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# Assessing the Mixing Quality of Hetero-Aggregates: Applying Mixing Theory to STEM-EDX Elemental Maps

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The hetero-aggregation of carbon black and colloidal silica in a spray flame is a promising approach to improve dispersibility and stability of carbon black. The mixing quality, i.e., homogeneity of the materials within the hetero-aggregate, is important for both properties, yet challenging to quantify. For this purpose, multiple STEM-EDX (Scanning Transmission Electron Microscopy with Energy Dispersive X-Ray Spectroscopy) elemental maps are conducted on two distinct process conditions, leading to hetero-aggregates two different mixing qualities. By sampling cutouts of the images at randomized locations and applying principles adapted from classical mixing theory, hetero-aggregates are characterized with respect to mixing quality, irrespective of composition and primary particle size. Furthermore, correlation coefficient functions give insight to the length scale of primary particle clusters in the hetero-aggregates. In addition, on the basis of the *intra-aggregate* mixing state and the distribution of hetero-aggregate composition, a suitable description of the *inter-aggregate* mixing state is achieved. The developed methods present a generally valid, precise characterization of the mixing quality of hetero-aggregates.

## 1 | Introduction

Nanoparticles are widely used in industrial products because they combine material-inherent physico-chemical properties with high surface to volume ratios [1–3]. These properties are multiplied and complemented by the functional mixing with other nanoparticle species [4, 5]. Subsequently, hetero-contacts are formed at the distinct interface of the two materials, which are of fundamental importance for the desired functional properties of the mixture [6, 7]. Recently, research on hetero-aggregates has been intensified, especially due to possible applications as battery materials [8–10]. Carbon black is added therein to improve the conductivity by ensuring electrical pathways through binder polymers and active material. Therefore, high dispersion on nanoscale levels is necessary. However, during dry mixing, cluster formation and aggregate breakage of carbon black may lead

to decreased conductivity [11, 12]. Hence, in order to ease the manufacturing of batteries, methods to structure the conductive carbon are investigated [13–15].

One possible approach is the hetero-aggregation of colloidal silica nanoparticles onto the aggregate structure of carbon black in spray flames as investigated in a previous publication [16]. By sintering the two distinct materials on the primary particle level, it is possible to enhance their aggregate stability. In order to reach the desired macroscopic properties, the hetero-aggregation process and its influencing factors have to be understood on a microscopic level. However, the resulting hetero-aggregates exhibit highly fractal nanoscale structures that consist of two different amorphous materials and therefore, a reliable characterization can be challenging. Usually, multiscale characterization yields information ranging from primary particle properties to

aggregate properties [17]. Other common characterization methods for hetero-aggregates are laser light diffraction [18, 19], X-ray diffraction [20] and electron microscopy [21]. The characterization of the internal mixing of hetero-aggregates is a focal point of research because functional properties like catalytic performance or lithium transport are governed by it [19, 22]. One approach is the introduction of a heterogeneous coordination number as a measure for the quality of hetero-aggregation in which the number of hetero-contacts is compared to the number of homo-contacts [23]. This concept has been developed further by including convolutional neural networks and machine learning [24, 25]. This way, metal oxide hetero-aggregates with substantial difference in electron density are characterizable. However, due to weaker intensity contrast in high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM), carbon-based materials or materials with lower coordination numbers cannot be characterized by this method. Furthermore, both primary particle species need to be distinguishable and 3D reference is necessary, limiting the general applicability of this method. Therefore, to investigate carbon black-silica hetero-aggregates, the determination of coordination numbers is not feasible. Furthermore, the heterogeneous coordination number depends on the concentration of the respective materials, making it difficult to compare process conditions with varying concentrations.

Other possible approaches to describe mixing quality include the usage of the Rényi entropy [26]. In this context, the distribution of mixing concentrations within twin-screw extruders has been evaluated [27]. Further measures for the mixing quality are two-point correlation functions which yield information on the state of mixing at different length scales and are directly applicable to 3D and even 2D image data [28]. Furthermore, insights on co-segregation phenomena of terbium and oxygen within aluminum nitride on the basis of electron energy-loss spectrum (EELS) imaging has also been investigated [29]. This imaging technique has also been used in the investigation of hetero-atom infusion of metal oxides in soot [30].

An alternative methodology involves the adaptation of the empirical variance of concentration to scanning electron microscopy with energy dispersive X-ray spectroscopy (SEM-EDX). The basic principle is the evaluation of the local enrichments in cutouts and comparing it with the bulk enrichment. The adaption to SEM is conducted by choosing image cutouts of an investigated hetero-aggregate as samples and the whole hetero-aggregate as bulk [31, 32]. However, the use of SEM-EDX analysis is constrained by the fact that it only allows for the consideration of surface composition.

To date, only the empirical variance of local concentrations has been considered, which is significantly influenced by the size of the image cutout used for quantifying the variance and the composition of the hetero-aggregate. Hence, a quantitative comparison of mixing quality is reliable primarily for hetero-aggregates of similar shape and composition with a consistent number of particles in the considered cutout, making the characterization prone to misinterpretation. For nanoscale hetero-aggregates, STEM-EDX elemental maps yield a 2D projection of the material distribution within the aggregate. More detailed information would only be available by tomography which is

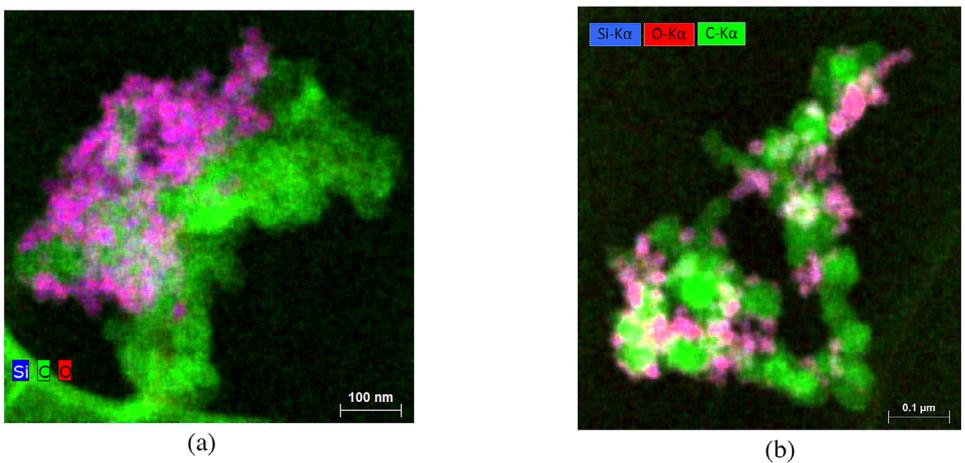
not feasible for a statistical analysis. Another possibility to gain insight on the material distribution within micron-sized samples is the employment of FIB-STEM [33]. Therefore, the goal of this paper is the development of a method to precisely describe the mixing quality of hetero-aggregates from STEM-EDX elemental maps, regardless of dimension, shape and material, with a focus on the nanoscale. To the best of the author's knowledge, no such adaption of mixing theory for the determination of intra-aggregate mixing on the nanoscale has been achieved yet. Furthermore, the correlation between mixing quality on the nanoscale (*intra-aggregate*) and the macroscopic scale (*inter-aggregate*) of the hetero-aggregates poses a scientific issue which is yet to be solved. It is pivotal that the resulting measure of mixing quality is able to describe differences in mixing quality for distinctly formed hetero-aggregates, allowing the comparison of multiple manufacturing processes.

## 2 | Materials and Methods

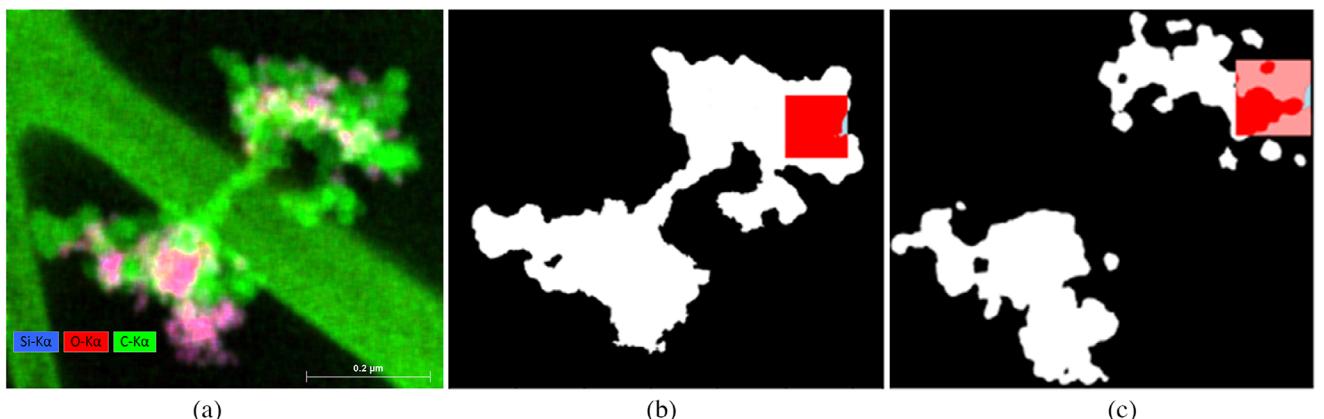
The manufacturing process of the hetero-aggregates considered in this paper is described in Section 5. In this context, silica particles are sintered on top of an already existing carbon black aggregate structure by spray flame synthesis. Section 2.1 deals with the workflow used for the processing of EDX scans. Finally, relevant aspects of classical mixing theory and their adaptation to transmission images are explained in Section 2.2.

