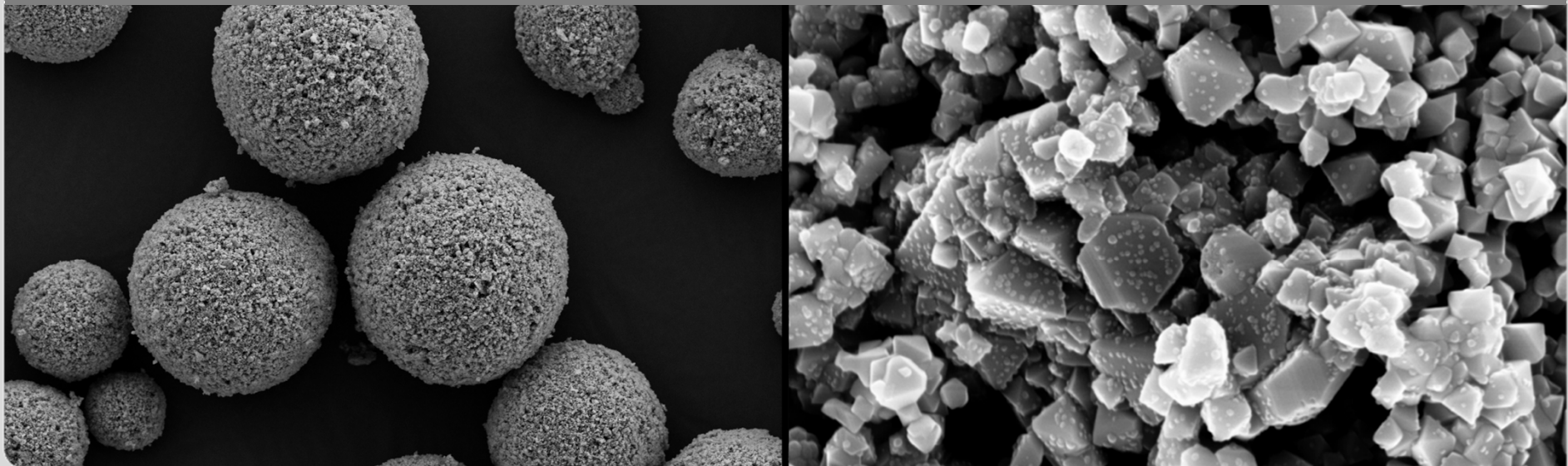


Surface Modification of Nanoscale $\text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_4$ with (Lithium) Metal Fluorides for Application as Cathode Material in Lithium-Ion Batteries

A. Höweling, C. Bühler, G. Lieser, S. Glatthaar and J.R. Binder

Institute for Applied Materials (IAM-WPT)



Fundamentals and Motivation

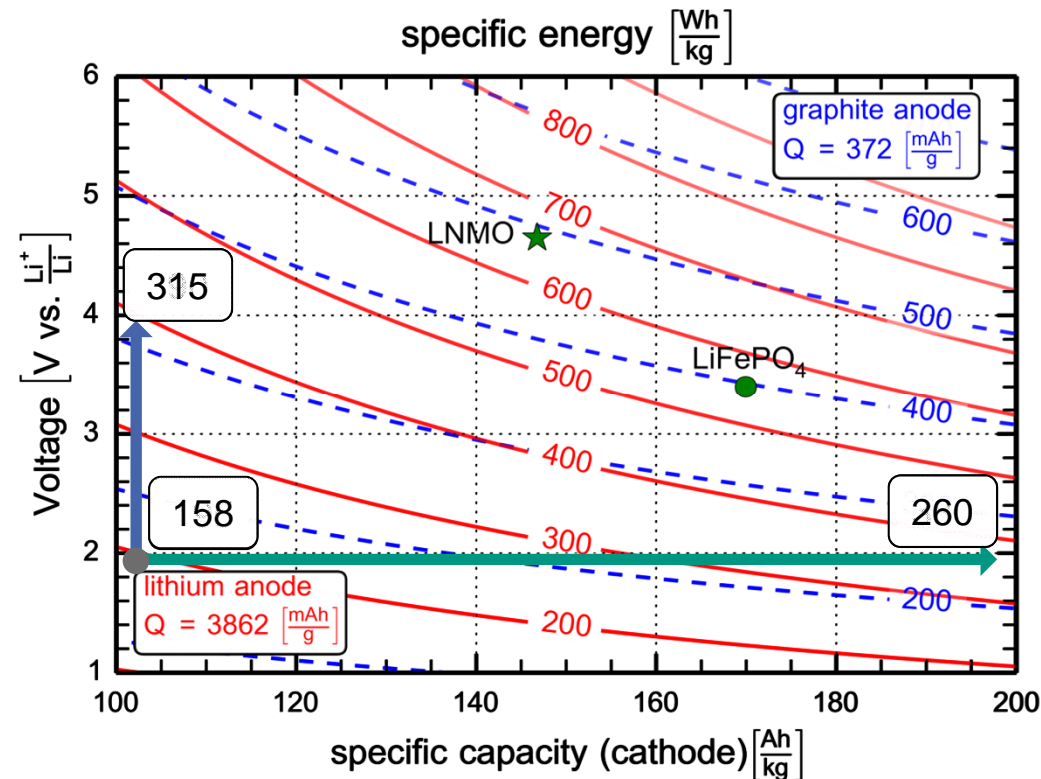
Why do we use high voltage spinels?

