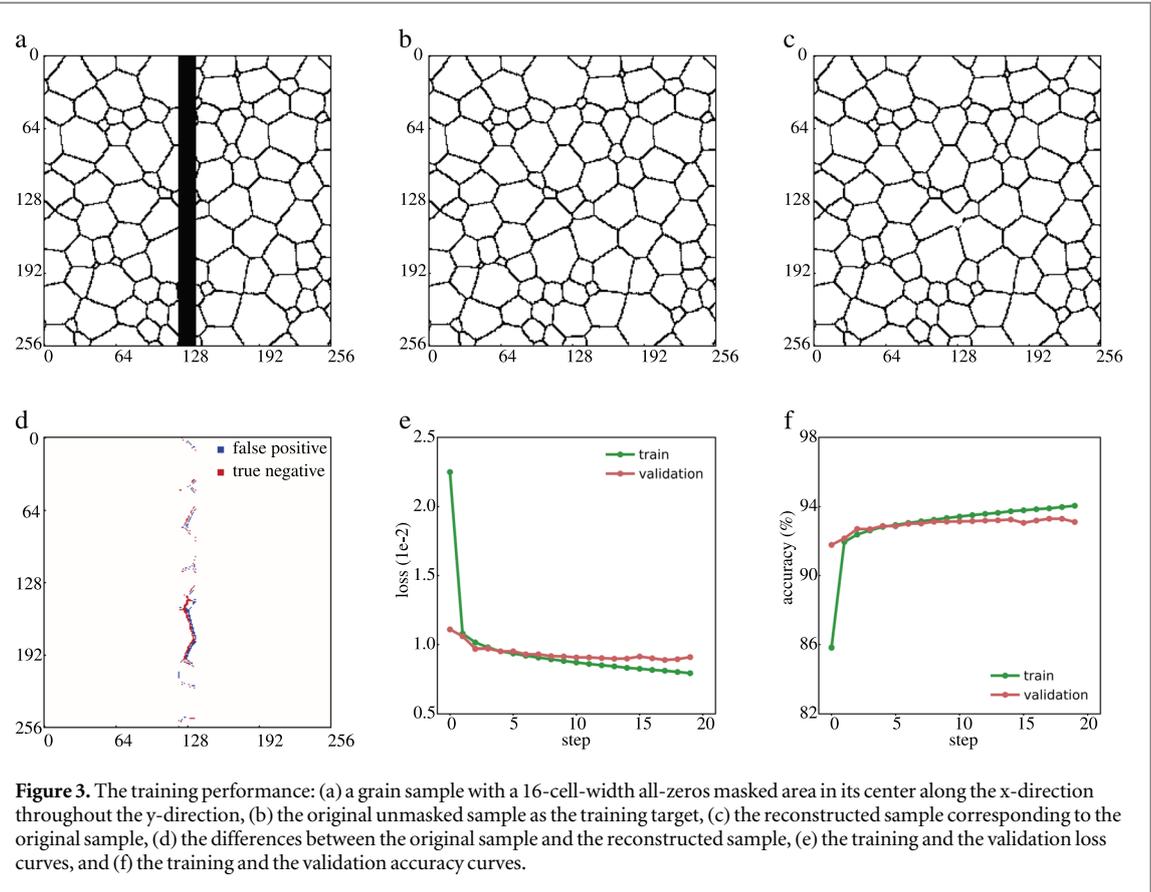
data is $256 \times 256 \times 1$, and a convolutional layer is applied. In this layer, the number of the filter is 32, with a filter size of 3. In the study performed here, a rectified leaky activation function is used. The padding strategy is to always maintain the same dimension, which results in a $256 \times 256 \times 32$ feature map. Then, the feature map is fed into the same convolutional layer once more, with unchanged dimension. Following this, a batch normalization layer is appended to regulate the feature map. Finally, the output is passed to a lower module and bypassed to the last module for subsequent use. In the following modules, the parameters for the convolutional layer, pooling layer, and transpose convolutional layer can be inferred from the dimension of the corresponding output data. When dropout is used, the dropout rate is 0.3. U-Net is implemented with TensorFlow [24], and Keras [25].

