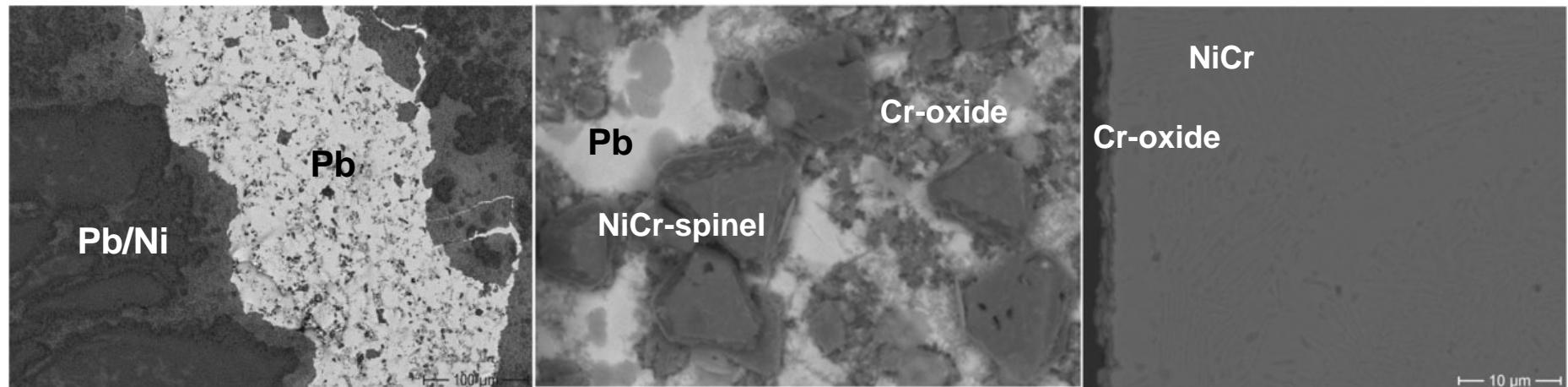


Oxidation and dissolution of binary Ni-Cr and ternary Ni-Cr-Al alloys in stagnant liquid lead at 750°C

O. Picho, C. Schroer, V. Trouillet and J. Konys



Application of Lead Technology to Subcritical (ADS) and Critical (LFR) Nuclear Systems/Reactors

LFR

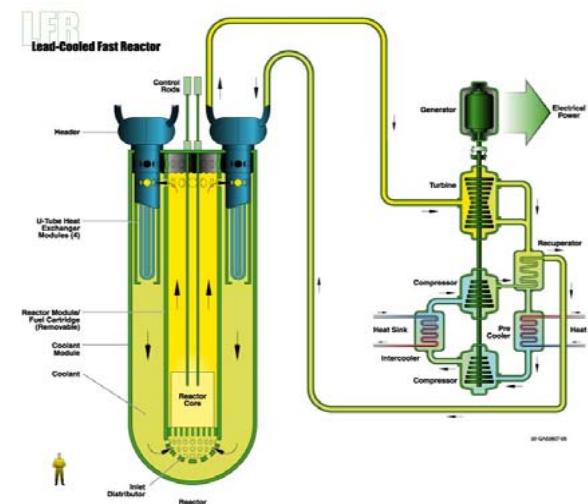
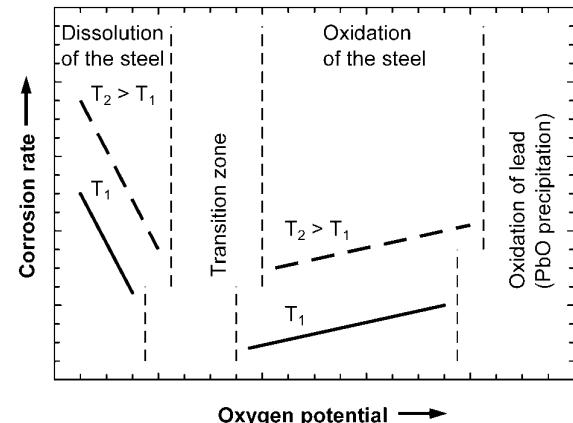
- One of the concepts for the 4th generation of nuclear power plants (Gen IV)
- In the long-term, Pb as primary coolant at maximum 800°C
- Short- to mid-term: Pb- or LBE-cooled at 450 – 550°C

ADS

- Transmutation of nuclear waste
- Power generation
- Liquid lead (Pb) or lead-bismuth eutectic (LBE) as spallation target and primary coolant
- Maximum temperature, typically
- 450 – 550°C

Corrosion of structure materials/ steels

- Dissolution of steel constituents
- Minimized by adding oxygen to liquid metal/ oxidation

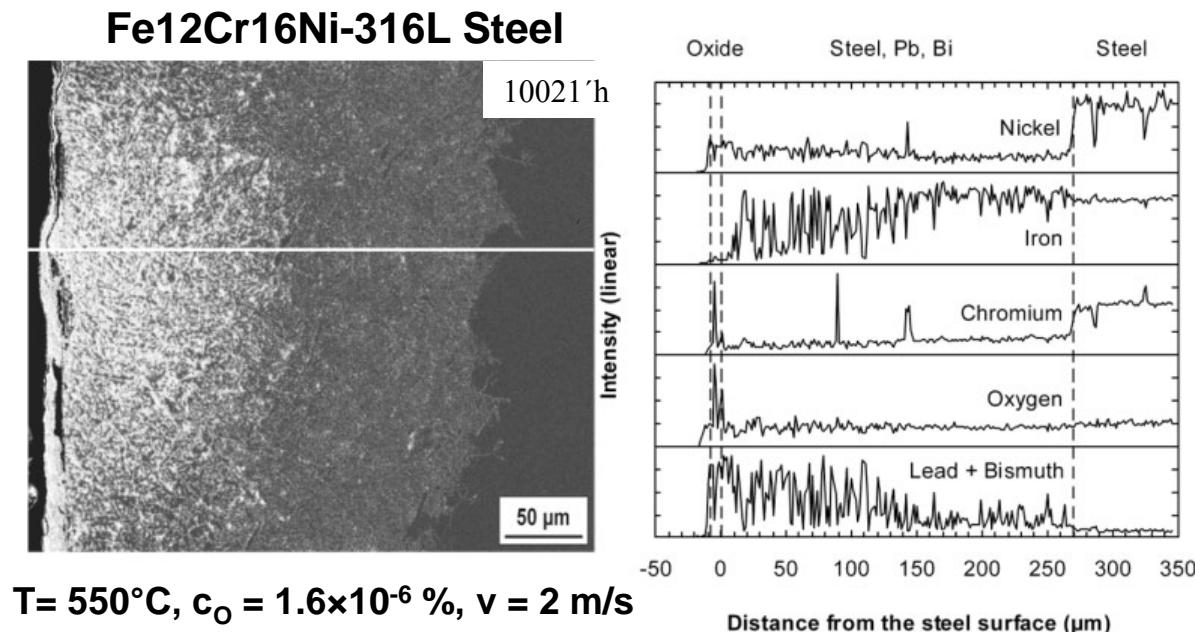


Motivation

At $T > 600^\circ\text{C}$ Ni-based alloys because of appropriate thermomechanical properties

BUT:

- Selective leaching of Nickel
- Penetration of liquid Pb into the material

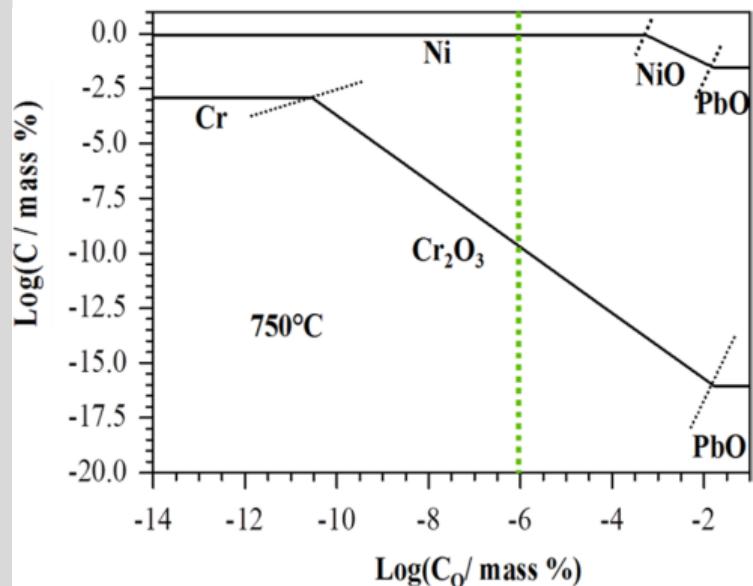


Which Cr-content can suppress Ni-dissolution ?

Material

Elements	Ni	Ni25Cr	Ni30Cr	Ni35Cr	Ni48Cr	Ni35Cr1Al	Ni35Cr3Al	Ni35Cr5Al
Ni (mass%)	99,9	74,36	69,36	64,43	52,39	63,74	61,74	60,31
Cr (mass%)		25,33	30,35	35,25	48,01	35,10	35,20	34,64
Al (mass%)		0,31	0,29	0,32	0,30	1,10	3,05	5,04
Ti (mass%)					0,30			

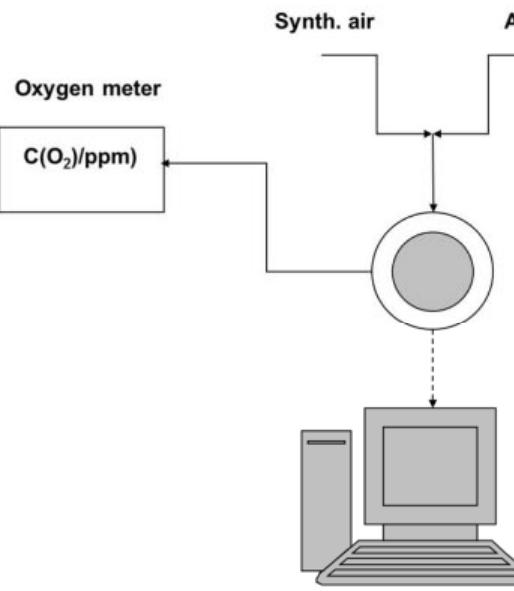
Experimental conditions



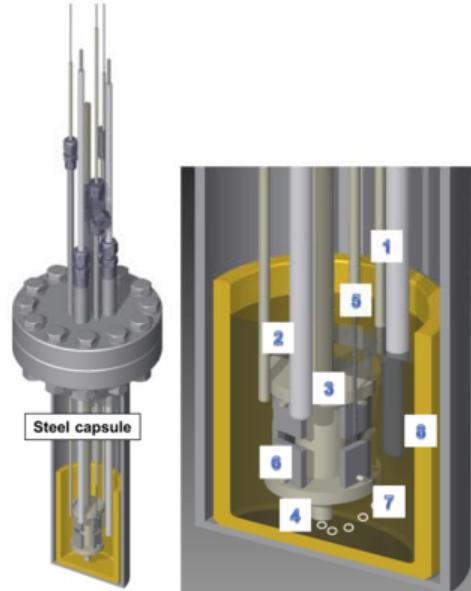
- Static liquid lead at 750°C - as basic for modelling of corrosion behavior
- Co=10⁻⁶ mass % - corresponds to typical reactor operating conditions

Corrosion test apparatus

Gas inlet system

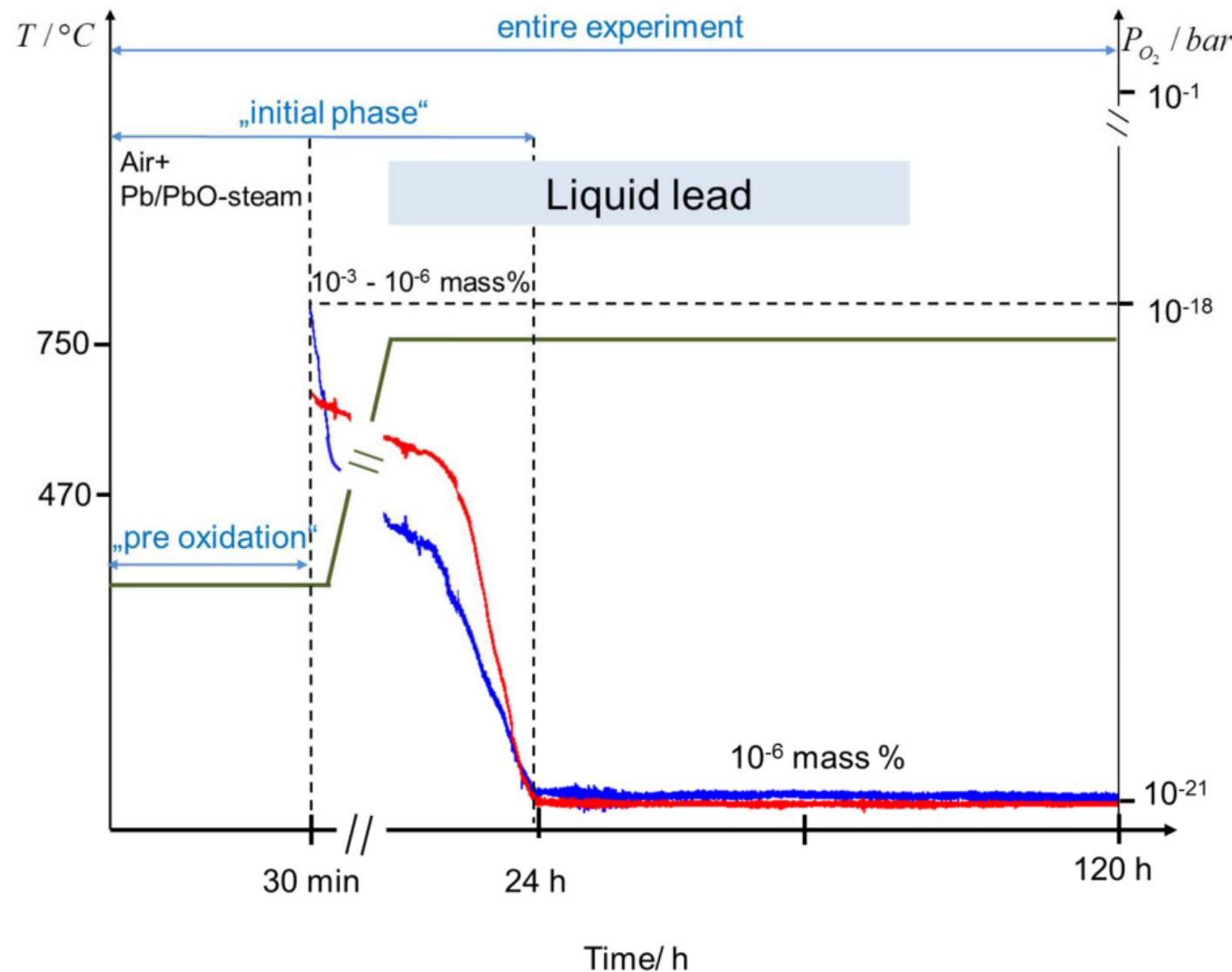


Experimental capsule
in the furnace



1. O₂-sensor + thermocouple on the 1st level
2. O₂-sensor + thermocouple on the 2nd level
3. Sample holder (Al₂O₃)
4. Gas bubbling tube (Al₂O₃)
5. Mo-electrode
6. Specimens (15x10x2 mm)
7. Oxygen containing liquid lead
8. Crucible (Al₂O₃)