Options for improvement of the energy density

$$w = \frac{U \cdot n \cdot F}{M}$$

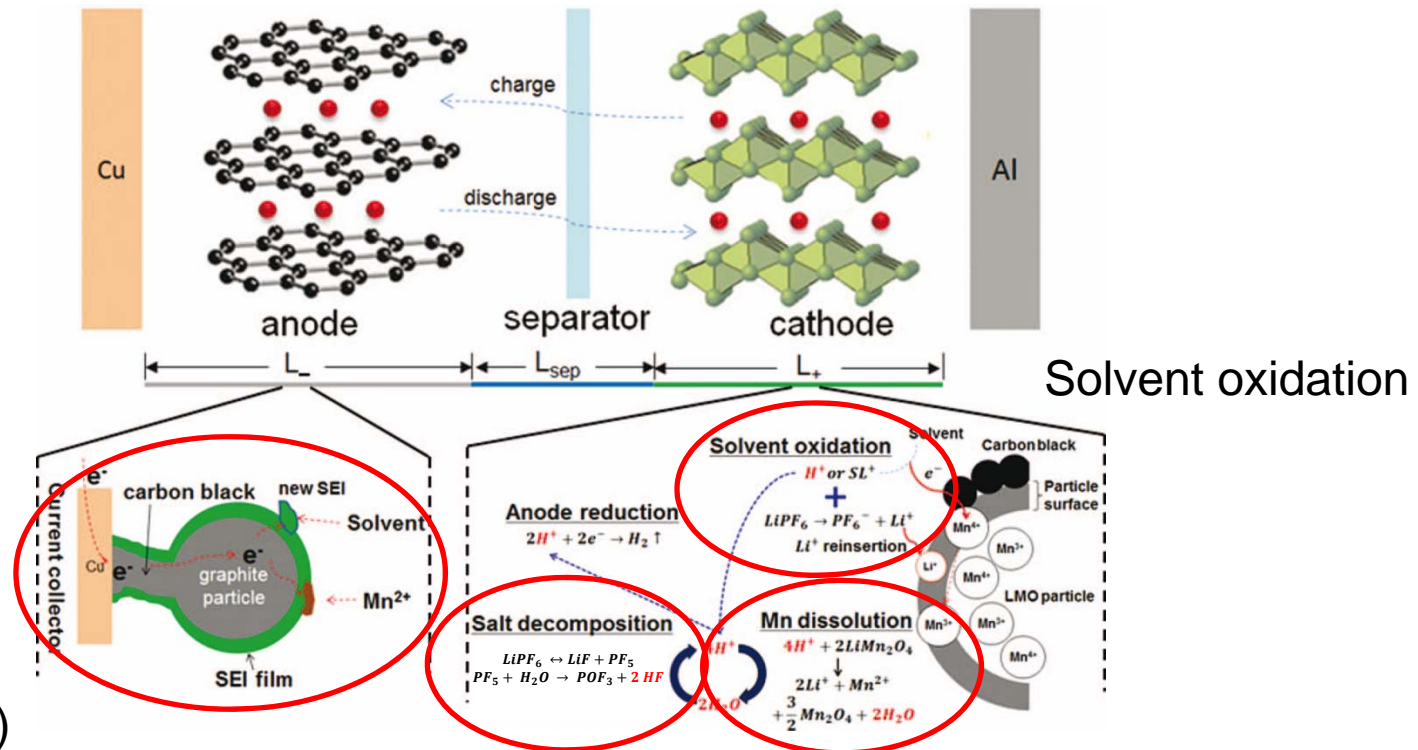
Increase of cathode capacity ($M = m_K + m_A$)
 Increase of cell voltage

	energy vs. Li [Wh kg ⁻¹]	energy vs. C [Wh kg ⁻¹]
LFP	545	391
LNMO	650	484



Fundamentals and Motivation

Degradation



Loss of Li (SEI)

Deposition of metal ions (e.g. Mn^{2+})

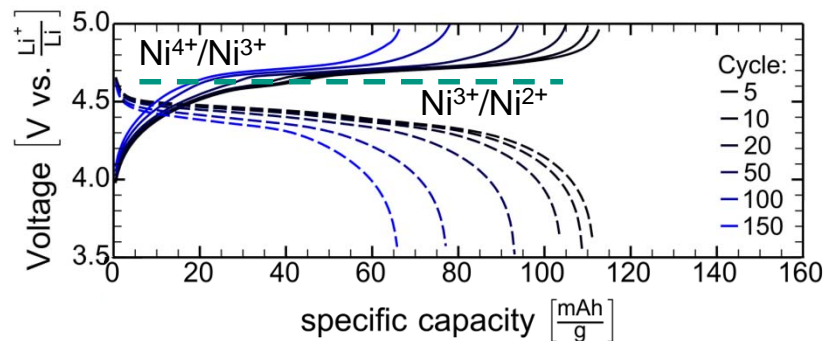
Dissolution of metal ions

Formation of HF and decomposition of AM

[after Lin et al., Journal of the Electrochemical Society, **160** (2013) A1701-A1710]

Fundamentals and Motivation

■ Example: 3 electrode cell (spinel vs. graphite)



■ Spinel/graphite:

- Both Ni redox couples observable
- No $\text{Mn}^{4+}/\text{Mn}^{3+}$ -plateau

■ Spinel/reference:

- Shift of the transition between the Ni redox couples to lower capacities
- End of Discharge at 4.6 V

■ Graphite/reference:

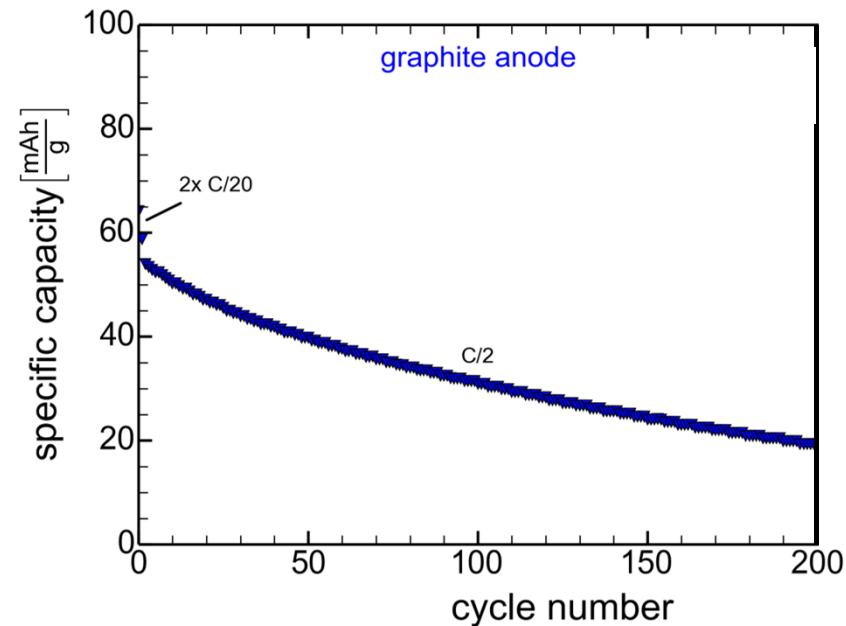
- Voltage rise when all lithium is deintercalated

➤ **Loss of active lithium**

cycling conditions: C/2, GF/C, LP30

Fundamentals and Motivation

■ Example: Change of anode



■ Rise of capacity

- Lithium metal anode compensates for loss of active lithium

- **No degradation of active material!!**
- **Coating of active material is essential!!**

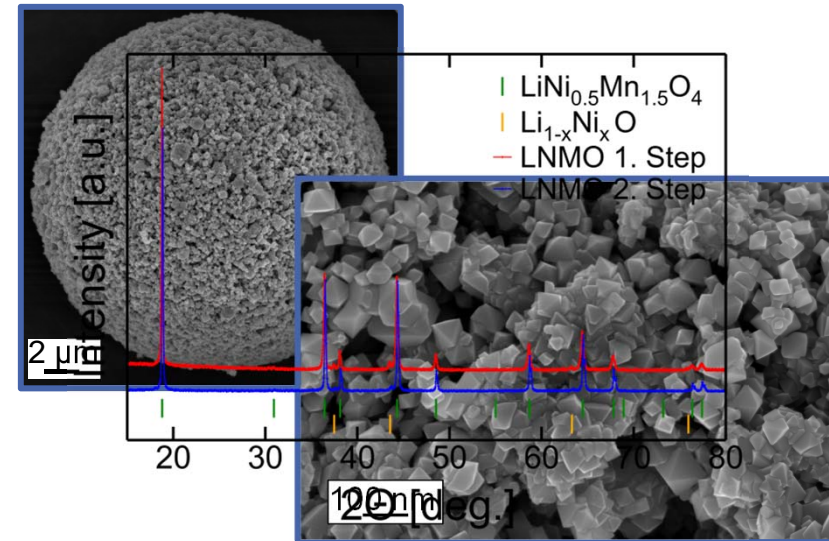
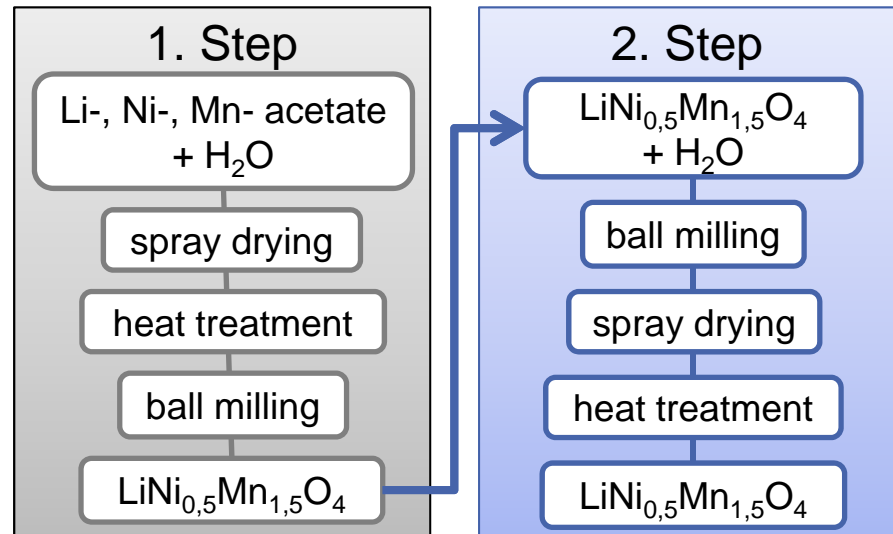
cycling conditions: GF/C, LP30