3. Results

3.1. The training of U-Net

In the proposed self-stitching method, the U-Net should be able to generate a reasonable structure between any two given boundaries. This ability will be acquired through training with artificially masked data. If a grain sample is intersected with two non-intersecting lines, the right boundary of the left part and the left boundary of the right part do not coincide, but they can always be connected by the inner part. In this investigation, the two boundaries are connected by the inserted structure. Based on this straightforward idea, we can manually mask the inner part and force the U-Net to reconstruct the masked structure on the basis of the unmasked structure. In principle, the shape of the boundary is not limited, but straight boundaries occur naturally in computational modeling. In addition, with a two-dimensional grain sample, it is convenient to use a rectangular mask. The long edges of the mask area are considered as the newly created boundaries. That means that the self-stitching is done only in the direction that is parallel to the short edge. We believe that once self-stitching is achieved in one direction, periodicity in two-directions could be achieved by following the same procedure in each direction. In addition, the mask is placed in the middle of the grain sample, for reasons of information symmetry.

As shown in figure 3(a), the black stripe means that all values are set to zero, which acts as the mask. The width of the mask is 16 cells in the horizontal direction and throughout the vertical direction. The masked sample is used as input to U-Net, and the original sample is the target for the output, as shown in figure 3(b). The masked area occupies a small part of the whole structure, ensuring enough structural information the rest area possesses is passed to the U-Net. The dataset contains 20 000 pairs of masked samples and their original counterparts. The dataset is divided into the training dataset, the validation dataset, and the test dataset. The split ratio is 0.6:0.2:0.2. Since data are categorical, we use sparse categorical cross entropy as the loss function. A batch



size of 8 is chosen along with the default Adam optimizer to train U-Net for 10 epochs on two GeForce GTX 1080 GPU nodes. After the training, the prediction for the masked area in figure 3(a) is shown in figure 3(c). Since the prediction is close to the ground truth, the difference between the two is shown in figure 3(d) for visual clarity. In figure 3(d), the colored pixels occur only within the masked area, which means that the rest of the structure is completely preserved. Here, the red pixel (true negative) means that this point should belong to a boundary, but the prediction varies from the ground truth; the blue pixel (false positive) refers to the fact that this point should not belong to a boundary, but the prediction is right. Despite the presence of falsely predicted values, it is observed that the red and blue pixels are usually correlated, which could be translated as the predicted boundaries are shifted. The boundary shift is tolerable, compared to randomness. Figure 3(e) shows the loss curve, where both the training loss and the validation loss decrease. The training loss is slightly less than the validation loss, which indicates that any overfitting is not severe. A modified spatial accuracy is also calculated and displayed in figure 3(f) based on the number of correctly predicted points versus total points within the masked area. Figure 3(f) shows that the trend in accuracy is consistent with the loss curve, and that the overall accuracy is over 90%, despite the boundary shifts. Therefore, it can be assumed that U-Net correctly scans the existing grain structures to infer and reproduce the missing structure from the given structural information.

3.2. Self-stitching of a nonperiodic structure

With this generative capability, we then investigate its feasibility in producing periodic boundaries. The process starts from a normal grain sample, illustrated in figure 4(a). The left half of the sample is indicated by a green box, while the right half of the sample is indicated by a blue box. The red dotted line marks the borders of the mask and is helpful to visually follow the manipulation. In figure 4(b), the left and right parts are swapped so that the boundary pair to be stitched is joined. Near the central line, the mismatch is visible. Then, a masked area is inserted in figure 4(c), which increases the total width of the sample. The width of the mask is identical to the one used in the training process. The sample is then cropped along the red lines on both sides. The remaining structure is shown in figure 4(d). The cropped size matches the shape of the input data of the U-Net, so that the information can be predicted within the mask, as shown in figure 4(e). Finally, the structure previously cut out can be restored. In figure 4(f), we see a periodic sample in the horizontal direction. The imperfection is that some grain boundaries are not fully closed in the stitched sample. However, the predicted structure outlines the possible structure where the line segment can be extended a little to close most of the grains. This issue will be