Experimental procedure

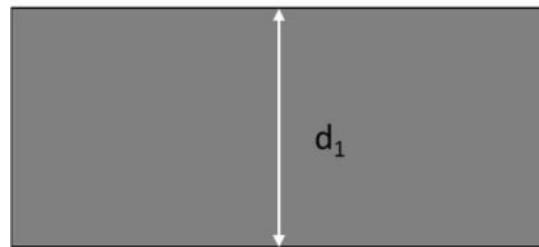


Quantification of specimens bevor and after tests

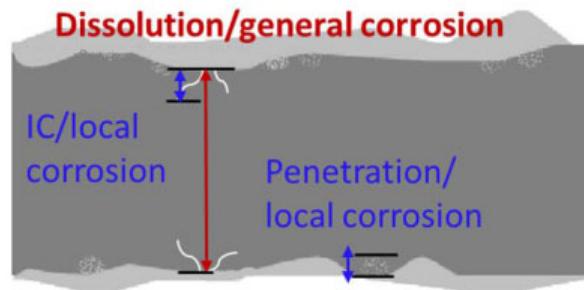
Material loss

Cross-section

as received



after exposure in Pb



$$\Delta x = (d_1 - d_2)/2$$

Surface

- XPS
 - „pre oxidation“ 30 min



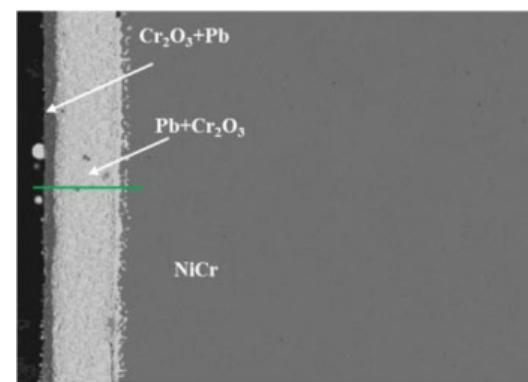
- SEM/EDS
 - „Initial phase“/ 24 h
 - 120 h



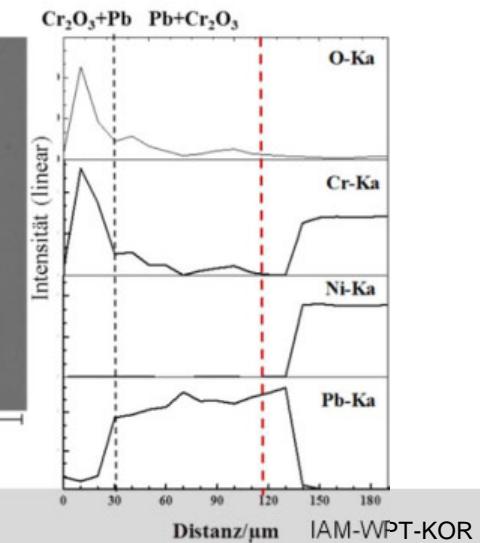
d_2

SEM/EDS

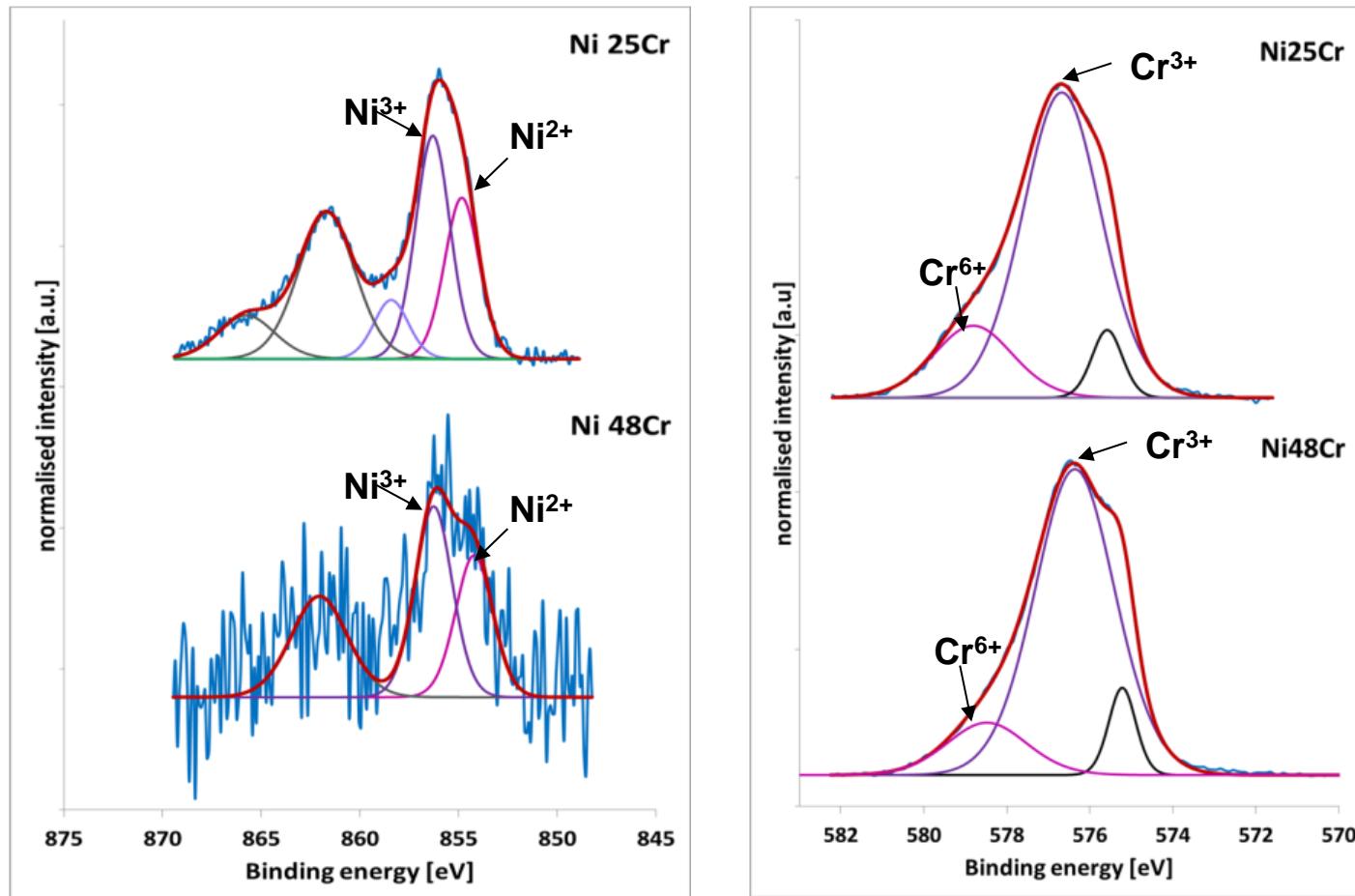
- „Initial phase“/24 h
- 120 h



Cross-section



XPS-Analysis from binary Ni-Cr-alloys with 25 and 48 mass% of Cr



- Ni25Cr has higher concentration of Ni-Oxide on the material surface
- Both materials provide a thin Cr-oxide film