### 2.1 | Image Processing and Sampling

For the production of the hetero-aggregates two distinct process conditions with two different kinds of carbon black were chosen. Exemplary STEM-EDX elemental maps of the synthesized hetero-aggregates are depicted in Figure 1. For the first process conditions (Figure 1a), denoted with KB for the Ketjenblack-type carbon black employed in synthesis, segregation was observed. On the other hand, in the hetero-aggregates synthesized with a C65-type carbon black, silica was well dispersed on top of the carbon black structure as shown in Figure 1b. The STEM-EDX elemental maps provide spatially resolved detection of local silica and carbon enrichments within the hetero-aggregates. However, background noise, the lacey carbon TEM grid, and low signal strength of carbon and silicon make image processing necessary. The workflow is illustrated in Figure 2. Separate images of the isolated signals of silicon and carbon are available directly from the EDX measurement. The images were then processed with the image editing software GNU Image Manipulation Program [34]. Concerning the carbon scan, first the lacey carbon film was edited out manually. To further reduce the background noise, a threshold was introduced. Subsequently, the images were transformed into binary images, so that the background consists of black pixels and the elemental information of white pixels. It is necessary to verify that the background is completely black. Otherwise the resulting mixing quality is not representative of the actual mixing state. The results were two separately segmented images of the carbon and silicon distribution within the hetero-aggregate. For image analysis, a Python script has been developed in which digital cutouts with a fixed number  $N$  of particles in the cutouts are collected on  $n_{\text{samples}}$  randomized locations for both carbon and silicon. Such a cutout is visualized by the rectangles in Figure 2b,c.



**FIGURE 1** | Exemplary STEM-EDX elemental maps of the hetero-aggregates obtained for two different process conditions. The carbon content is depicted in green, whereas the  $\text{SiO}_2$  particles appear pinkish. (a) Hetero-aggregates with Ketjenblack-type carbon black, exhibiting segregation. (b) Hetero-aggregates with C65-type carbon black, exhibiting a homogeneous distribution of silica particles.



**FIGURE 2** | Workflow of image processing. (a) Unedited EDX scan of a hetero-aggregate. Segmented image of the carbon structure (b) and the silica structure (c) within the hetero-aggregate with evaluated image cutout. In the evaluated image cutout in (c), the silicon content is marked with dark red, the carbon content within the image cutout in muted red and the background in blue.

In this context, pixel-wise information on silicon and carbon content is collected. Note that the results are not showing actual chemical concentrations of elements at any point, for which  $k$ -factors and detector sensitivities would have to be calculated, but only their lateral distribution. However, in the reference frame of binary images of STEM-EDX elemental maps of carbon black - silica hetero-aggregates, the term concentration denotes the ratio of carbon counts divided by the combined count of carbon and silica pixels to keep a consistent terminology with regards to classical mixing theory. More precisely, the local concentration of carbon  $X_i$  in the  $i$ -th cutout, for each  $i \in \{1, \dots, n_{\text{samples}}\}$ , denotes the number of carbon pixels divided by the total number of foreground pixels (number of pixels with carbon and number of pixels with silica) in the cutout. Encountered background pixels are ignored. From the primary particle size distributions, the mean primary particle diameter of both materials is calculated. For both materials, the assumption of a spherical shape of the primary particles is valid as already demonstrated in earlier work [16]. For a proper assessment of the mixing quality it is necessary that each sample yields the same number of pixels corresponding to a specified number of particles. Hence, this

mean projected area in pixels is calculated by combining the mean primary particle diameter and the mass ratio employed in synthesis. To generate an individual cutout, a random foreground pixel is selected as the center of a square whose size is chosen such that its intersection with the foreground region corresponds to the expected area of  $N$  primary particles.

## 2.2 | Adaption of Mixing Theory for STEM-EDX Elemental Maps

In particle technology, the mixing quality is of fundamental importance for the properties of mixtures. One example is the pharmaceutical industry, where the correct amount of active ingredients in a certain product (e.g. a tablet) is crucial to ensure proper healing. It is pivotal that the homogeneity of mixtures is high and assessed correctly, since otherwise some products will contain too much, while others contain too little active ingredients. In this context, classical mixing theory was developed in the 1950–1970s [35–37], summarized for example by J. Bridgwater [38]. Notably, P.V. Danckwerts [39] suggested the

usage of two measures for the characterization of mixing quality: the intensity and scale of segregation. The basic concept of the *intensity of segregation* is the evaluation of differences between cutout and bulk concentration. In order to determine the *intra-aggregate* mixing quality,  $n_{\text{samples}}$  cutouts with a number of pixels corresponding to an area with  $N$  primary particles are evaluated at randomly chosen locations within the 2D image, see Section 2.1.

### 2.2.1 | Empirical Variance of local Concentration

As already mentioned above, the local carbon concentration  $X_i$  is computed as carbon content divided by the sum of carbon and silica contents of the  $i$ -th cutout, for each  $i \in \{1, \dots, n_{\text{samples}}\}$ . Furthermore, the local enrichment of carbon, denoted with the carbon concentration  $P$ , of a given hetero-aggregate is calculated from the binarized STEM-EDX elemental map. By evaluating the differences between local and hetero-aggregate carbon concentrations for  $n_{\text{samples}}$  cutouts within the investigated hetero-aggregate, the empirical variance of concentration

$$s_{\text{intra}}^2 = \frac{1}{n_{\text{samples}} - 1} \sum_{i=1}^{n_{\text{samples}}} (X_i - P)^2 \quad (1)$$

is obtained. Note that under some regularity assumptions on the sampling scheme, the empirical variance  $s_{\text{intra}}^2$  given in Eq. (1) converges to some limit  $\sigma^2$  for  $n_{\text{samples}} \rightarrow \infty$ , which can be interpreted as theoretical (true) variance of local carbon concentrations. Hence, the higher the number of cutouts  $n_{\text{samples}}$ , the higher the accuracy of the empirical variance  $s_{\text{intra}}^2$ .

Analogously, the empirical variance  $s_{\text{inter}}^2$  of carbon concentration for  $n_{\text{HetAgg}}$  hetero-aggregates is given by

$$s_{\text{inter}}^2 = \frac{1}{n_{\text{HetAgg}} - 1} \sum_{i=1}^{n_{\text{HetAgg}}} (P_i - \bar{P})^2 \quad (2)$$

which is a descriptor of the *inter-aggregate* mixing quality. In eq. (2) the composition of individual hetero-aggregates is compared with the mean of all hetero-aggregates. Here,  $P_i$  denotes the calculated carbon concentration from the binarized STEM-EDX elemental map of the  $i$ -th hetero-aggregate, for each  $i \in \{1, \dots, n_{\text{HetAgg}}\}$ , whereas  $\bar{P}$  denotes the mean carbon concentration of the  $n_{\text{HetAgg}}$  hetero-aggregates.

### 2.2.2 | Normalizing the Variance to Its Bounds

Recall that the composition of the hetero-aggregates investigated in this paper consist of two different materials. However, STEM-EDX elemental maps are considered in which either carbon or silicon content is depicted in 2D transmission images through the 3D hetero-aggregates. Hence, for pixels of the 2D images four cases are observed: Pixels that exhibit solely carbon black or silica which are denoted by [1,0] and [0,1], respectively. Because of transmission of the electron beam through the hetero-aggregate, two further cases occur, where [1,1] denotes pixels which exhibit both carbon black and silica. Finally, [0,0] denotes pixels which yield no information on the hetero-aggregates (background) and are disregarded. Depending on the composition of a hetero-

aggregate, the fraction  $P_{1,0}$  of [1,0] pixels associated with the hetero-aggregate and, analogously, the fractions  $P_{0,1}$ ,  $P_{1,1}$  of [0,1] and [1,1] pixels are determined, where

$$P_{1,0} + P_{0,1} + P_{1,1} = 1 \quad (3)$$

Hence, the values of  $P_{1,0}$ ,  $P_{0,1}$ , and  $P_{1,1}$  are (empirical) estimates of the probabilities that a pixel belongs to class [1,0] with a carbon concentration of 1, [0,1] with a carbon concentration of 0 and [1,1] with a carbon concentration of 0.5, respectively.