Coating: Requirements

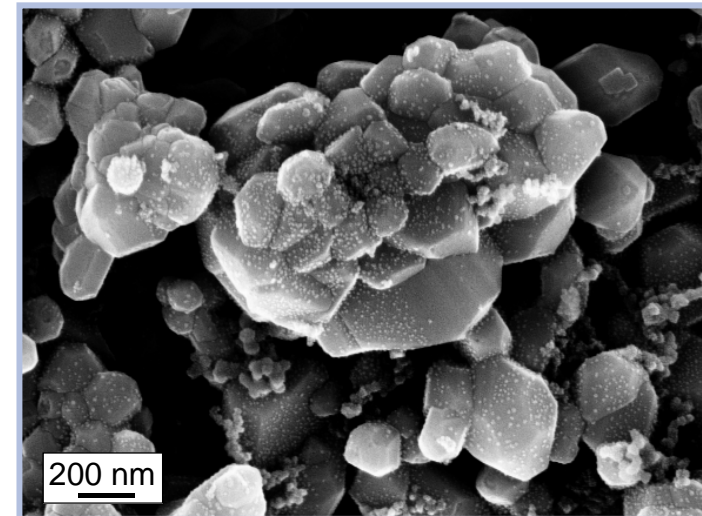
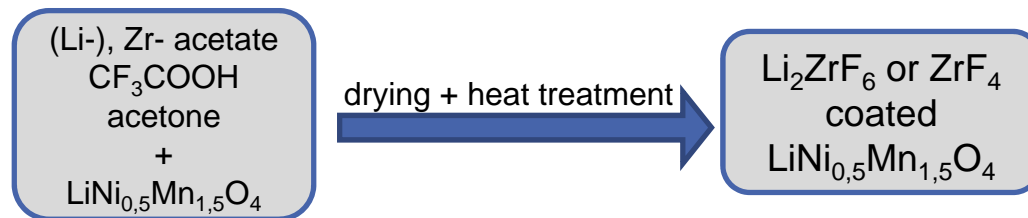
- Prevent dissolution of metal ions (reduce HF formation)
 - Resistance is reduced
- Electrochemical and chemical inactive
- Covering the surface of the active material
 - Inhibit parasitic reactions

[after Chen et al., J. Mater. Chem. **20** (2010) 7606-7612]