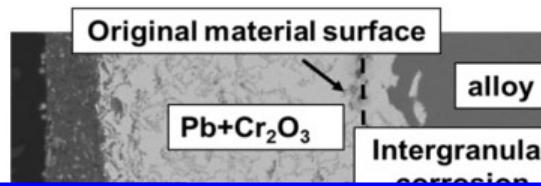
Binary Ni-Cr alloys

Ni30Cr

Initial Phase/ 24 h

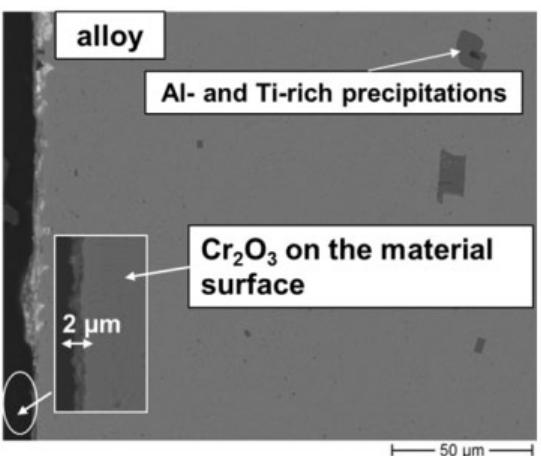
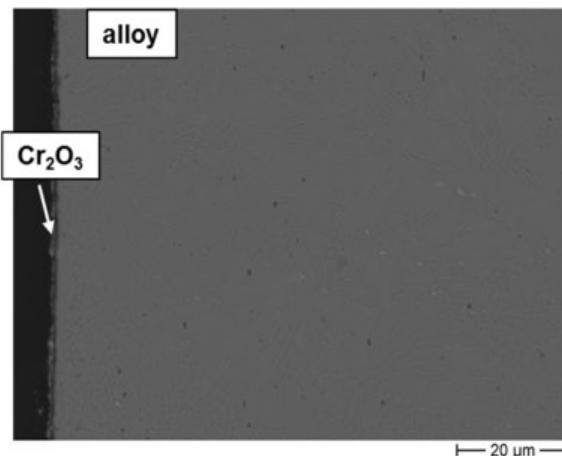
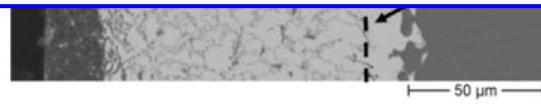