The notion of an *unmixed state* is common in mixing theory and represents the upper bound of the *intra-aggregate* variances  $s_{\text{intra}}^2$  given in Eq. (1) [40]. Considering an unmixed hetero-aggregate, we assume that all pixels of a local cutout only belong either to class [1,0], [0,1] or [1,1], i.e, the materials are perfectly separated. Recall that cutouts consisting of [1,1] pixels can occur even in the unmixed state due to transmission imaging. For such cutouts, the ratio of carbon and silica is equal, leading to a carbon concentration of 0.5. Thus, the variance of the unmixed state, denoted by  $s_{\text{intra},\text{max}}^2$ , of a hetero-aggregate with the carbon concentration  $P$  is given by

$$s_{\text{intra},\text{max}}^2 = P_{1,0}(1 - P)^2 + P_{0,1}(0 - P)^2 + P_{1,1}(0.5 - P)^2 \quad (4)$$

On the other hand, the empirical variance  $s_{\text{intra}}^2$  considered in Equation (1) can take arbitrary small values and even zero, when all cutouts – by chance – exactly match the overall carbon concentration  $P$ . In practice, however, a certain lower bound  $s_{\text{intra},\text{min}}^2$  of the *intra-aggregate* variance  $s_{\text{intra}}^2$  is commonly referred to as the state of *completely random mixing* [38]. It depends on the primary particle size distributions as well as on the chosen number of particles  $N$  in the image cutout and is given by

$$s_{\text{intra},\text{min}}^2 = \frac{P(1 - P)}{N} \left( \frac{d_{\text{CB,PP}} d_{\text{SiO}_2,\text{PP}}}{d_{\text{mean,PP}}^2} \right)^2 \times \left( 1 + (1 - P) \sigma_{\text{SD,CB}}^2 + P \sigma_{\text{SD,SiO}_2}^2 \right) \quad (5)$$

where  $d_{\text{CB,PP}}$  and  $d_{\text{SiO}_2,\text{PP}}$  denote the mean primary particle diameters of carbon black and silica, respectively, and  $d_{\text{mean,PP}}$  is the combined mean primary particle diameter of the two materials, depending on the carbon concentration  $P$  of the hetero-aggregate. Furthermore,  $\sigma_{\text{SD,CB}}^2$  and  $\sigma_{\text{SD,SiO}_2}^2$  are the variances of the primary particle size distributions, by which the polydispersity of the materials is taken into account.

The mixing quality index according to Ashton and Schmahl [41]

$$M_{\text{AS}} = \frac{\log(s_{\text{intra},\text{max}}^2 / s_{\text{intra}}^2)}{\log(s_{\text{intra},\text{max}}^2 / s_{\text{intra},\text{min}}^2)} \quad (6)$$

ranks the empirical variance of carbon concentration of a given hetero-aggregate between 0 and 1, respective to its bounds: When the index  $M_{\text{AS}}$  equals zero, the variance  $s_{\text{intra}}^2$  of the hetero-aggregate is equal to the variance  $s_{\text{intra},\text{max}}^2$  of the unmixed state, whereas  $M_{\text{AS}} = 1$  indicates that the variance of the hetero-aggregate is equal to the variance of completely random mixing [41]. Hence, the index  $M_{\text{AS}}$ , briefly called *Schmahl index* in the following, allows to characterize the mixing quality of

hetero-aggregates, indifferent from carbon concentration, number of particles in the cutouts and primary particle size and, thus, represents a more objective measure for characterizing the mixing quality of hetero-aggregates.

### 2.2.3 | Empirical Covariance and Correlation Coefficient Function

According to Danckwerts [39], the *scale of segregation* describes the state of mixing with respect to varying length scales. In this context,  $n_h$  pairs of pixels of the binarized STEM-EDX elemental maps (can be considered to be cutouts with size 1 x 1) at distances  $h \geq 0$  are examined with the corresponding local carbon concentrations  $X_i$  and  $Y_i = X_{j(i)}$  for each  $i \in \{1, \dots, n_h\}$ , where  $j(i)$  denotes the index of a pixel that has a distance of  $h$  to the pixel with index  $i$ . In other words,  $X_i$  and  $Y_i$  denote the carbon concentrations (1, 0.5 or 0) of two randomly chosen pixels, which have a distance of  $h$  from each other. For other material systems, for which binarization of the STEM-EDX elemental maps is not necessary, the original integer STEM-EDX elemental maps might be used for potentially more accurate results. In this context, the investigation of line scans for improved insights of multiple particle overlapping scenarios is also possible [42]. For each distance  $h \geq 0$  such that  $n_h > 1$ , the empirical covariance  $C(h)$  of local carbon concentrations is then given by

$$C(h) = \frac{1}{n_h - 1} \sum_{i=1}^{n_h} (X_i - P)(Y_i - P) \quad (7)$$

Subsequently, the values  $R(h)$  of the correlation coefficient function for distances  $h \geq 0$ , with  $n_h > 1$ , is given by

$$R(h) = \frac{C(h)}{s_{\text{intra},\max}^2} \quad (8)$$

The correlation coefficient function characterizes the state of mixing at different length scales. More precisely, for values of  $R(h)$  close to 1 or -1, the local carbon concentrations are strongly positively or negatively correlated at distances of  $h$ , which indicates segregation. On the other hand, values of  $R(h)$  close to zero indicate that the local carbon concentrations of two pixels with a distance of  $h$  are uncorrelated, indicating a good state of mixing at that range. Therefore, if the hetero-aggregate exhibits primary particle clusters, the local concentrations are correlated more strongly over a larger distance, ultimately leading to a slower decay rate of  $R(h)$ .

In practice, using Equation (8), the correlation coefficients  $R(h)$  are computed for finitely many distances  $h$  from the interval  $[0, h_{\max}]$  for some integer  $h_{\max} > 0$ , e.g.,  $h = 0, 1, \dots, h_{\max}$ , given that  $n_0, \dots, n_{h_{\max}} > 1$ . However, the values of  $R(0), R(1), \dots, R(h_{\max})$  obtained in this way may exhibit fluctuations due to spatial variability or noise, which necessitates smoothing. This is achieved, for instance, by smoothing with a Gaussian kernel [43] to obtain a more robust and interpretable representation of the correlation coefficient function  $R : [0, h_{\max}] \rightarrow [-1, 1]$  that is defined on the continuous interval  $[0, h_{\max}]$ . This approach enables the interpolation of the values of  $R(h)$  for non-integer-valued distances  $h$ . By applying

this procedure to several hetero-aggregates,  $n_{\text{HetAgg}}$  smoothed correlation coefficient functions are obtained, denoted by  $R_i$  for  $i \in \{1, \dots, n_{\text{HetAgg}}\}$ . To compute the mean correlation coefficient function  $\bar{R} : [0, h_{\max}] \rightarrow [-1, 1]$  for the respective set of process conditions, C65 and KB, point-wise averaging is performed, i.e.,

$$\bar{R}(h) = \frac{1}{n_{\text{HetAgg}}} \sum_{i=1}^{n_{\text{HetAgg}}} R_i(h) \quad (9)$$

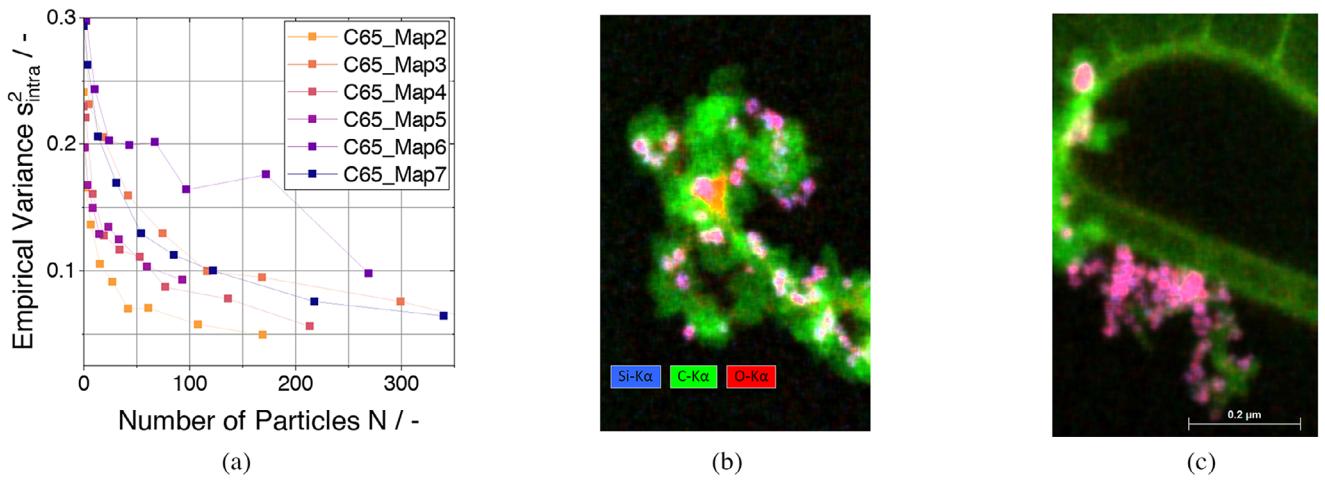
for each  $h \in [0, h_{\max}]$ . Note that, the correlation coefficient function for the silica concentration (which results by substituting  $X_i$  by  $(1 - X_i)$  and  $Y_i$  by  $(1 - Y_i)$  in the computation of the correlation coefficient function) actually coincides with  $\bar{R}$ , due to the bilinearity of the covariances  $C(h)$ .