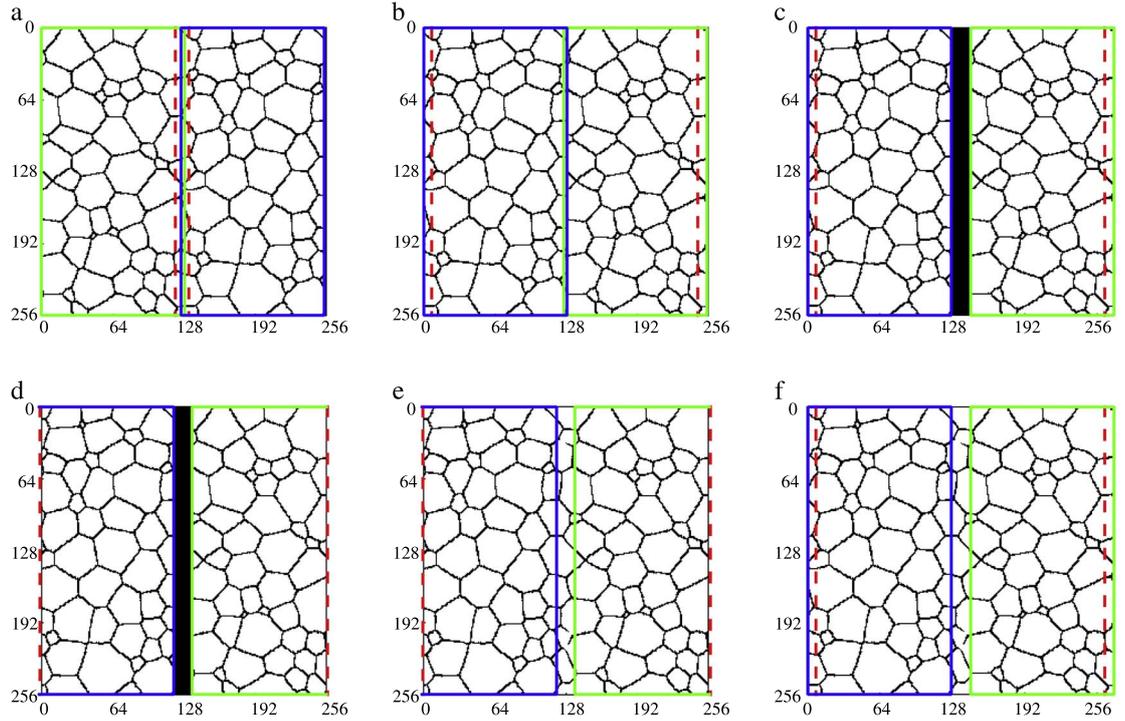


Figure 4. The prediction procedure: (a) splitting the original sample into two parts from the middle, (b) switching the left and right parts, (c) inserting an all-zeros area between the two switched parts, (d) cropping the data to fit the input size of U-Net, (e) the prediction from the U-Net, and (f) a periodic sample obtained after restoring the cropped area. For visual guidance, the green and blue frames show the left and right parts in the original sample, respectively. The red dotted line marks the area to be masked.

discussed in the next section. It is worth noting that in the masked area, not only the straight lines are connected, but also triple points are added, which demonstrates its power.

Since there is no ground truth that can be compared to the stitched structure, it is not easy to quantify the performance. On the one hand, the generated area should be somehow similar to the rest of the original sample, by definition. This is the desired result. On the other hand, the stitched structure should be created other than by just putting the edges together, without interpolation. This is the worst result, which should be surpassed by the method. Therefore, it is helpful to compare the original sample, the switched sample, and the restored sample. Using the results from figures 4(a), (b), and (f), we perform a watershed segmentation and show the results in figures 5(a)–(c), respectively. The watershed algorithm [26] defines the regions by recursively searching around the peaks in an elevation map, which could be obtained from a binary image, by a Euclidean distance transformation. Watershed segmentation captures the local feature of a structure, ensuring that the segmented structure is closed. In this way, the evaluation of boundaries is transformed into a comparison of the regularity of the shape of the grains. The segmentation results are not spatially comparable point-to-point, but it can be seen that only in figure 5(b), as highlighted in the magnification box, boundaries with sharp spikes are formed. These spikes are not seen in the original sample, so they are not desirable for a realistic grain sample.