120 h

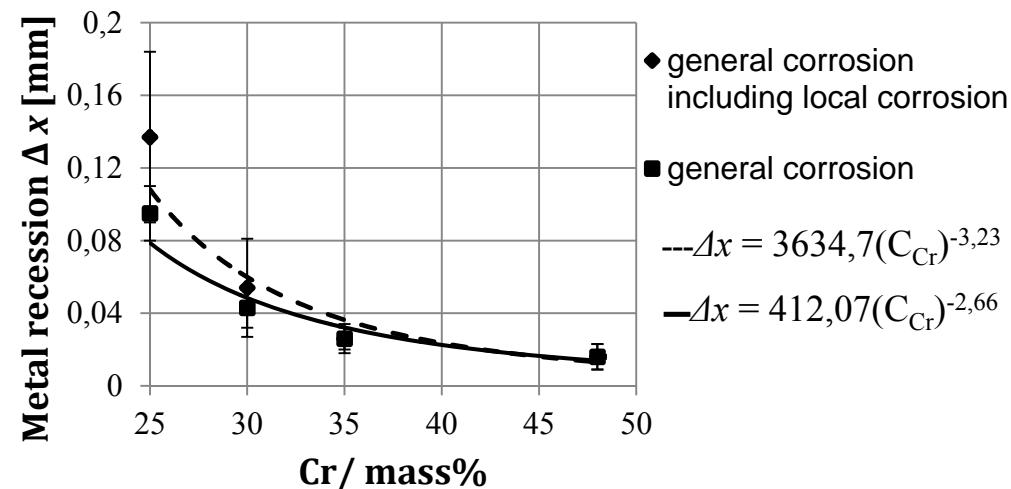


Typical corrosion scale for the binary alloys
with 25-35 mass% of Cr

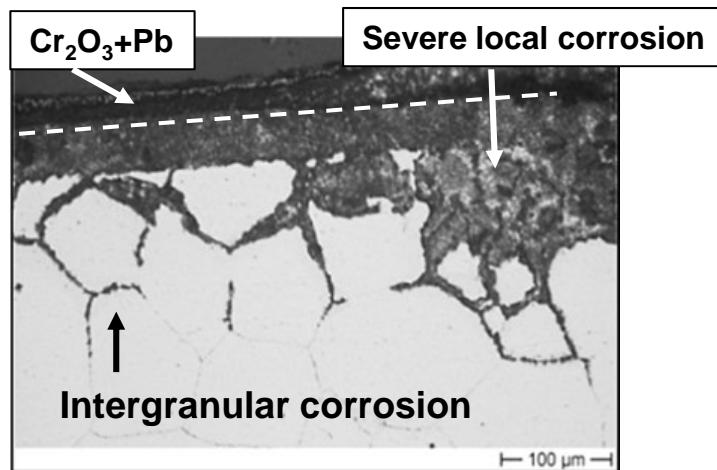
Ni48Cr



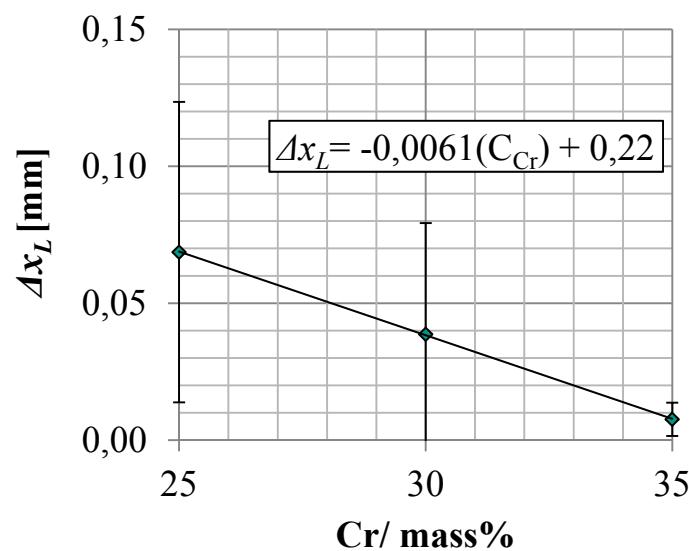
Material degradation



Ni25Cr



Intergranular corrosion



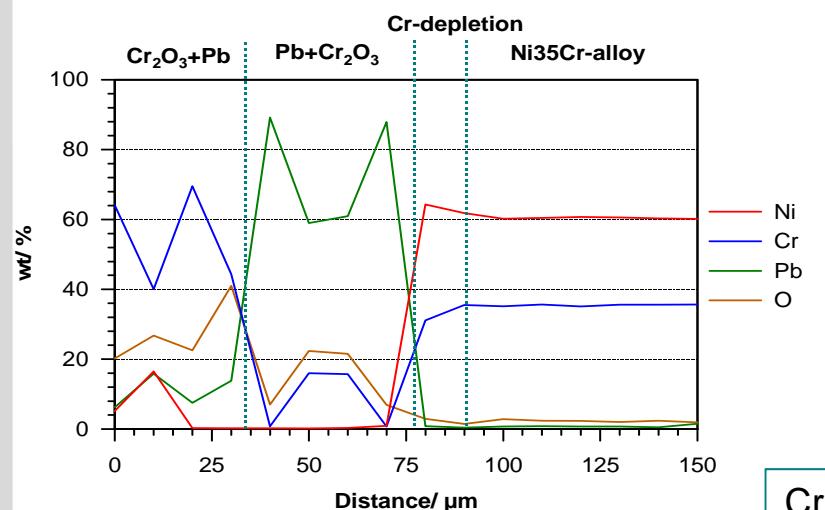
Analysis of oxidation and dissolution behavior of binary NiCr-alloys

Cr-Oxidation

$$m_{Cr_{depletion}} / \frac{g}{cm^2} = \rho_{NiCr} (\chi_{Cr}^0 - \chi_{Cr}^*) x_{deplet}$$

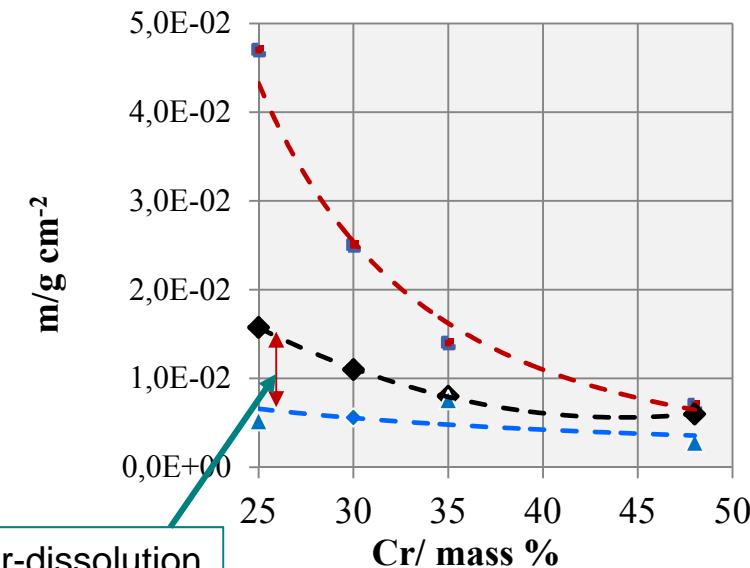
$$m_{Cr_{consumption}} / \frac{g}{cm^2} = \rho_{NiCr} \chi_{Cr}^0 x_{consum}$$

$$m_{Cr_{oxidation}} / \frac{g}{cm^2} = \rho_{Cr_2O_3} \chi_{Cr_2O_3} x_{oxide}$$



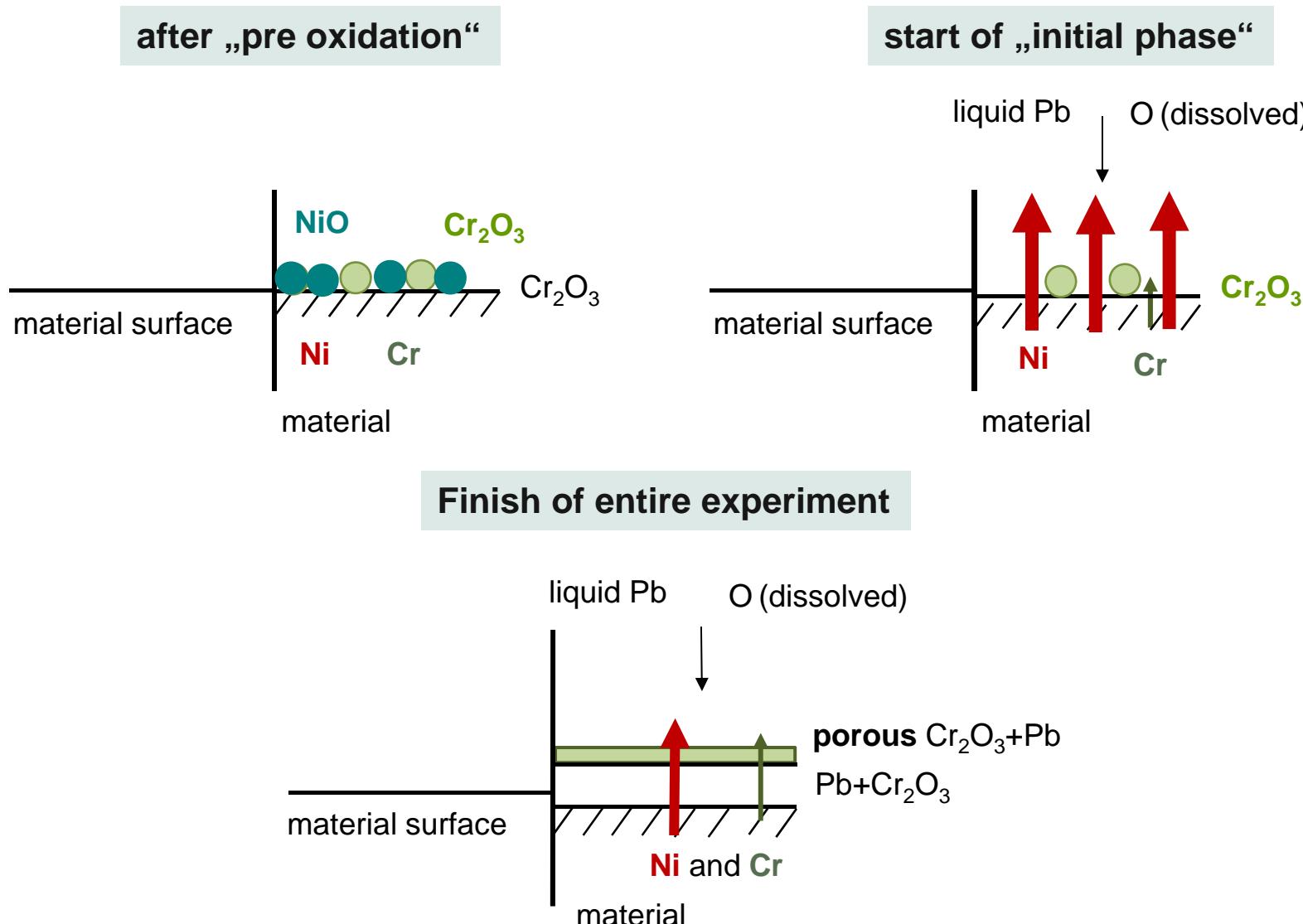
Ni-Dissolution

$$m_{Ni_{dissolution}} / \frac{g}{cm^2} = \rho_{NiCr} \chi_{Ni}^0 x_{dissolved}$$