## 3 | Results

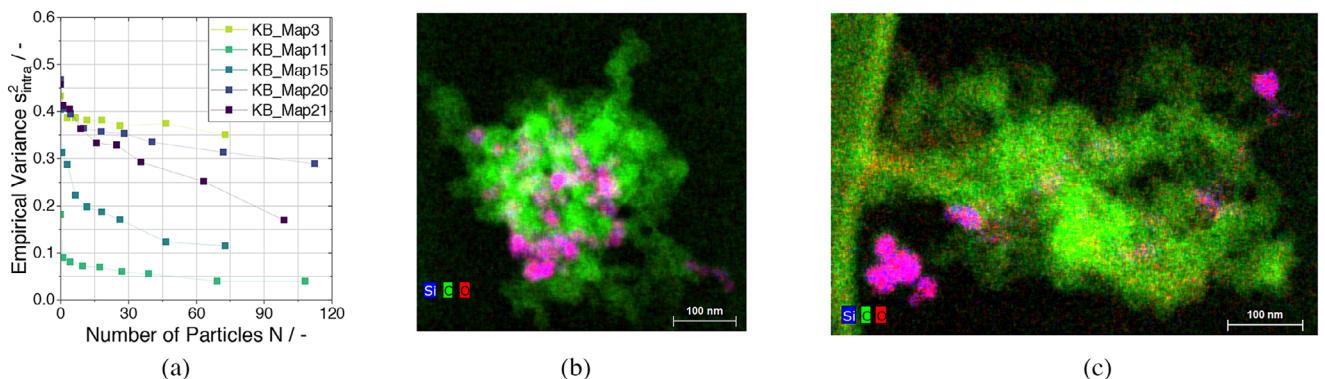
First, in Section 3.1, the empirical variance of local carbon concentration is considered in order to determine the *intra-aggregate* mixing quality of C65-type and KB-type hetero-aggregates. In this context, sampling is repeated multiple times for different numbers of primary particles in the image cutouts, to assess the dependency of the empirical variance on the number of particles within the cutouts. The initial assessment investigates six STEM-EDX elemental maps of C65-type hetero-aggregates with perceived good mixing quality. Moreover, five STEM-EDX elemental maps for KB-type hetero-aggregates with inhomogeneous hetero-aggregation were investigated to check if the empirical variance of local carbon concentration is able to differentiate between these mixing states. Then, in Section 3.2, 50 hetero-aggregates of the C65-type and 30 hetero-aggregates of the KB-type are evaluated at fixed image cutout sizes corresponding to  $N = 25$  primary particles. In this context, the Schmahl index of each hetero-aggregate is determined for the respective variances of completely random mixing and complete segregation. Furthermore, Section 3.3 discusses how the state of *inter-aggregate* mixing can be described by combining the mean Schmahl index, its empirical variance and the empirical variance of hetero-aggregate concentration. Finally, Section 3.4 investigates the usage of the correlation coefficient function to characterize the *scale of segregation* in hetero-aggregates.

### 3.1 | Empirical Variance of Local Concentration as Measure for Mixing Quality

In order to achieve a fundamental understanding of hetero-aggregation, a proper assessment of *intra-aggregate* mixing quality is necessary. In this context, higher deviations of the local carbon concentration in randomized cutouts result in an increased empirical variance. Consequently, a high empirical variance of carbon concentration translates into a low homogeneity of the produced hetero-aggregates. This holds also true for the investigated case of silica particles sintered on top of an already existing carbon black aggregate. Figure 3a depicts the changes in empirical variance of local carbon concentration of six C65-type hetero-aggregates for different numbers  $N$  of particles in image cutouts taken at 120 random locations. The figure highlights the importance of choosing a suitable number



**FIGURE 3** | (a) Empirical variance  $S_{\text{intra}}^2$  of local carbon concentration over a total of 120 cutouts for varying number of particles in the cutouts. Six segmented STEM-EDX elemental maps of C65 type hetero-aggregates were investigated (Map2–Map7). Exemplary STEM-EDX elemental maps of Map2 (b) and Map6 (c) with carbon (green) and  $\text{SiO}_2$  (pink) signals. Map2 exhibits a homogeneous distribution of silica particles on the carbon black aggregate structure, whereas Map6 shows a cluster of silica particles, leading to a reduced mixing quality.

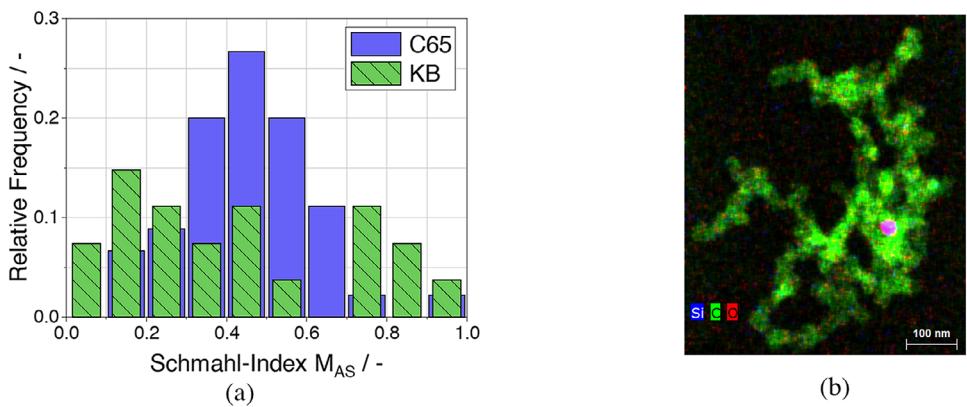


**FIGURE 4** | (a) Changes in the empirical variance  $S_{\text{intra}}^2$  of local carbon concentration over a total of 120 cutouts for varying number of particles in the cutouts. Six segmented EDX scans of KB-type hetero-aggregates were investigated. Exemplary EDX scans of two hetero-aggregates of the KB-type (b, c) which exhibit similar empirical variances of local carbon concentration for  $N = 25$  primary particles in the cutouts.

of particles within the cutouts: The smaller the number  $N$  of primary particles, the higher the resulting empirical variance  $S_{\text{intra}}^2$ . This is especially evident when a cutout of only a single pixel is taken. In this case, the variance of the unmixed state  $S_{\text{intra},\text{max}}^2$  given in Equation (4) is obtained, as expected according to its definition. For an increasing number of particles in the cutouts, the local carbon concentration is approaching the (global) hetero-aggregate carbon concentration, resulting in a decrease toward zero. In between, the variance  $S_{\text{intra}}^2$  is only weakly dependent on the image cutout size, resulting in a range where the values of the empirical variance of local concentration of the hetero-aggregates are sound. For a hetero-aggregate with a more homogeneous distribution of carbon and silica particles (“C65\_Map2”), as depicted in Figure 3b, the variance  $S_{\text{intra}}^2$  is lower with a value of 0.08 for  $N = 60$  primary particles in the cutouts. Remarkably, for “C65\_Map6,” as depicted in Figure 3c, an accumulation of silica primary particles was present, resulting in a higher variance of 0.2 for the same number of particles in the cutouts. This shows the general feasibility of this method to quantify apparent differences in hetero-aggregate homogeneity.

Figure 4a depicts the variance  $S_{\text{intra}}^2$  of Ketjenblack hetero-aggregates for different numbers  $N$  of particles in the cutouts, ranging from 1 pixel to 100 primary particles. It is apparent that the observed increase in variance is associated with diminished quality of hetero-aggregation. This is particularly evident in the STEM-EDX elemental map denoted by KB\_Map3, where a nearly constant empirical variance of local concentration with  $S_{\text{intra}}^2 = 0.38$  was observed, independently of the image cutout size.

However, by comparing the EDX maps in Figure 4b,c, a major deficit of the empirical variance of local concentration as mixing quality criterion is highlighted: For  $N = 25$  primary particles in the cutouts, both hetero-aggregates exhibit nearly the same empirical variance  $S_{\text{intra}}^2$  of local concentration between 0.14 and 0.15, even though in Figure 4c segregation of the materials is apparent. The similarity of variances can be explained by differences in concentration. The hetero-aggregate depicted in Figure 4c exhibits a carbon content of  $P = 0.92$ , whereas the hetero-aggregate shown in Figure 4b has a lower carbon content of  $P = 0.77$ . This means that the variance is not always suitable as



**FIGURE 5** | (a) Histograms of the Schmahl index  $M_{AS}$  for 50 STEM-EDX elemental maps of C65-type hetero-aggregates, and for 30 TEM-EDX maps of the KB-type hetero-aggregates. The respective empirical variances of concentration were calculated on the basis of 150 cutouts and cutout sizes corresponding to  $N = 25$  primary particles. (b) STEM-EDX elemental map of a KB-type hetero-aggregate which exhibits a Schmahl index  $M_{AS} \approx 1$ , due to the low silica content of the hetero-aggregate.

a measure of mixing quality: The closer the cutout composition is toward an even distribution of the materials, the higher is the empirical variance  $s_{\text{intra}}^2$  of local concentration. Therefore, the bounds of the empirical variance of local concentration given in Equations (4) and (5) have to be regarded as well in order to compare hetero-aggregates with varying compositions and primary particle sizes.