Synthesis

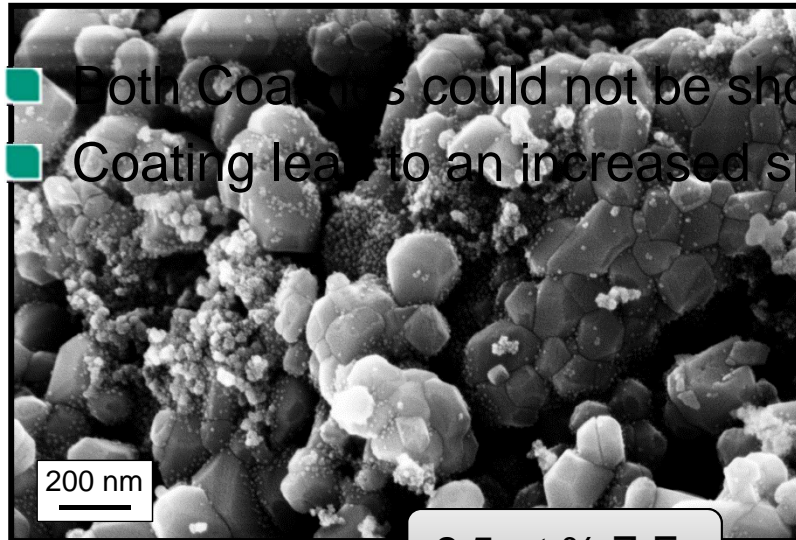


Surface modification with Li₂ZrF₆ and ZrF₄

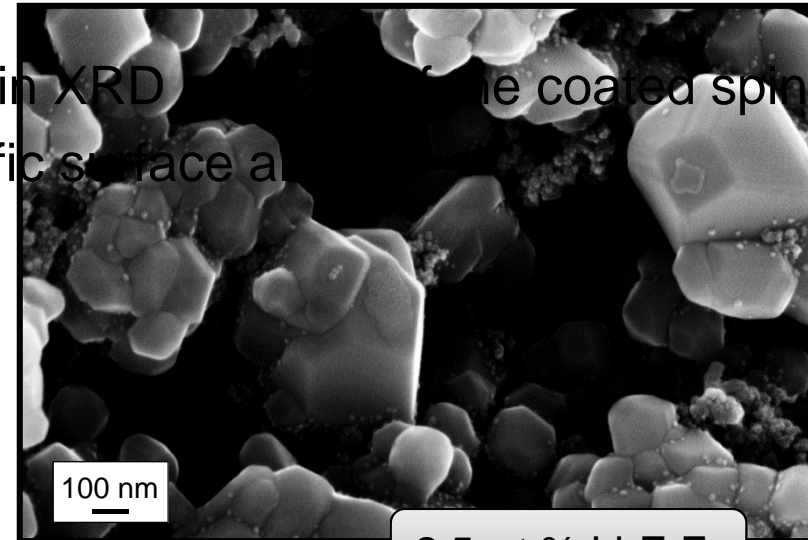


Powder Properties & Morphology

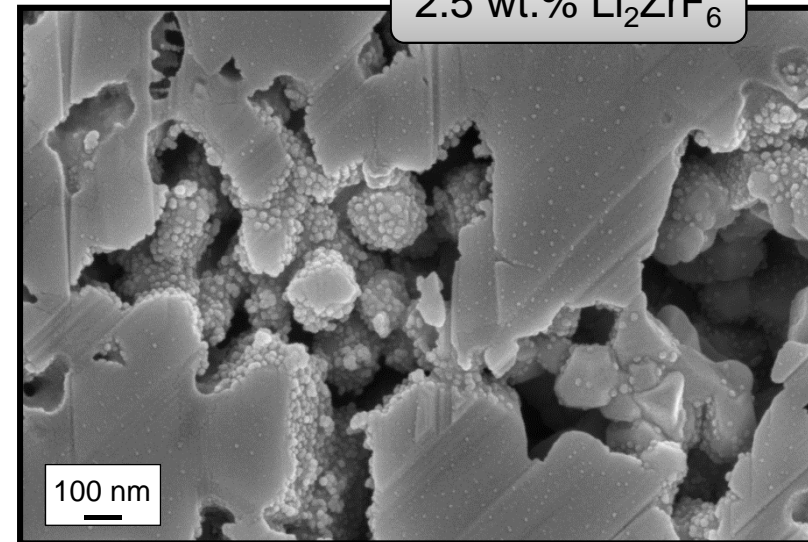
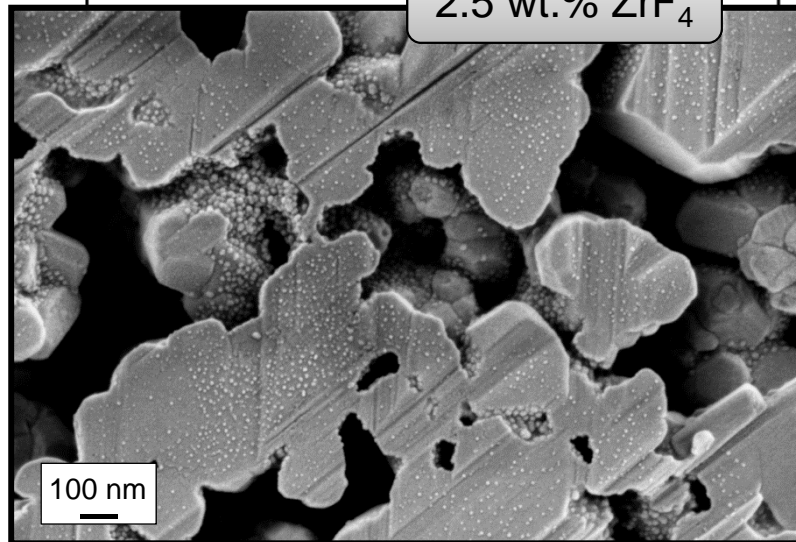
- Both Coatings could not be shown in XRD
- Coating leads to an increased specific surface area



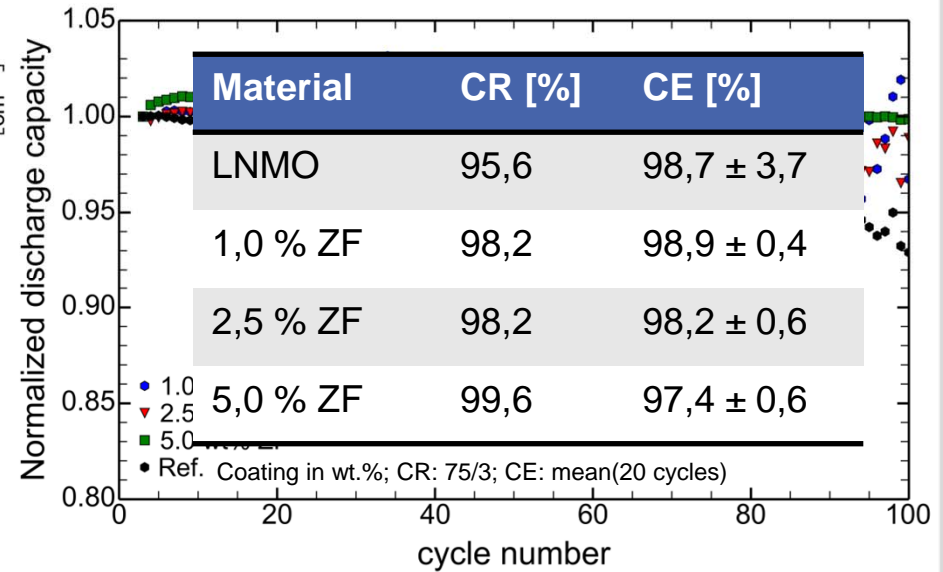
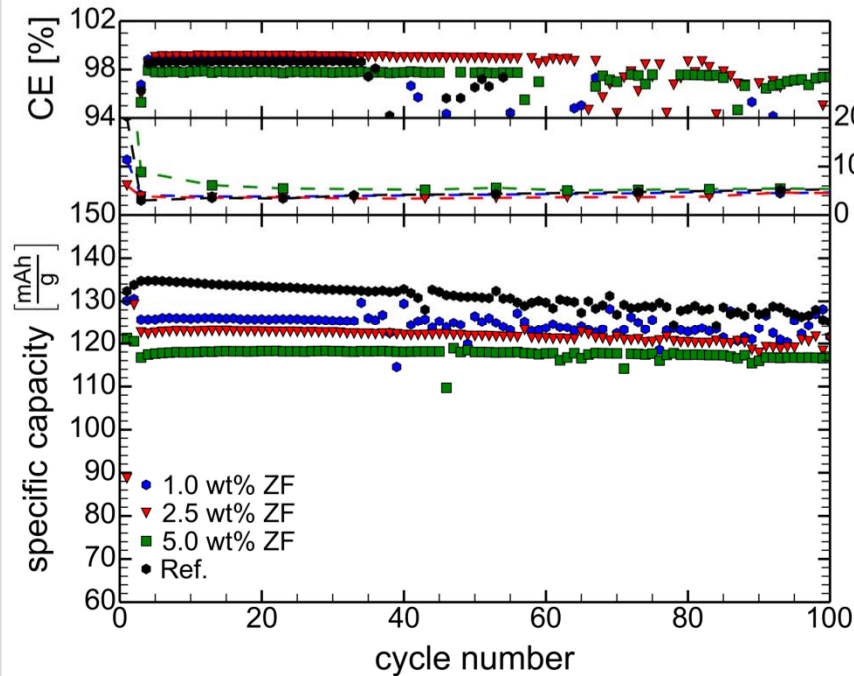
2.5 wt.% ZrF_4