Corrosion mechanism of binary Ni-Cr-alloys

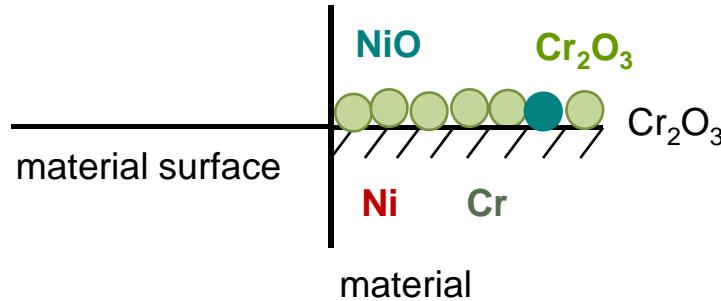
Ni25-35Cr



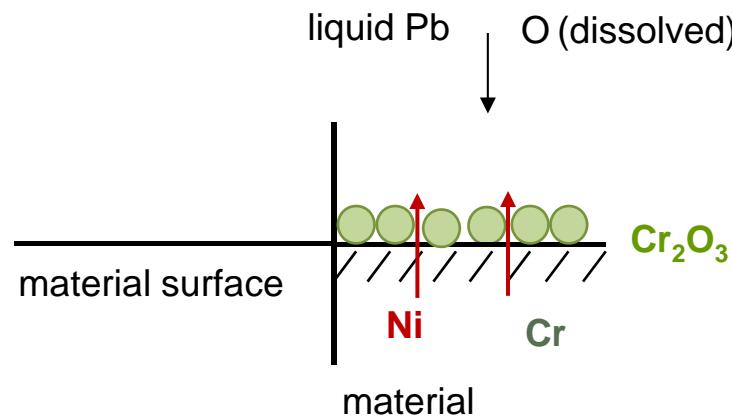
Corrosion mechanism of binary Ni-Cr-alloys

Ni48Cr

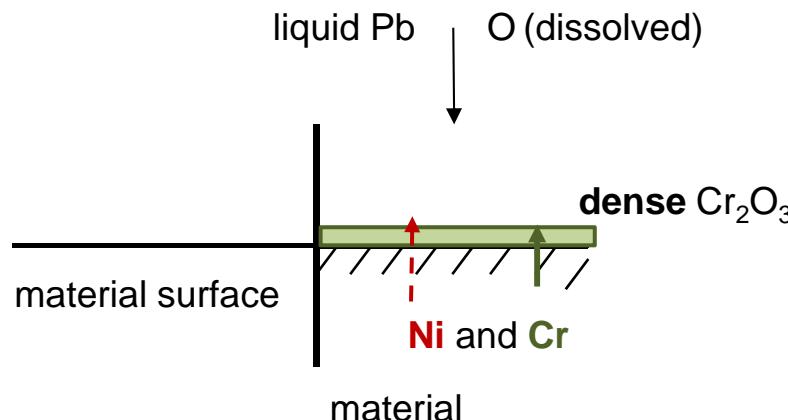
after „pre oxidation“



start of „initial phase“



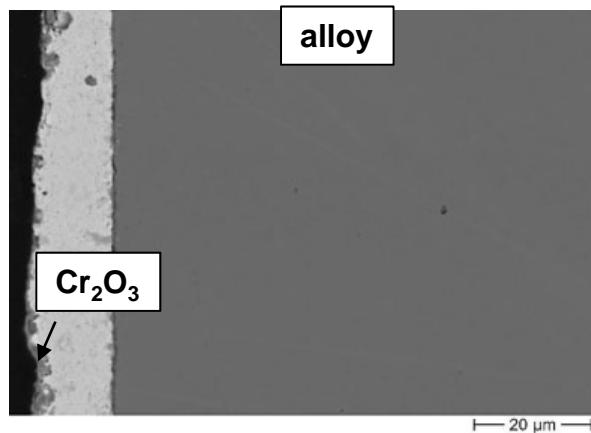
Finish of entire experiment



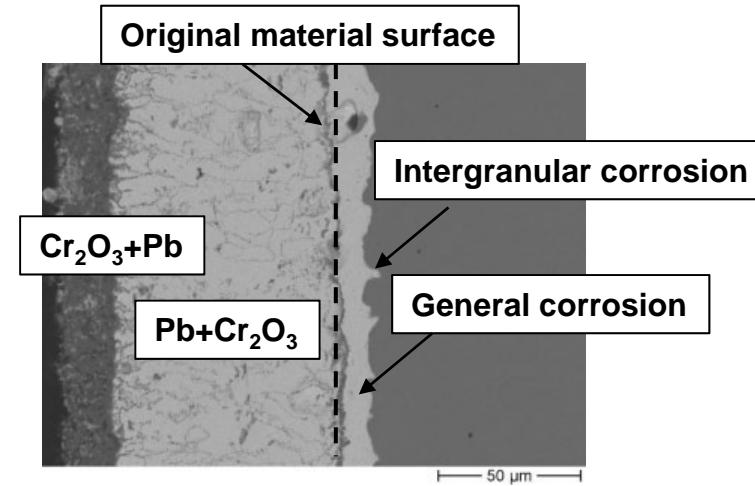
Ternary Ni35Cr1-5Al alloy

Ni35Cr

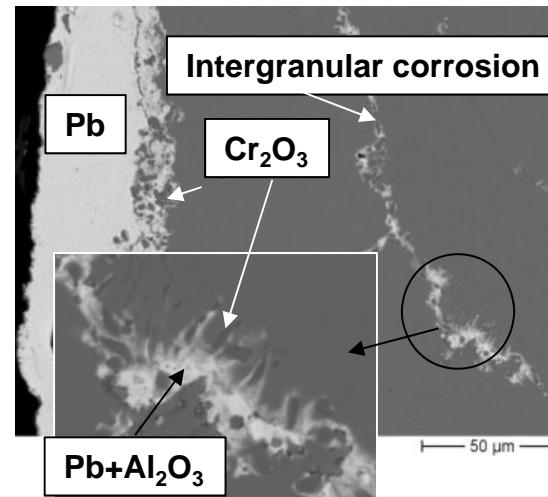
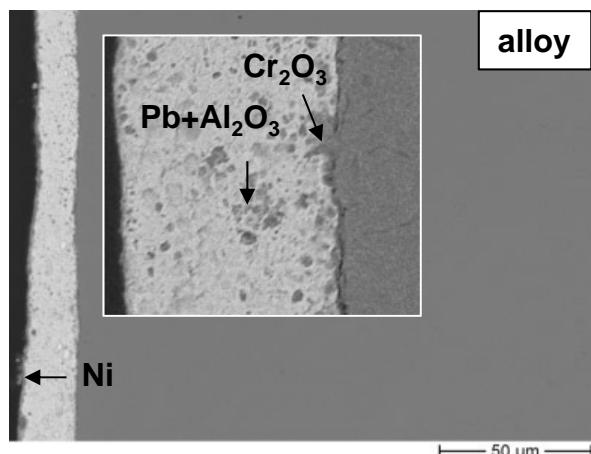
Initial Phase/ 24 h