### 3.2 | Schmahl Index as Metric to Characterize the Mixing Quality of Hetero-Aggregates

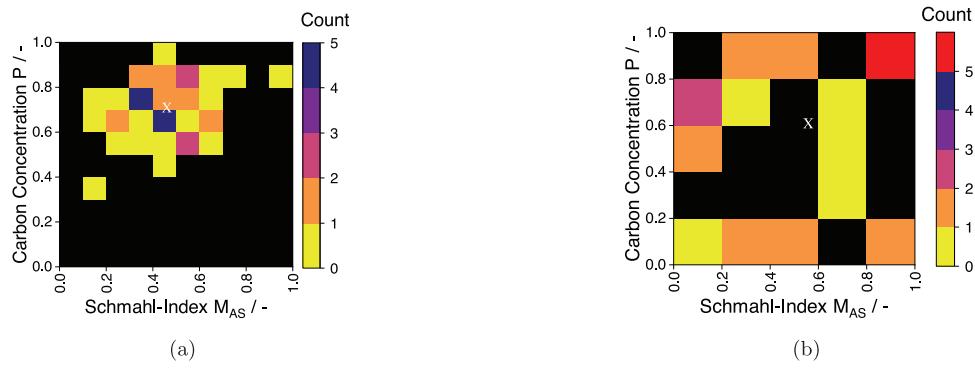
The main advantage of mixing indices like the Schmahl index  $M_{AS}$  given in Equation (6) is that information about composition (via the variance  $s_{\text{intra,max}}^2$  of the unmixed state given in Equation (4)) and particle size (via the variance  $s_{\text{intra,min}}^2$  of completely random mixing given in Equation (5)) is included in the mixing quality measure. Consequently, by using such measures, hetero-aggregates of different composition and particle sizes can be compared to each other in terms of their mixing quality. Figure 5a depicts histograms of the Schmahl index  $M_{AS}$  computed from 50 EDX maps of C65-type hetero-aggregates and 30 EDX scans of KB-type hetero-aggregates, where the histogram for C65-type hetero-aggregates (purple columns) indicates that in this case the probability distribution of the Schmahl index  $M_{AS}$  can be reasonably well approximated by a conventional parametric family of distributions, such as the normal or log-normal distribution. Conversely, the probability distribution of  $M_{AS}$  for KB-type hetero-aggregates is multi-modal and, thus, does not align with conventional parametric distribution types. The mean value of mixing quality assigned by the Schmahl index  $M_{AS}$  is equal to 0.458 for the C65-type hetero-aggregates, whereas the KB-type hetero-aggregates exhibit a mean Schmahl index of 0.557. For reference, the corresponding empirical variance of  $M_{AS}$  for all evaluated STEM-EDX elemental map is equal to 0.033 for the C65-type hetero-aggregates, and 0.125 for the KB-type hetero-aggregates.

In contrast to the anticipated probability distribution of the Schmahl index  $M_{AS}$  for C65-type hetero-aggregates, the distribution of  $M_{AS}$  for KB-type hetero-aggregates exhibits an

accumulation of hetero-aggregates with a Schmahl index above 0.8, indicating a high *intra-aggregate* mixing quality despite the apparent segregation of carbon black and silica. Notably, the mean Schmahl index  $\bar{M}_{AS}$  of 0.557 for KB-type hetero-aggregates exceeds the mean value of 0.458 for C65-type hetero-aggregates, despite the segregation of carbon black and silica qualitatively observed in the STEM-EDX elemental maps. In this context, Figure 5b depicts an exemplary STEM-EDX elemental map of a KB-type hetero-aggregate with  $M_{AS} \approx 1$ . Due to the segregation of both materials, hetero-aggregates with a skewed composition are synthesized, in which a single silica particle is sintered on top of the carbon black backbone. Sampling will naturally result in low silica concentrations, which are close to the true concentration  $P$ . Furthermore, a single primary particle can – by definition – not be clustered. Although not intuitive, this results in a Schmahl index  $M_{AS} \approx 1$ , independently of the position of the single silica particle. Therefore, the more segregated both materials are, the more hetero-aggregates with skewed concentrations are obtained that do not correspond to the overall mean concentration  $\bar{P}$ . This results in more hetero-aggregates with  $M_{AS} \approx 1$ . Consequently, quantifying the *intra-aggregate* mixing quality for highly segregated systems with the Schmahl index  $M_{AS}$  alone appears infeasible. This argument is supported by exclusion of all KB-type hetero-aggregates with a carbon content  $P$  above 0.95: In this test case, a mean Schmahl index  $\bar{M}_{AS}$  of 0.35 with a respective empirical variance of 0.05 is obtained. In comparison to C65-type hetero-aggregates, the corrected Schmahl index of KB-type hetero-aggregates is lower while the empirical variance of the Schmahl index of KB-type hetero-aggregates is higher.

### 3.3 | Combining Schmahl Index and Inter-Aggregate Variance

Although the *intra-aggregate* distribution of the Schmahl index  $M_{AS}$ , as discussed in Section 3.2, encompasses principal information for comparing different types of hetero-aggregates, it is insufficient to conclusively describe the *inter-aggregate* mixing state of segregated systems like KB. It is essential to also assess the distribution of concentration  $P$  to ensure a comprehensive interpretation of the mixing state. Figure 6 depicts heatmaps



**FIGURE 6** | Heatmaps of the Schmahl index  $M_{AS}$  for two different types of hetero-aggregates and their corresponding carbon concentration  $P$ , where 50 EDX scans of C65-type hetero-aggregates (a), and 30 EDX scans of KB-type hetero-aggregates (b) have been evaluated.

of the evaluated hetero-aggregates in which the carbon concentration  $P$  of each individual hetero-aggregate is linked to its corresponding Schmahl index  $M_{AS}$ . As illustrated in Figure 6a, the heatmap of C65-type hetero-aggregates exhibits a joint distribution of  $M_{AS}$  and  $P$  that is clustered around the respective mean values. The calculated mean carbon concentration of 0.71 and the corresponding variance of 0.015 agree well to the mass concentration of 0.75 employed in synthesis. In contrast, the heatmap of KB-type hetero-aggregates (shown in Figure 6b) reveals a different pattern, where aggregates are either grouped at high or low carbon concentrations. This results in a substantially higher variance in carbon concentration of the hetero-aggregates, reaching a value of 0.125, while the mean carbon concentration of these hetero-aggregates approaches the uniform mass ratio employed in synthesis, with a value of 0.61, including the hetero-aggregates with a carbon concentration  $> 0.95$ . The mean values of Schmahl index and carbon concentration are highlighted with the symbol x in Figure 6.

In the following, we summarize these observations in a quantitative measure of mixing quality and therefore ascertain a unified description of the *inter-aggregate* mixing state. Note that the inter-aggregate empirical variance  $s_{\text{inter}}^2$  of carbon concentration, as defined in Equation (2), has the upper bound  $\bar{P}(1 - \bar{P})$ , which corresponds to the variance of a Bernoulli distribution, i.e., to the case where only hetero-aggregates with carbon concentrations of 0 and 1 occur with probability  $1 - \bar{P}$  and  $\bar{P}$ , respectively. By linking  $s_{\text{inter}}^2$  with the upper bound  $\bar{P}(1 - \bar{P})$ , the homogeneity of the carbon concentration between aggregates can be quantified by considering the mixing quality measure  $\psi_p \in [0, 1]$ , which is given by

$$\psi_p = 1 - \frac{s_{\text{inter}}^2}{\bar{P}(1 - \bar{P})} \quad (10)$$

Analogously, the homogeneity of the Schmahl index between aggregates can be quantified by

$$\psi_{M_{AS}} = 1 - \frac{s_{M_{AS}}^2}{\bar{M}_{AS}(1 - \bar{M}_{AS})} \quad (11)$$

where  $\bar{M}_{AS} = \frac{1}{n_{\text{HetAgg}}} \sum_{i=1}^{n_{\text{HetAgg}}} M_{AS,i}$  denotes the sample mean of the aggregate-wise Schmahl indices  $M_{AS,1}, \dots, M_{AS,n_{\text{HetAgg}}}$ , and  $s_{M_{AS}}^2$  is

the empirical variance given by

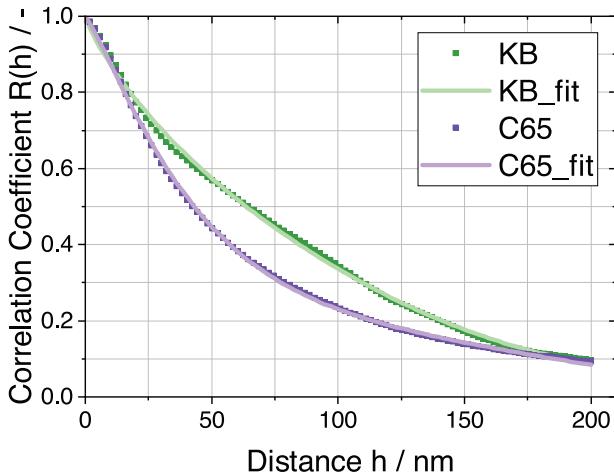
$$s_{M_{AS}}^2 = \frac{1}{n_{\text{HetAgg}} - 1} \sum_{i=1}^{n_{\text{HetAgg}}} (M_{AS,i} - \bar{M}_{AS})^2 \quad (12)$$

Both  $\psi_p$  and  $\psi_{M_{AS}}$  are mixing quality measures that take values in the interval  $[0, 1]$ . More precisely, these measures quantify the variability of carbon concentrations and Schmahl indices, respectively. In particular, values of  $\psi_p$  and  $\psi_{M_{AS}}$  close to zero indicate a large variance of carbon concentrations and Schmahl indices, whereas values of one indicate that each considered hetero-aggregate has the same *intra-aggregate* carbon concentration and Schmahl index. By computing the measures  $\psi_p$  and  $\psi_{M_{AS}}$  for both investigated systems of hetero-aggregates, it turns out that  $\psi_{M_{AS}} = 0.493$  and  $\psi_p = 0.474$  for KB-type hetero-aggregates, whereas  $\psi_{M_{AS}} = 0.867$  and  $\psi_p = 0.927$  for C65-type hetero-aggregates. This encapsulates the distinct dispersion patterns of the 2D heatmaps of KB- and C65-type hetero-agglomerates visualized in Figure 6.