To measure the difference, we introduce the local curvature [27], which is defined by counting the number of phases in the neighborhood of a point. Compared to global descriptors such as the grain size distribution function [28] and the two-point correlation function [29], the curvature metric offers a distinct advantage by capturing local information for each edge of the grain. This allows us to identify and analyze local irregularities of edges without them being suppressed by contributions from surrounding areas. The grain boundary maps illustrating the curvature distributions of figures 5(a)–(c) are given in figures 5(d)–(f), respectively. We note that the curvature distributions must be calculated based on the segmented map, as this is the only way to distinguish the grains. It can be observed that only the boundaries are non-zero, and the brightness of junctions is higher than that of the straight boundaries. These features are useful for detecting the irregular boundary shown magnified in the yellow box of figure 5(e). In figure 5(f), no irregular boundary is found, which confirms that the U-Net-assisted stitching gives better results than the direct concatenation of two boundaries.

3.3. Sensitivity analysis of the mask width

So far, we have shown that the U-Net can fill the mask area with proper training. The question arises whether the U-Net is hyperparameter sensitive. In other words, is there a requirement that ensures that U-Net provides

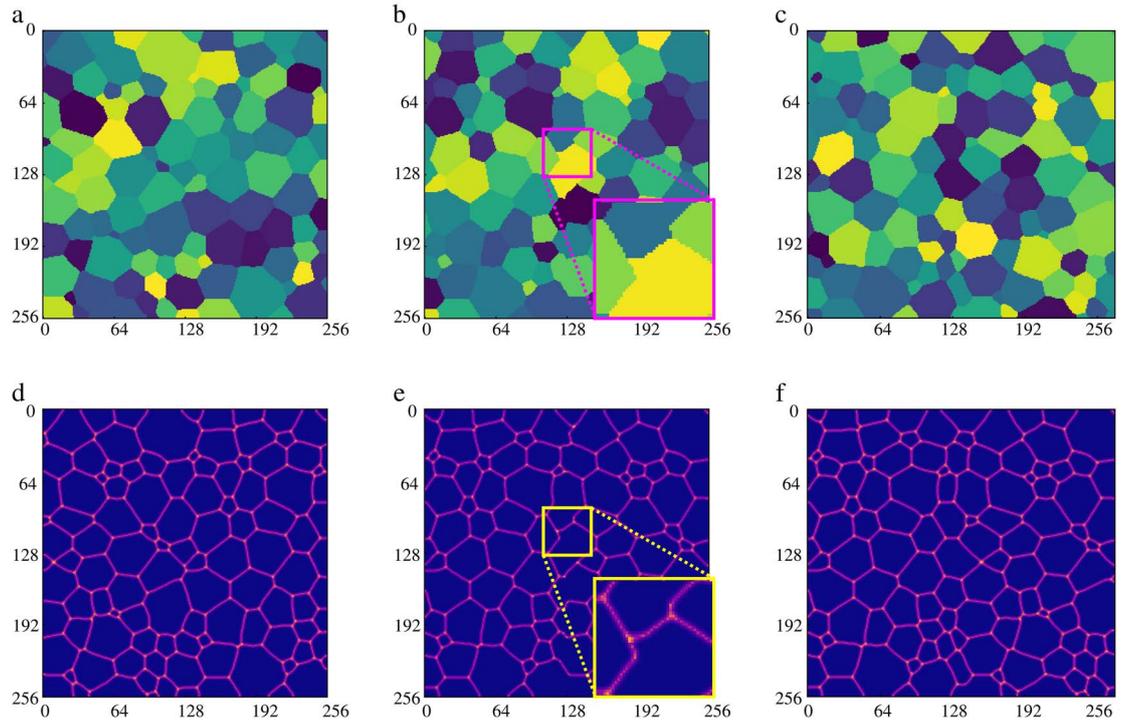


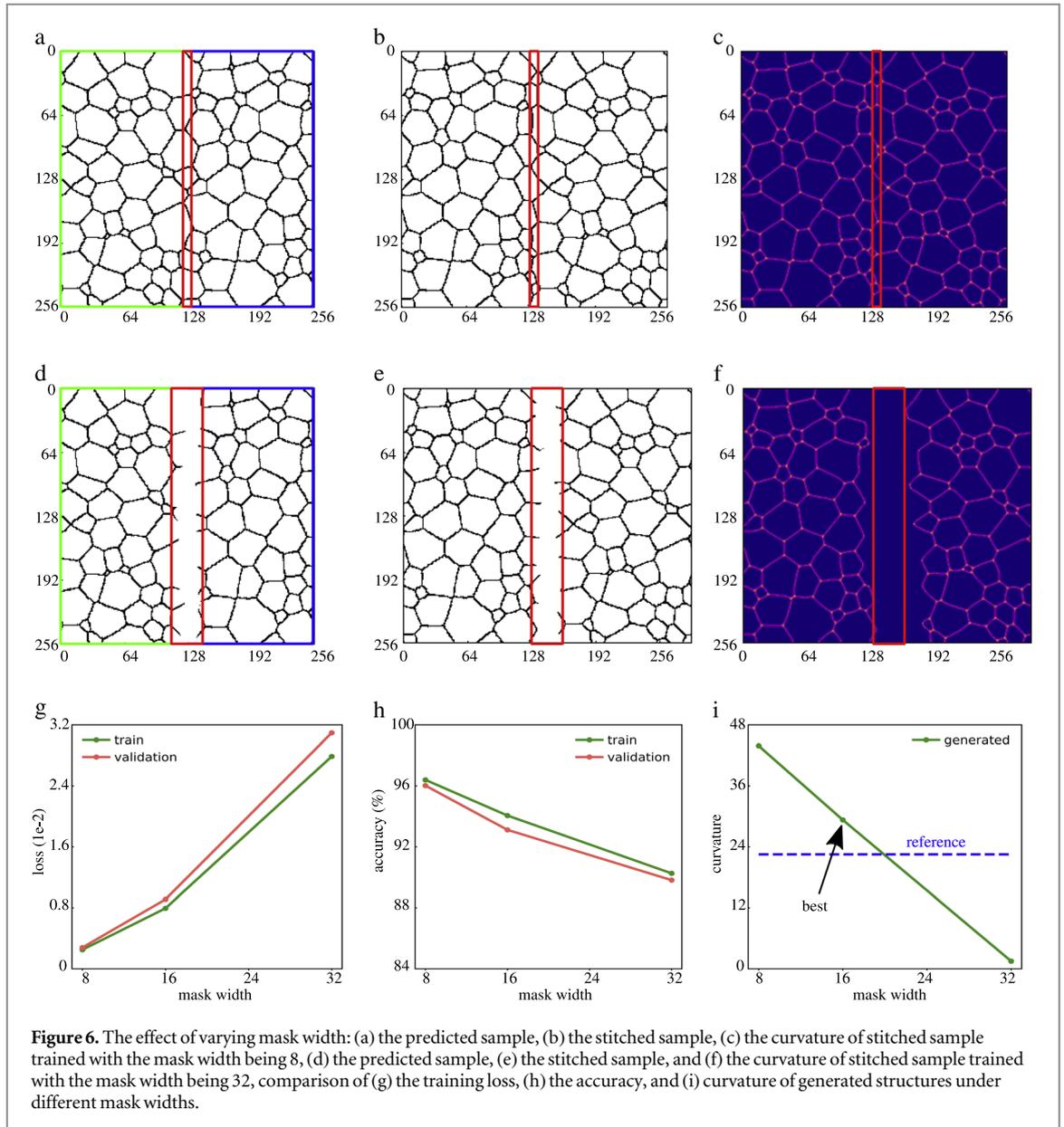
Figure 5. The evaluation of the stitching: segmentation of (a) original configuration, (b) switched configuration, and (c) stitched configuration, corresponding to figures 4(a), (b), and (f), respectively. (d)–(f) the grain boundary map illustrating curvature of (a)–(c), respectively.