120 h



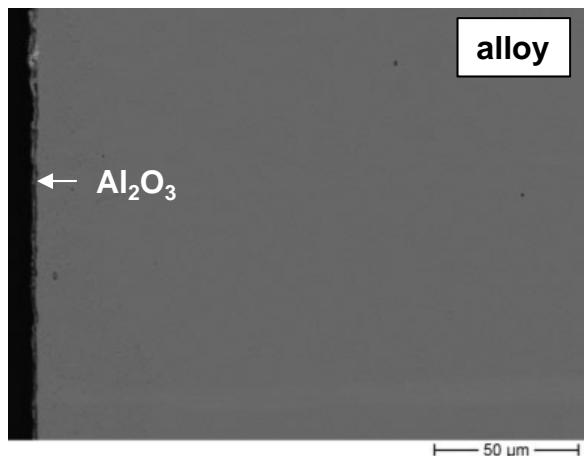
Ni35Cr1Al



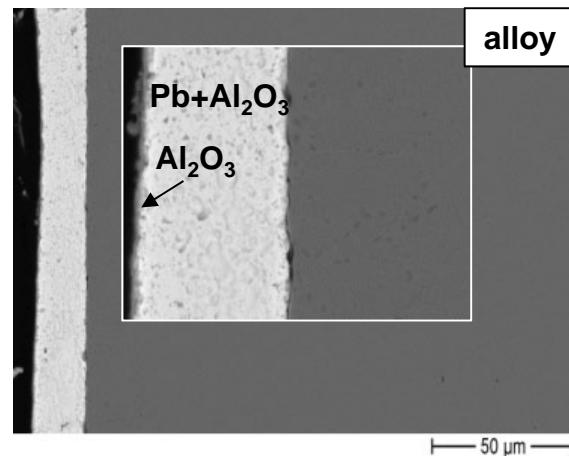
Ternary Ni35Cr1-5Al alloy

Ni35Cr3Al

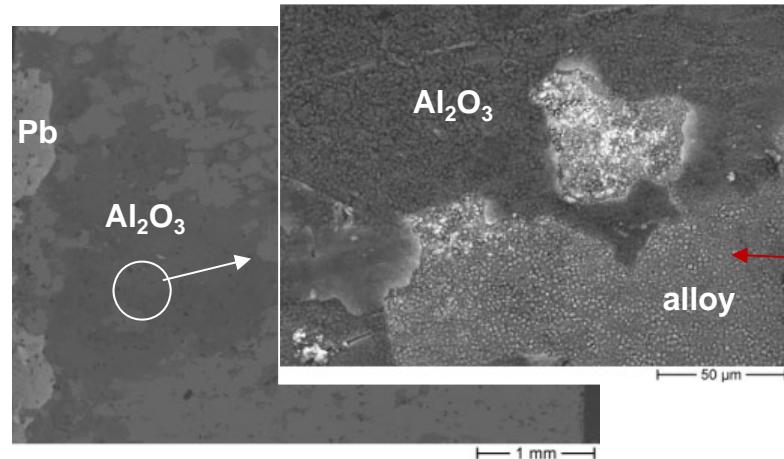
Initial Phase/ 24 h



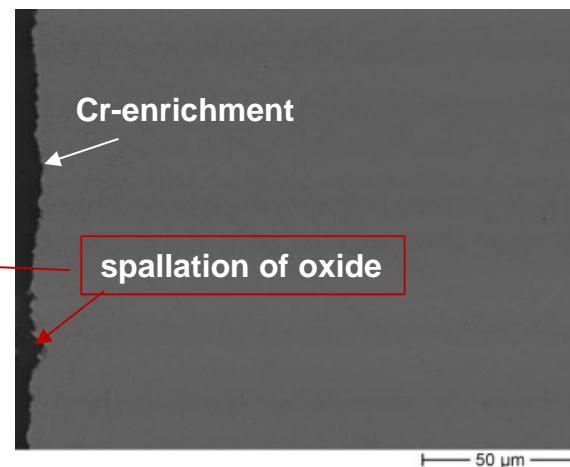
120 h



surface/ 120 h



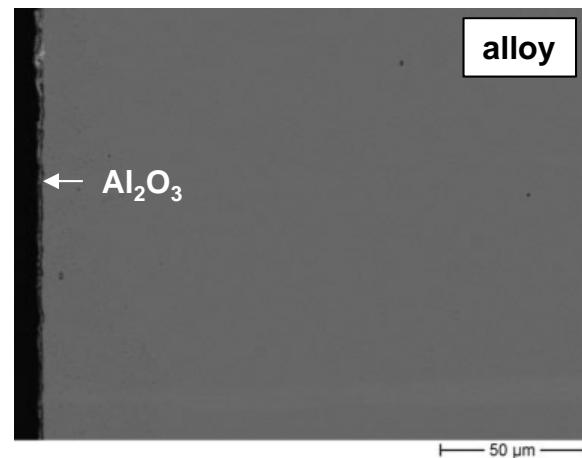
120 h



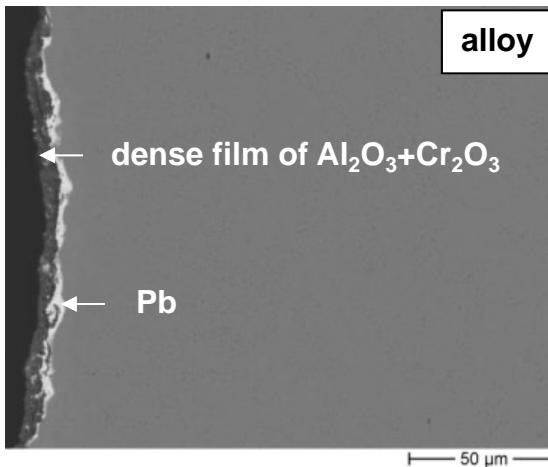
Ternary Ni35Cr1-5Al alloy

Ni35Cr5Al

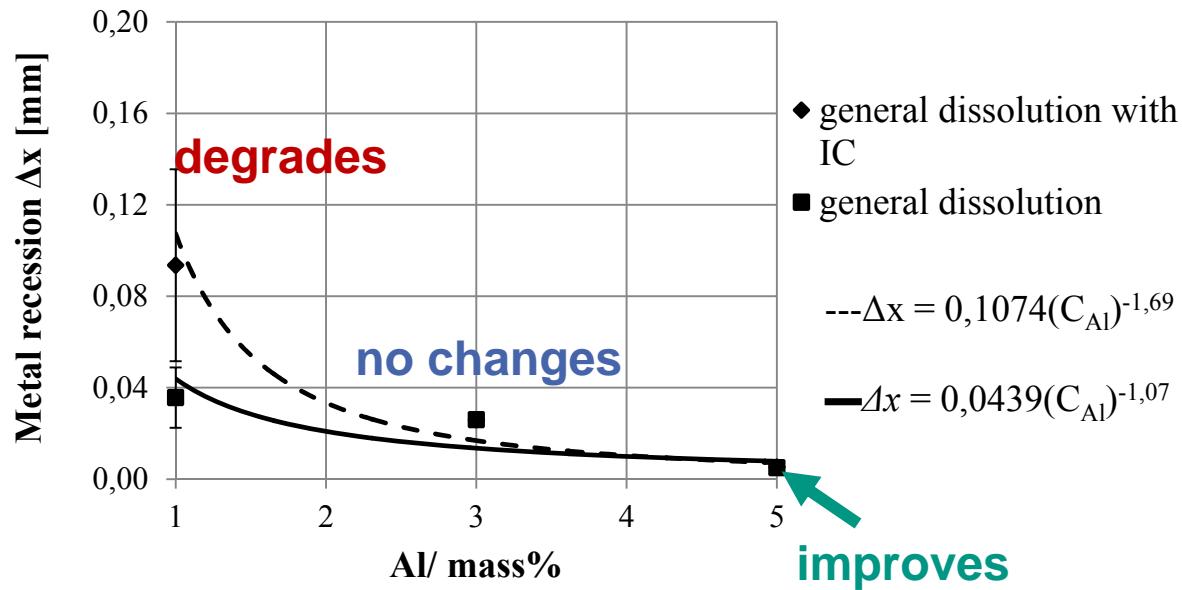
Initial Phase/ 24 h



120 h



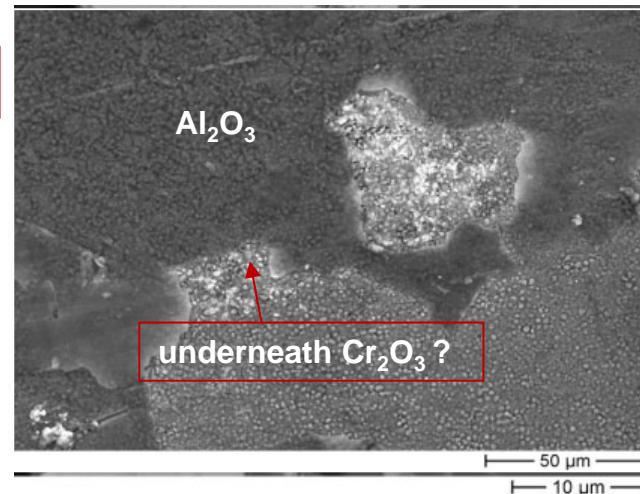
Material loss depending on Al-content



Obstruction of Cr-oxidation on the material surface



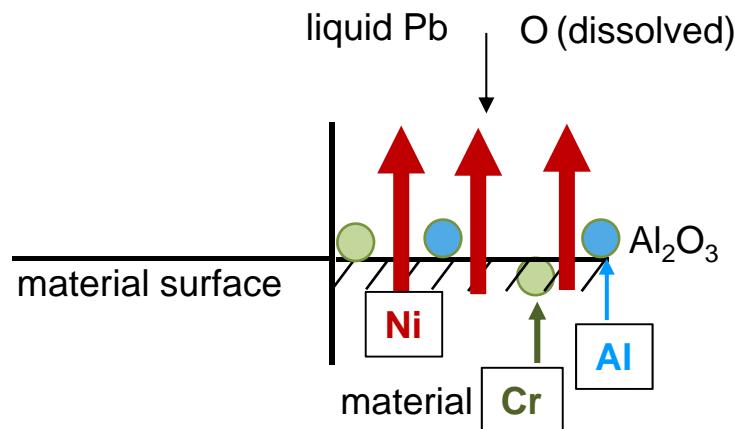
Low adhesion of oxide film



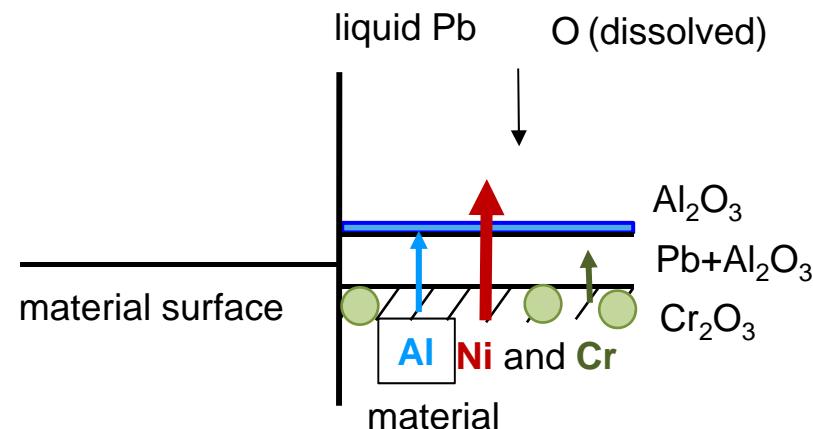
Corrosion mechanism on ternary Ni35Cr-alloys

Ni35Cr1-3Al

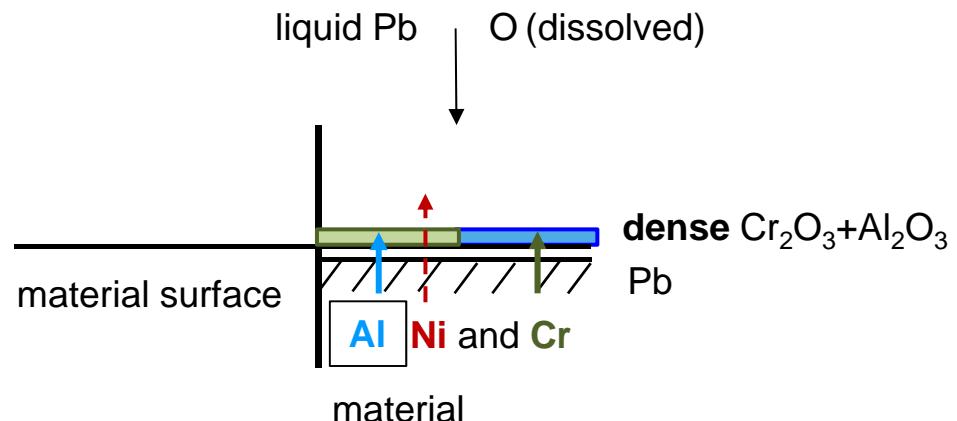
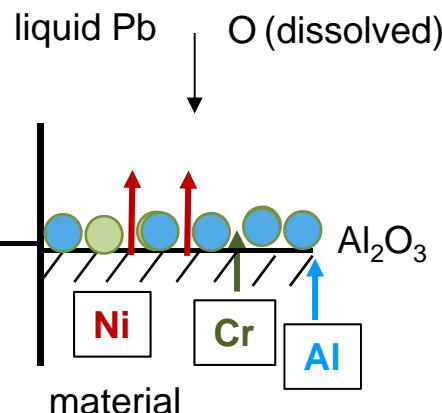
start



finish



Ni35Cr5Al



Conclusions

1. The significant corrosion of binary (Ni25-35Cr mass%) and ternary Ni-Cr-alloys (1-3 mass%) could be explained by:
 - a) formation of porous oxide film away from material surface
 - b) intergranular corrosion
 - c) spallation of oxide film
2. This is supported by high Ni and Cr-dissolution instead of only oxidation of Cr. Also addition of less noble Al leads to a decrease of Cr-oxidation, but the Al-content is not high enough to form stable Al_2O_3 .
3. Increasing of Cr (48 mass%) and Al-content (5 mass%) improves corrosion performance with respect to close position of the dense oxide layer to the material surface.
4. Controlling of dissolution process (reduction of metal recession) by addition of alloying elements is a main issue for further investigation of both materials.

Thanks for your attention

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Programme within the cross-cutting project
GETMAT (contract no.FP7-212175) is
gratefully acknowledged.