2.5 wt.% Li_2ZrF_6



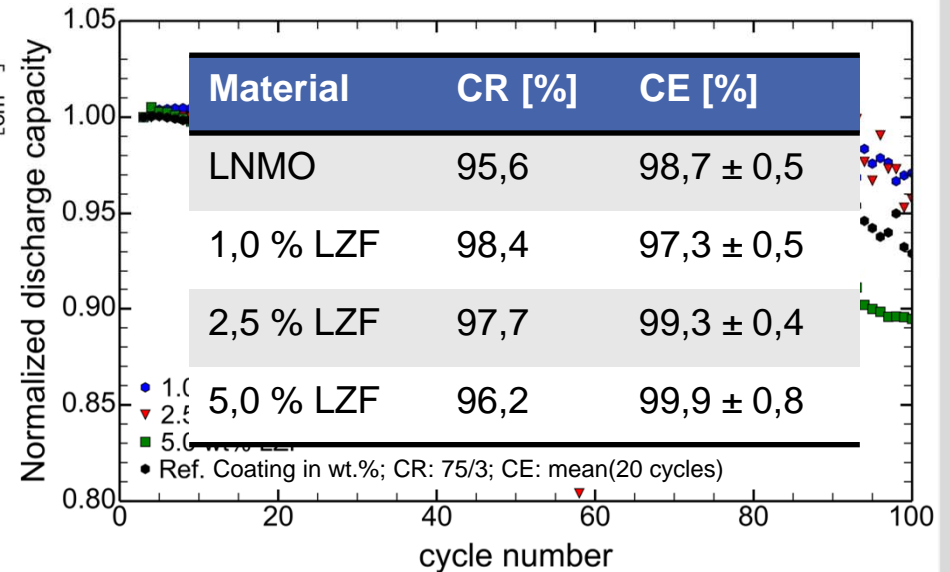
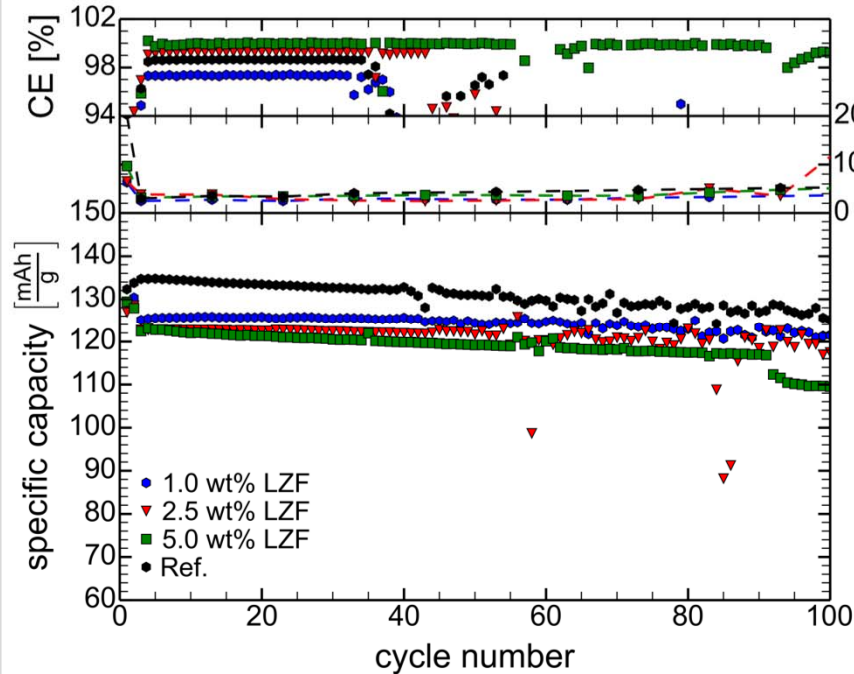
ZrF₄ – Cycling Stability (C/2 @20°C)