The question remains how to define a “well-mixed” hetero-aggregate system. Based on all previous observations, we can state that a well-mixed system should possess a high average *intra-aggregate* mixing quality, while simultaneously possess narrow *inter-aggregate* distributions in mixing quality and concentration. By using the Schmahl index and the quantities derived from it as measure for the mixing quality, all these effects can be combined into a single mixing quality measure  $\psi_{\text{comb}}$ , where

$$\psi_{\text{comb}} = \frac{\alpha \bar{M}_{AS} + \beta \psi_{M_{AS}} + \gamma \psi_p}{\alpha + \beta + \gamma} \quad (13)$$

entirely describing the mixing state of a hetero-aggregate system. In Equation (13) the influence of the three measures  $\bar{M}_{AS}$ ,  $\psi_{M_{AS}}$ ,  $\psi_p$  is weighted with the factors  $\alpha, \beta, \gamma \in (0, 1]$  to account for specific applications. Using an equal weighting, i.e., putting  $\alpha = \beta = \gamma = 1$ , we obtain that  $\psi_{\text{comb}} = 0.751$  for C65-type hetero-aggregates and  $\psi_{\text{comb}} = 0.508$  for KB-type hetero-aggregates, finally differentiating the two process conditions at a glance and encapsulating the qualitative observations in a single number. Although desirable for a quick evaluation of different processing conditions, this kind of mixing quality measure has the disadvantage of data convolution and the dilution of information. Simply speaking, a low value of  $\psi_{\text{comb}}$  could be due to any of the three involved effects, making the identification of the responsible one impossible.



**FIGURE 7** | Correlation coefficients  $\bar{R}(h)$  of carbon concentrations as a function of distance  $h \geq 0$ . Note that  $\bar{R}$  coincides with the correlation coefficient function of  $\text{SiO}_2$  concentrations, see Section 2.2.3.

Thus, it is advisable to untangle the index  $\psi_{\text{comb}}$  into the three aforementioned indices to detect whether the *intra-aggregate* mixing quality is low ( $\bar{M}_{\text{AS}}$ ), the mixing is non-reproducible ( $\psi_{\text{AS}}$ ), or the composition of the individual hetero-aggregates is inhomogeneous ( $\psi_{\text{P}}$ ) for detailed analysis.

### 3.4 | Correlation Coefficient Function

Recall that, according to Danckwerts [39], the intensity of segregation and scale of segregation depict two distinct measures for the assessment of mixing quality. Hence, the behavior of the mixing quality at different length scales has been investigated by determining the correlation coefficient function  $\bar{R} : [0, h_{\max}] \rightarrow [-1, 1]$  as given in Equation (9). More precisely, for each of the STEM-EDX elemental maps of the KB-type hetero-aggregates, a correlation coefficient function has been computed followed by pointwise averaging, see the green line in Figure 7.

Analogously, the values of  $\bar{R}$  have been computed for C65-type hetero-aggregates, see the purple line in Figure 7. The significantly lower values of correlation coefficients in the range from 25 to 100 nm for C65-type hetero-aggregates indicate that the carbon concentrations (as well as the  $\text{SiO}_2$  concentrations) at such pairs of locations have a less pronounced dependency between each other. Thus, a better mixing quality for C65-type hetero-aggregates at smaller length scales is obtained. At coarser length scales, i.e., at distances around 175 nm both hetero-aggregate types exhibit similar mixing behavior. However, the total length of most investigated hetero-aggregates spans only a few hundred nm. Hence, the correlation coefficient function at these length scales is not well defined and may not reflect the actual mixing behavior. In some cases, the field of view also limits the applicability of the method toward large distances. For small distances, the method is limited by the effective spatial resolution of the STEM-EDX elemental map. Because the mixing process happens at the primary particle level (size range 10–40 nm) the influence of the spatial resolution of the STEM-EDX elemental maps is expected to be negligible.

**TABLE 1** | Parameters  $s_1, s_2, \nu, \lambda$  of the correlation coefficient function  $\bar{R}_{s_1, s_2, \nu, \lambda}$  fitted to STEM-EDX elemental map data, and coefficients of determination  $R^2$ .

Hetero-aggregate type	$s_1$	$s_2$	$\nu$	$\lambda$	$R^2$
KB	5.13	9.34	0.86	0.13	0.9984
C65	9.93	4.70	1.19	0.22	0.9995

The values of  $\bar{R}(h)$  visualized in Figure 7 can be considered as non-parametric estimates of the correlation coefficient function. Fitting a low-parametric model to these functions provides further insights, because such models are often described by interpretable parameters [44]. There are numerous parametric families of correlation coefficient functions (or covariance functions when considering the non-normalized scenario) [45]. In the case of the hetero-aggregates investigated in the present paper, it turned out that the parametric model  $\bar{R}_{s_1, s_2, \nu, \lambda} : [0, h_{\max}] \rightarrow [-1, 1]$  given as a convex combination of the scaled sinc function and an exponential term, i.e.,

$$\bar{R}_{s_1, s_2, \nu, \lambda}(h) = \lambda \text{sinc}(h/s_1) + (1 - \lambda) \exp(-(h/s_2)^\nu) \quad (14)$$

for each  $h \in [0, h_{\max}]$ , where  $s_1, s_2 > 0$  are scale parameters,  $\nu \in (0, 2]$  is a parameter that controls the decay rate, and  $\lambda \in [0, 1]$  controls the weighting of the scaled sinc function. Note that the parametric function given in Equation (14) fulfills the necessary conditions for being a valid correlation coefficient function, i.e., the parametric function  $\bar{R}_{s_1, s_2, \nu, \lambda}$  is positive semidefinite. By deploying `scipy.optimize.curve_fit` of the Python package Scipy [46], the parameters  $s_1, s_2, \nu$  and  $\lambda$  have been fitted in a least squares sense, see Table 1. Both the coefficients of determinations ( $R^2$ ) of the fits (Table 1) as well as their visualizations shown in Figure 7 (dashed lines) indicate an excellent fit. It turns out that, for both KB- and C65-type hetero-aggregates, the weighing factor  $\lambda$  exhibits relatively low values of 0.13 and 0.22 respectively. This means that the dependence structure is dominated by the exponential term in Equation (14). Furthermore, the parametric fit for C65-type hetero-aggregates indicates a significantly larger value of  $\nu$ , i.e.  $\nu = 1.19$ , in comparison to KB-type hetero-aggregates where  $\nu = 0.86$ . Thus, with the parametric fit a quantification of the larger decay rate of the spatial correlation of  $\text{SiO}_2$  for C65-type hetero-aggregates has been achieved. Therefore, the obtained results characterize the finer dispersion of silica particles in the C65-type hetero-aggregates and yield a description of the length scale of the hetero-aggregation. In conclusion, another key insight into the structure of hetero-aggregates that can influence their material properties is provided.

### 4 | Conclusion

In this paper, a methodology is presented to determine the mixing quality of hetero-aggregates by the analysis of STEM-EDX elemental map data. This includes an initial assessment of the empirical *intra-aggregate* variance of local carbon concentration as well as the Schmahl index, by integrating the variances of complete segregation and completely random mixing, i.e. theoretical upper and lower bounds. Typically, the objective in the synthesis of hetero-aggregates is to maximize the number

of hetero-contacts, which corresponds to a Schmahl index close to one [23]. However, for the hetero-aggregates investigated in this paper, the mean Schmahl index was roughly equal to 0.5 which indicates a mixing state where particles are neither well-mixed nor segregated. On the other hand, the evaluation of this specific value needs to be done in context of the synthesis: Silica particles are sintered onto a pre-existing carbon black backbone. These homo-contacts between the carbon black primary particles limit the number of hetero-contacts, thereby reducing the maximum achievable Schmahl index. The KB-type hetero-aggregates showed clear segregation of carbon black and silica upon qualitative assessment. However, applying the Schmahl index to these segregated hetero-aggregates yielded unexpectedly high values close to perfect mixing. Due to a highly inhomogeneous material distribution, hetero-aggregates were synthesized in which only a single silica particle was present in the carbon black aggregate structure. Since single particles can not be clustered by definition, classical mixing theory leads to values close to one, which does not correspond to a well-mixed aggregate.

To overcome this limitation from classical mixing theory, in the this paper, the *inter-aggregate* variance in concentration and in mixing quality are taken into account to achieve a quantitative measure for the degree of segregation. A well-mixed hetero-aggregate system is characterized by a high mean *intra-aggregate* mixing quality (e.g., a Schmahl index close to one) and narrow *inter-aggregate* distributions in concentration and mixing quality. Individual measures of mixing quality are introduced that characterize each of these effects and a holistic measure is presented that encapsulates all effects. This allows for a thorough description of the mixing quality of a hetero-aggregate system by means of a single value. Moreover, the comparison between disparate experiments conducted under distinct process conditions, irrespective of primary particle size and concentration, is enabled. This paves the way for scientific exchange across a range of projects. Additionally, by assessing the scale of segregation via correlation coefficient functions, the length scale of silica clusters within the hetero-aggregate is determined.