meaningful predictions? During the preprocessing step, the width of the mask seems to be an essential parameter because the convolutional neural network always captures the local feature, which implies that there should be an effective action length. The valid range of width is between zero and positive infinity. If it is small, there is not enough space for the connecting line segment, resulting in a distorted configuration. In the extreme case, there is no space, which was discussed in figures 4(b) and (e). If, on the other hand, it is so large that exceeds the width of the entire sample, the expected information is even more than the given information, which is unlikely. The extreme case of an infinite width, corresponding to generating an RVE from scratch, is also impractical.

In the previous results, the width of the mask is 16-cells (1/16 of sample width). A narrow width is tested for comparison, and the corresponding results are plotted in figures 6(a)–(c). A wide width is tested, and the related results are plotted in figures 6(d)–(f). Only the most representative configurations of the workflow are shown in these figures. In this context, figures 6(a) and (d) are the counterparts of figure 3(c); figures 6(b) and (e) are the counterparts of figure 4(f), and figure 6(c) and 6(f) are the counterparts of figure 5(f). The same visualization styles from the previous section are retained. In addition, the shape of the mask is explicitly marked. As expected, the boundary is connected in the narrow case, while it is not connected in the wide case, although the lines extend into the masked area. It is noteworthy that the boundaries in figure 6(e) overlap with the masked area several times. In figure 6(f), however, they shrink. It is likely that watershed segmentation tends to eliminate incomplete boundaries here. Recalling that the incomplete boundaries are closed by the watershed operation in the previous case, the two situations suggest that the mask width should not be too large. In addition, the training loss, accuracy, and accumulative curvature from the three cases are summarized and shown in figures 6g–i. It can be observed that the loss increase monotonically with the mask width, while the accuracy decreases in a similar manner. In figure 6(i), the green line represents the average accumulative curvature in the masked area, while the blue dashed line denotes the reference value in the unmasked area. It can be clearly seen that the value of the generated structure intersects with the reference value, suggesting that the width should be neither too narrow nor too wide. Around the intersecting point, a smaller mask width outperforms a larger one because the incomplete boundaries tend to close rather than disappear after the watershed operation.

4. Conclusion

In this paper, we proposed a U-Net-based algorithm to convert a nonperiodic microstructure into a periodic microstructure. The validity of this method is verified using 2D grain samples along one direction. The U-Net is



trained with masked data as input and the original data as ground truth. The training converges and successfully predicts the masked structure. A series of manipulations is performed to obtain a variant structure containing two nonmatching boundaries, separated by a masked area. The masked area in the variant structure is filled, and a periodic structure is obtained by adding the cropped parts back. A metric for evaluating the quality of the generated pattern is developed and the effect of the mask width, based on this metric, is studied. It is assumed that there is an optimal value of the width of the mask.

The method uses a neural network to detect and stitch the boundaries. It is not limited to the grain structure but can be used for many porous structures, and even more structures with statistical homogeneity. It should also be able to extend to 3D systems for practical processes such as additive manufacturing [30, 31]. In any case, the mask width should be essential, which may rely on the characteristic length of the dominant features. Further, for a microstructure where multiscale features exist, systematic work will be needed in order to investigate what determines the optimum mask width.

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Data availability statement

All data that support the findings of this study are included within the article (and any supplementary files).

Author contributions statement

B N supervised and conceived the project. Y J and P A conducted the multi-phase field simulations and analyzed the results. Y J and A K built the machine learning model, trained the model, and analyzed the results. Y J wrote the manuscript, A K, P A, L G, D R, B N reviewed and revised the manuscript.

Conflict of interest

The authors declare no competing interests.

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