- Initial capacity decreases with coating content
- Capacity loss can be reduced by coating

cycling conditions: GF/C, LP30

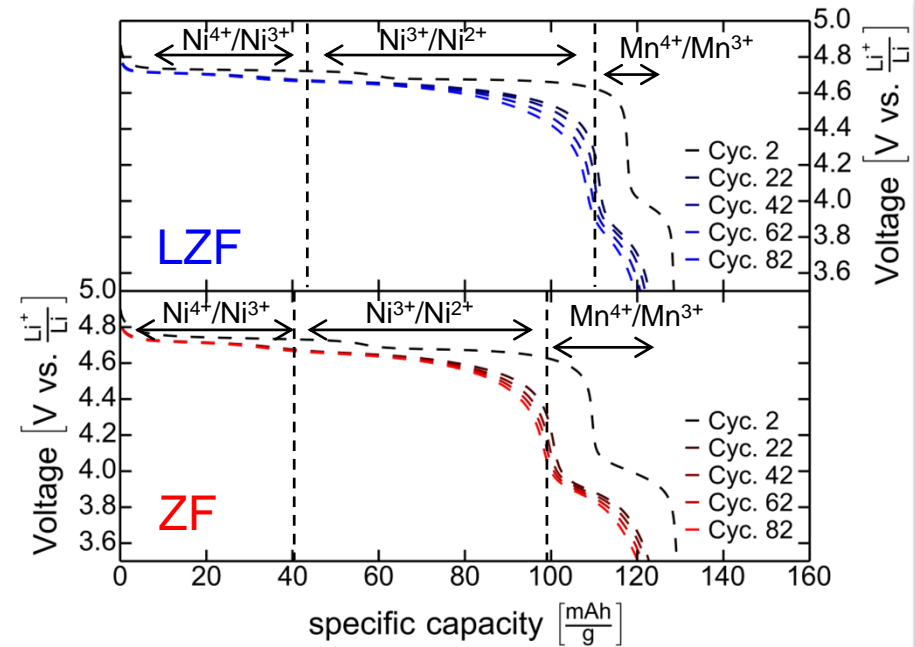
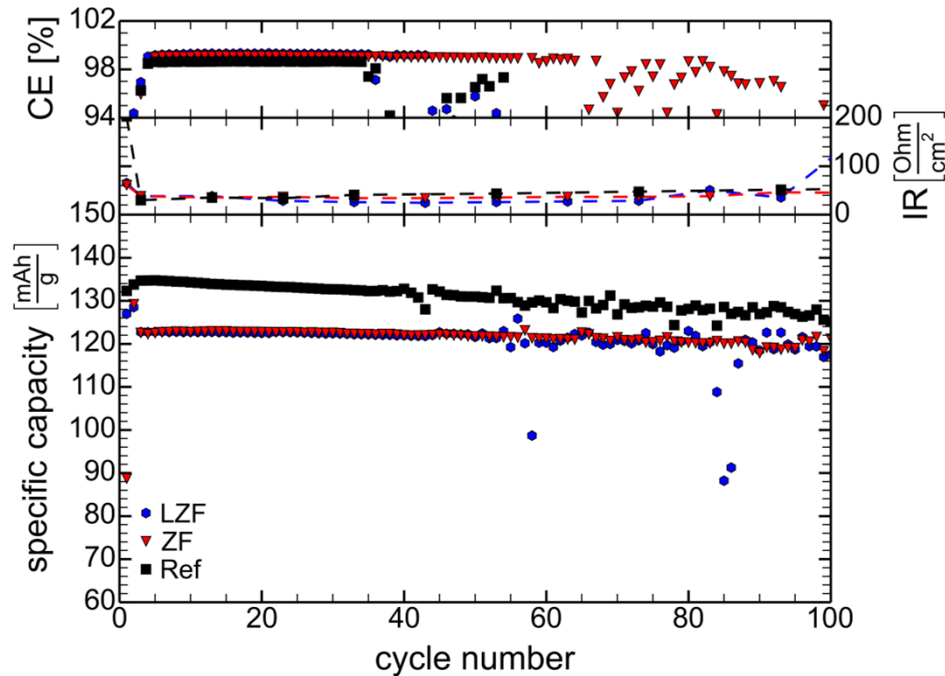
Li₂ZrF₆ – Cycling Stability (C/2 @20°C)



- Similar to the ZrF₄ coating, the initial capacity decreases
- Coating with 1.0 or 2.5 wt% Li₂ZrF₆ lead to reduced capacity fade