The limitations of STEM-EDX elemental maps also impact the applicability of the methodology. A statistical evaluation necessitates costly and time-consuming measurements. For hetero-aggregates in which the materials exhibit distinct electron densities, less expensive HAADF-STEM images are a viable option [47]. Furthermore, false positive measurements are possible when two segregated clusters are located on top of each other in transmission. Therefore, further studies are needed to clarify how intensity and scale of segregation of hetero-aggregates are influenced by the 2D projection in STEM. In this context, the evaluation of artificially generated 3-dimensional hetero-aggregates and their corresponding 2D-projections are planned to gain further insights on hetero-aggregation phenomena.

In conclusion, a generally applicable methodology is presented for the characterization of mixing quality by the analysis of STEM-EDX elemental map data. Furthermore, the gap between *intra-* and *inter-aggregate* mixing quality is bridged through the introduction of a combined mixing quality measure. By including correlation coefficient functions, both the intensity and scale

of segregation are considered. Hence, the presented methodology facilitates a paradigm shift toward a more accurate and precise determination of the mixing quality of nanoscale hetero-aggregates.

## 5 | Experimental Section

**Synthesis of the Hetero-Aggregates** For the production of the hetero-aggregates two distinct process conditions with two different kinds of carbon black were chosen. In both cases, the carbon black was suspended in ethanol with silica particles. The silica particles were synthesized using the Stoober process [48]. The hetero-aggregates are then synthesized in a SpraySyn-burner by atomizing the suspension of carbon black and silica in ethanol in a spray flame. A more detailed description of the experimental setup is available in [16, 49]. Exemplary STEM-EDX elemental maps of the synthesized hetero-aggregates are depicted in Figure 1. For the first process conditions, in which segregation was observed, the hetero-aggregates were synthesized by dispersing 30 nm silica particles with Ketjenblack (KB) (Nanografi technology, Turkey) carbon black at a weight ratio of 1:1, with a mass concentration of 0.5 weight-% of carbon black with standard SpraySyn conditions (Figure 1a). For the second process conditions, the silica particles were finely dispersed on top of the carbon black backbone as depicted in Figure 1b. These hetero-aggregates were synthesized by dispersing TIMCAL Super C65 (Nanografi technology, Turkey) with 10 nm silica particles at a weight ratio of 3:1. The mass concentration of carbon black was 0.5 weight-% in ethanol and the synthesis was conducted with standard SpraySyn conditions. Accordingly, the hetero-aggregates synthesized at these process conditions are referred to as KB-type and C65-type respectively.

**Characterization** Number-based size distributions of the primary particles of carbon black were obtained by statistical analysis of HAADF-STEM (FEI Tecnai Osiris, FEI company, USA) images using ImageJ. The corresponding mean particle diameter of the silica particles were  $\bar{d} = 32.0 \pm 8.1$  for the KB-type hetero-aggregates and  $\bar{d} = 11.6 \pm 3.4$  for the C65-type hetero-aggregates. The means  $\pm$  one standard deviation of the number-based size distribution was averaged over a triple measurement, which was determined by dynamic light scattering (Zetasizer nano ZS, Malvern Panalytical, United Kingdom). The KB carbon black primary particles exhibit a median diameter of  $d_{50,0} = 15.45 \pm 4.42$  nm, whereas the C65 carbon black primary particles exhibit a median diameter of  $d_{50,0} = 39.6 \pm 11.58$ . The particle size distributions of all materials considered in this work are available in the [Supporting Information](#).

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## Conflicts of Interest

The authors declare no conflict of interest.

## Data Availability Statement

The data that support the findings of this study are openly available in [KITopen] at <https://doi.org/10.35097/ptpkymb8ueq97v5d>, reference number [1000181730]. The developed Python script is available in the Github repository [https://github.com/pdhs-group/STEM-EDX\\_mixing-quality](https://github.com/pdhs-group/STEM-EDX_mixing-quality).

## References

1. A. Camenzind, W. R. Caseri, and S. E. Pratsinis, "Flame-made Nanoparticles for Nanocomposites," *Nano Today* 5, no. 1 (2010): 48–65, <https://www.sciencedirect.com/science/article/pii/S1748013210000022>.
2. C. A. Gray and H. Muranko, "Studies of Robustness of Industrial Aciniform Aggregates and Agglomerates—Carbon Black and Amorphous Silicas: A Review Amplified by New Data," *Journal of Occupational and Environmental Medicine* 48, no. 12 (2006): 1279–1290, <http://www.jstor.org/stable/45009351>.
3. M. E. Spahr, D. Goers, A. Leone, S. Stallone, and E. Grivei, "Development of Carbon Conductive Additives for Advanced Lithium Ion Batteries," *Journal of Power Sources* 196, no. 7 (2011): 3404–3413, <https://www.sciencedirect.com/science/article/pii/S0378775310011420>.
4. A. K. Al-Kamal, M. Hammad, M. Y. Ali, et al., "Titania/Graphene Nanocomposites from Scalable Gas-Phase Synthesis for High-Capacity and High-Stability Sodium-Ion Battery Anodes," *Nanotechnology* 35, no. 22 (2024): 225602, <https://iopscience.iop.org/article/10.1088/1361-6528/ad2ac7/meta>.
5. V. Kolck, J. Witte, E. Schmidt, and H. Kruggel-Emden, "Analysis of Process Parameter Sensitivities of Jet-Based Direct Mixing Gas Phase Hetero-Agglomeration by DEM/CFD-Modelling," *Powder Technology* 429 (2023): 118963, <https://linkinghub.elsevier.com/retrieve/pii/S0032591023007465>.
6. M. J. Height, S. E. Pratsinis, O. Mekasuwanumrong, and P. Praserthdam, "Ag-ZnO Catalysts for UV-Photodegradation of Methylene Blue," *Applied Catalysis B: Environmental* 63, no. 3 (2006): 305–312, <https://www.sciencedirect.com/science/article/pii/S0926337305003887>.
7. J. Witte, Z. C. Hua, V. Kolck, H. Kruggel-Emden, S. Heinrich, and E. Schmidt, "Investigation of a Jet-Based Direct Mixing Process for Improved Structuring of Conductive Battery Hetero-Agglomerates," *Processes* 11, no. 11 (2023): 3243, <https://www.mdpi.com/2227-9717/11/11/3243>.
8. W. An, J. Fu, J. Su, et al., "Mesoporous Hollow Nanospheres Consisting of Carbon Coated Silica Nanoparticles for Robust Lithium-Ion Battery Anodes," *Journal of Power Sources* 345 (2017): 227–236, <https://www.sciencedirect.com/science/article/pii/S0378775317301271>.
9. L. Cao, J. Huang, Z. Lin, et al., "Amorphous SiO<sub>2</sub>/c Composite as Anode Material for Lithium-Ion Batteries," *Journal of Materials Research* 33, no. 9 (2018): 1219–1225, <https://www.cambridge.org/core/journals/journal-of-materials-research/article/amorphous-sio2c-composite-as-anode-material-for-lithiumion-batteries/E0133170CB56BEC3B2E7D8C86CD29013>.
10. F. Frankenberg, M. Kissel, C. F. Burmeister, M. Lippke, J. Janek, and A. Kwade, "Investigating the Production of All-Solid-State Battery Composite Cathodes by Numerical Simulation of the Stressing Conditions in a High-Intensity Mixer," *Powder Technology* 435 (2024): 119403, <https://linkinghub.elsevier.com/retrieve/pii/S0032591024000457>.
11. E. Asylbekov, J. Mayer, H. Nirschl, and A. Kwade, "Modeling of Carbon Black Fragmentation During High-Intensity Dry Mixing Using the Population Balance Equation and the Discrete Element Method," *Energy Technology* 11 (2022): 2200867, <https://onlinelibrary.wiley.com/doi/full/10.1002/ente.202200867>.
12. H. Bockholt, W. Haselrieder, and A. Kwade, "Intensive Powder Mixing for Dry Dispersing of Carbon Black and Its Relevance for Lithium-Ion Battery Cathodes," *Powder Technology* 297 (2016): 266–274, <https://www.sciencedirect.com/science/article/pii/S003259101630170X>.
13. Y. Yao, J. Zhang, L. Xue, T. Huang, and A. Yu, "Carbon-Coated SiO<sub>2</sub> Nanoparticles as Anode Material for Lithium-Ion Batteries," *Journal of Power Sources* 196, no. 23 (2011): 10240–10243, <https://www.sciencedirect.com/science/article/pii/S0378775311015072>.
14. J. Meng, Y. Cao, Y. Suo, Y. Liu, J. Zhang, and X. Zheng, "Facile Fabrication of 3D SiO<sub>2</sub>@graphene Aerogel Composites as Anode Material for Lithium Ion Batteries," *Electrochimica Acta* 176 (2015): 1001–1009, <https://www.sciencedirect.com/science/article/pii/S0013468615302085>.
15. P. Lv, H. Zhao, J. Wang, X. Liu, T. Zhang, and Q. Xia, "Facile Preparation and Electrochemical Properties of Amorphous SiO<sub>2</sub>/c Composite as Anode Material for Lithium Ion Batteries," *Journal of Power Sources* 237 (2013): 291–294, <https://www.sciencedirect.com/science/article/pii/S037877531300445X>.
16. S. Buchheiser, F. Kistner, F. Rhein, and H. Nirschl, "Spray Flame Synthesis and Multiscale Characterization of Carbon Black-silica Hetero-aggregates," *Nanomaterials* 13, no. 12 (2023): 1893, <https://www.mdpi.com/2079-4991/13/12/1893>.
17. J. Stahl, C. Mahr, B. Gerken, S. Pokhrel, A. Rosenauer, and L. Mädler, "Quantification of the Hetero-contact Formation Process for a CuO/CeO<sub>2</sub> Hetero-aggregate Model System Prepared by Double Flame Spray Pyrolysis," *Powder Technology* 448 (2024): 120305, <https://linkinghub.elsevier.com/retrieve/pii/S0032591024009495>.
18. J. Men, S. R. Kolan, A. Massomi, et al., "Formulation of Nanostructured Heteroaggregates by Fluidization Technologies," *Chemie Ingenieur Technik* 95, no. 1 (2023): 107–113, <https://onlinelibrary.wiley.com/doi/full/10.1002/cite.202200139>.
19. M. Gäßler, J. Stahl, M. Schowalter, et al., "The Impact of Support Material of Cobalt-based Catalysts Prepared by Double Flame Spray Pyrolysis on CO<sub>2</sub> Methanation Dynamics," *ChemCatChem* 14, no. 16 (2022): 202200286, <https://chemistry-europe.onlinelibrary.wiley.com/doi/full/10.1002/cctc.202200286>.
20. J. Stahl, J. Ilsemann, S. Pokhrel, et al., "Comparing Co-Catalytic Effects of ZrO<sub>x</sub>, SmO<sub>x</sub>, and Pt on CO<sub>x</sub> Methanation over Co-Based Catalysts Prepared by Double Flame Spray Pyrolysis," *ChemCatChem* 13, no. 12 (2021): 2815–2831, <https://chemistry-europe.onlinelibrary.wiley.com/doi/10.1002/cctc.202001998>.
21. R. Strobel, "Flame spray synthesis of Pd/Al<sub>2</sub>O<sub>3</sub> Catalysts and Their Behavior in Enantioselective Hydrogenation," *Journal of Catalysis* 222, no. 2 (2004): 307–314, <https://www.sciencedirect.com/science/article/pii/S0021951703004238>.
22. M. Iwao, H. Miyamoto, H. Nakamura, E. Hayakawa, S. Ohsaki, and S. Watano, "Continuous Compositing Process of Sulfur/Conductive-Additive Composite Particles for All-Solid-State Lithium-Sulfur Batteries," *Advanced Energy and Sustainability Research* 5, no. 8 (2024): 2200206, <https://onlinelibrary.wiley.com/doi/10.1002/aesr.202200206>.
23. V. Baric, H. K. Grossmann, W. Koch, and L. Mädler, "Quantitative Characterization of Mixing in Multicomponent Nanoparticle Aggregates," *Particle & Particle Systems Characterization* 35, no. 10 (2018): 1800177, <https://onlinelibrary.wiley.com/doi/10.1002/ppsc.201800177>.
24. B. Gerken, C. Mahr, J. Stahl, et al., "Material Discrimination in Nanoparticle Hetero-Aggregates by Analysis of Scanning Transmission Electron Microscopy Images," *Particle & Particle Systems Characterization* 40 (2023): 2300048, <https://onlinelibrary.wiley.com/doi/full/10.1002/ppsc.202300048>.
25. L. Fuchs, T. Kirstein, C. Mahr, et al., "Using Convolutional Neural Networks for Stereological Characterization of 3D Hetero-Aggregates Based on Synthetic STEM Data," *Machine Learning: Science and Technology* 5 (2024): 025007, <https://iopscience.iop.org/article/10.1088/2632-2153/ad38fd>.
26. M. Camesasca and I. Manas-Zloczower, "Danckwerts Revisited—The Use of Entropy to Define Scale and Intensity of Segregation," *Macromolecular Theory and Simulations* 18, no. 2 (2009): 87–96, <https://onlinelibrary.wiley.com/doi/10.1002/mats.200800068>.

27. W. Wang, I. Manas-Zloczower, and M. Kaufman, "Characterization of Distributive Mixing in Polymer Processing Equipment Using Renyi Entropies," *International Polymer Processing* 16, no. 4 (2022): 315–322, <https://www.degruyter.com/document/doi/10.1515/ipp-2001-0002/html>.

28. T. G. Williams, K. Kreckel, F. Belfiore, et al., "The 2D Metallicity Distribution and Mixing Scales of Nearby Galaxies," *Monthly Notices of the Royal Astronomical Society* 509, no. 1 (2021): 1303–1322.

29. V. C. Angadi, F. Benz, I. Tischer, K. Thonke, T. Aoki, and T. Walther, "Evidence of Terbium and Oxygen Co-Segregation in Annealed AlN:Tb," *Applied Physics Letters* 110, no. 22 (2017).

30. H. Wiinikka, F. S. Hage, Q. M. Ramasse, and P. Toth, "Spatial Distribution of Metallic Heteroatoms in Soot Nanostructure Mapped by Aberration-Corrected STEM-EELS," *Carbon* 173 (2021): 953–967.

31. S. R. Kolan, R. Wang, T. Hoffmann, and E. Tsotsas, "Mixing Sub-Micron Particles in a ProCell Type Spouted Bed," *Powder Technology* 428 (2023): 118828, <https://www.sciencedirect.com/science/article/pii/S0032591023006125>.

32. B. Daumann, J. A. Weber, H. Anlauf, and H. Nirschl, "Discontinuous Powder Mixing of Nanoscale Particles," *Chemical Engineering Journal* 167, no. 1 (2011): 377–387, <https://linkinghub.elsevier.com/retrieve/pii/S1385894710012672>.

33. F. Matteocci, Y. Busby, J.-J. Pireaux, et al., "Interface and Composition Analysis on Perovskite Solar Cells," *ACS Applied Materials & Interfaces* 7, no. 47 (2015): 26176–26183.

34. The GIMP Development Team, "Gnu Image Manipulation Program (GIMP), Version 3.0.2. Community, Free Software (License GPLv3)," 2025, version 3.0.2, Free Software, <https://gimp.org/>.

35. K. Stange, "Die Mischgüte einer Zufallsmischung als Grundlage zur Beurteilung von Mischversuchen," *Chemie Ingenieur Technik* 26, no. 6 (1954): 331–337, <https://onlinelibrary.wiley.com/doi/10.1002/cite.330260607>.

36. K. Sommer and H. Rumpf, "Varianz der stochastischen Homogenität bei Körnermischungen und Suspensionen und praktische Ermittlung der Mischgüte," *Chemie Ingenieur Technik* 46, no. 6 (1974): 257–257, <https://onlinelibrary.wiley.com/doi/10.1002/cite.330460609>.

37. G. Schmahl, "Der Homogenitätsgrad pastöser Mischungen beim diskontinuierlichen Kneten," *Chemie Ingenieur Technik* 41, no. 5 (1969): 359–363, <https://onlinelibrary.wiley.com/doi/10.1002/cite.330410528>.

38. J. Bridgwater, "Mixing of Powders and Granular Materials by Mechanical Means—A Perspective," *Particuology* 10, no. 4 (2012): 397–427.

39. P. V. Danckwerts, "The Definition and Measurement of Some Characteristics of Mixtures," *Applied Scientific Research* 3, no. 4 (1952): 279–296, <http://link.springer.com/10.1007/BF03184936>.

40. P. Lacey, "The Mixing of Solid Particles," *Chemical Engineering Research and Design* 75 (1997): S49–S55.

41. M. Ashton and F. Valentin, "The Mixing of Powders and Particles in Industrial Mixers," *Transactions of the Institution of Chemical Engineers* 44, no. 5 (1966): 166–188.

42. L. V. Saraf, "Imaging Chemical Aggregation of Ni/NiO Particles from Reduced NiO-YSZ," *Electrochemical and Solid-State Letters* 14, no. 10 (2011): B100, <https://doi.org/10.1149/1.3610426>.

43. T. Hastie, R. Tibshirani, and J. Friedman, *The Elements of Statistical Learning: Data Mining, Inference, and Prediction* (Springer, 2009).

44. M. Neumann, P. Gräfensteiner, C. Santos de Oliveira, et al., "Morphology of Nanoporous Glass: Stochastic 3D Modeling, Stereology and the Influence of Pore Width," *Physical Review Materials* 8, no. 4 (2024): 045605.

45. S. N. Chiu, D. Stoyan, W. S. Kendall, and J. Mecke, *Stochastic Geometry and its Applications* (J. Wiley & Sons, 2013).

46. P. Virtanen, R. Gommers, T. E. Oliphant, et al., "SciPy 1.0: Fundamental Algorithms for Scientific Computing in Python," *Nature Methods* 17 (2020): 261–272.

47. C. Mahr, J. Stahl, B. Gerken, et al., "Characterization of Mixing in Nanoparticle Hetero-Aggregates by Convolutional Neural Networks," *Nano Select* 5, no. 4 (2024-04): 2300128, <https://onlinelibrary.wiley.com/doi/10.1002/nano.202300128>.

48. W. Stóber, A. Fink, and E. Bohn, "Controlled Growth of Monodisperse Silica Spheres in the Micron Size Range," *Journal of Colloid and Interface Science* 26, no. 1 (1968): 62–69.

49. F. Schneider, S. Suleiman, J. Menser, et al., "SpraySyn-A Standardized Burner Configuration for Nanoparticle Synthesis in Spray Flames," *The Review of Scientific Instruments* 90, no. 8 (2019): 085108.

## Supporting Information

Additional supporting information can be found online in the Supporting Information section.

Supporting Information