cycling conditions: GF/C, LP30

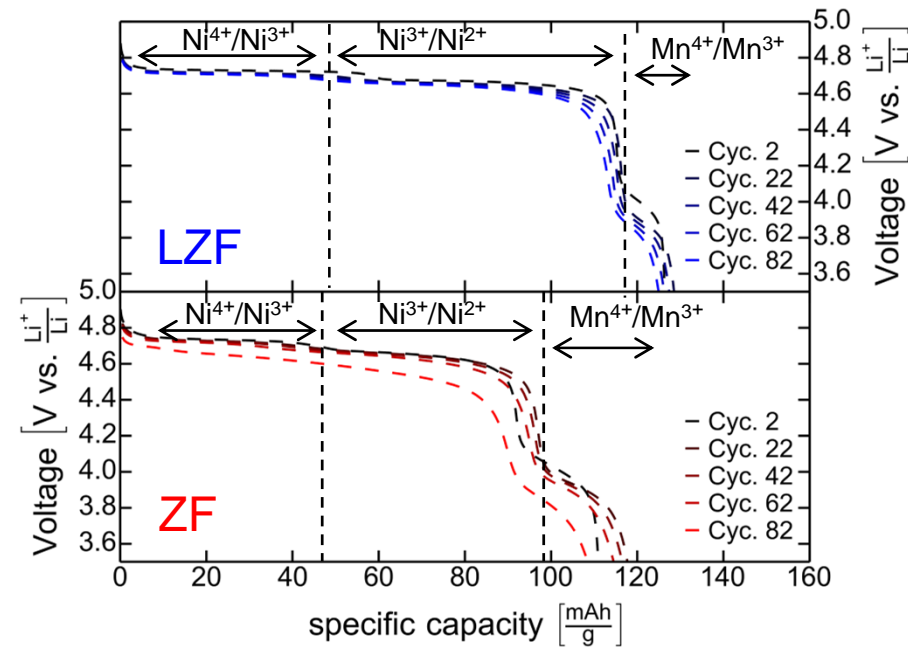
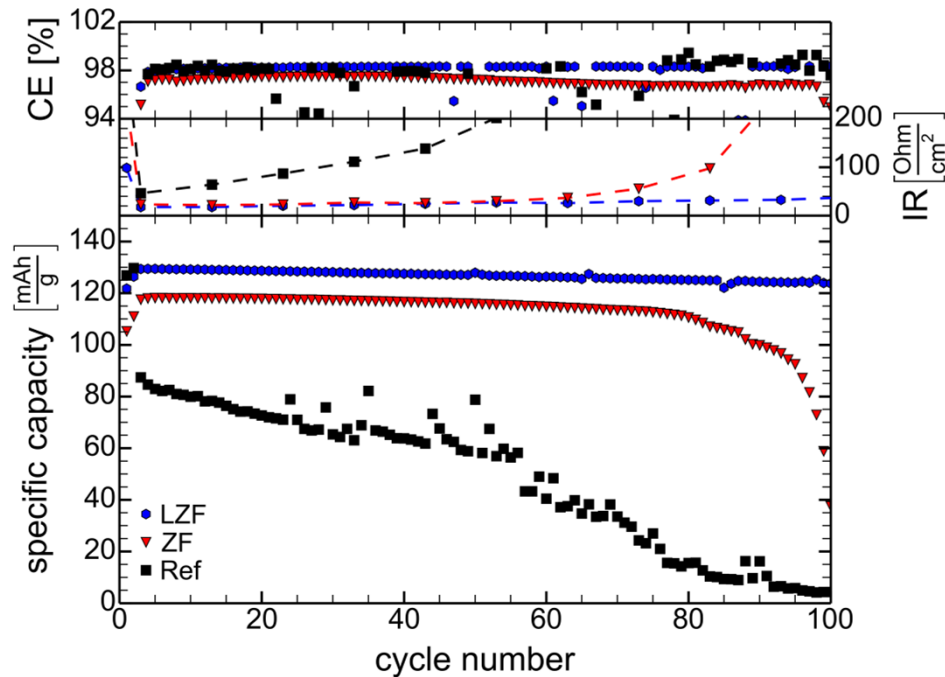
Comparison of 2.5 wt.% Coatings (C/2 @20°C)



- Coating leads to lower initial capacities
- The capacity loss per cycle can be reduced by coating
- Changes in the electrochemistry occur due to partial doping

cycling conditions: GF/C, LP30

Comparison of 2.5 wt.% Coatings (C/2 @40°C)

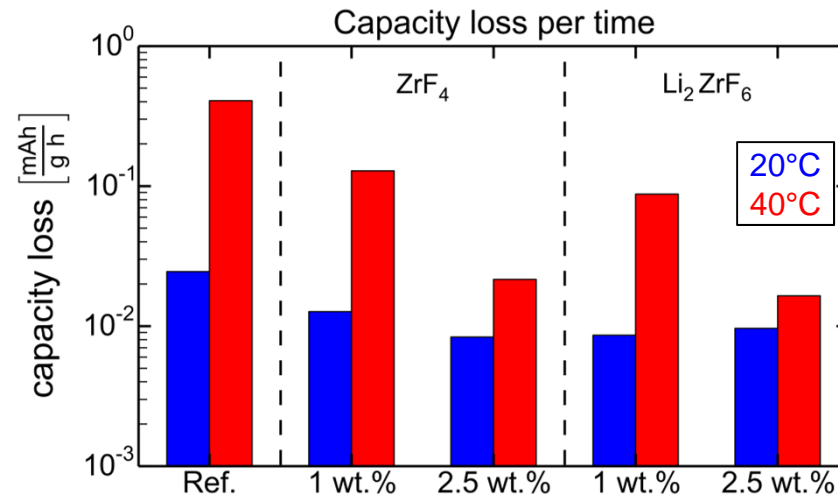


- The uncoated spinel suffers from severe capacity fade
- Coated samples exhibit excellent capacity retention
- Abrupt capacity loss occurs because of a rising internal resistance

cycling conditions: GF/C, LP30

Conclusion

- Coating with ZrF_4 and Li_2ZrF_6 was performed by a simple sol-gel route
- Significant improvement of capacity retention is achieved



- High temperatures during coating process can lead to partial doping
 - Influences stability
 - Can have an impact on energy density

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